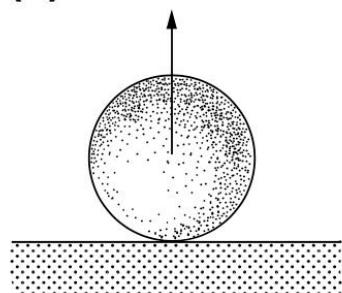
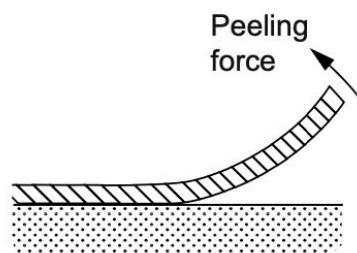


Interaction and force measurements

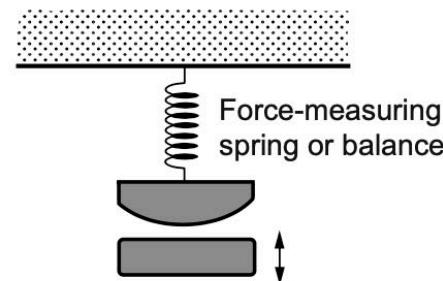
(a) Adhesion force



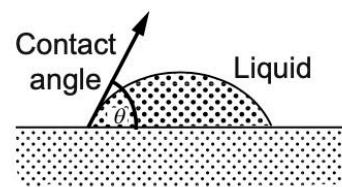
(b)



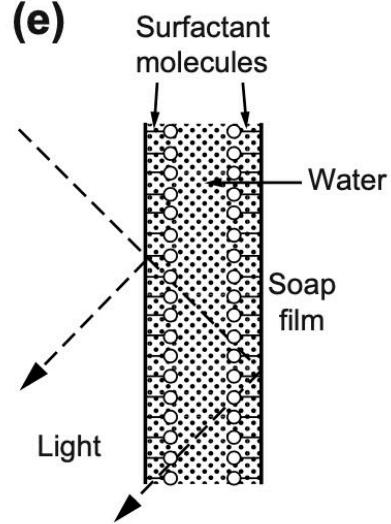
(c)



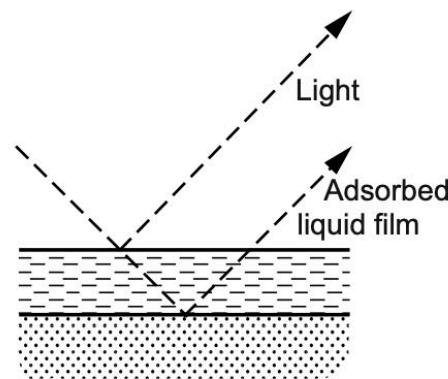
(d)



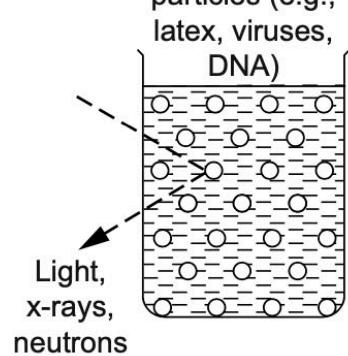
(e)



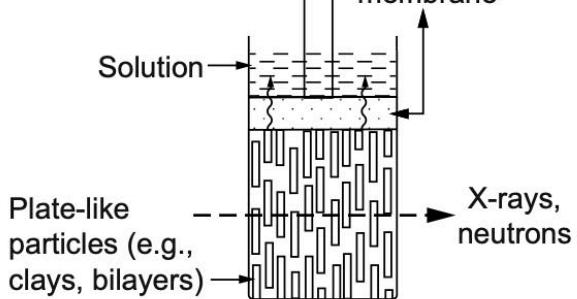
(f)



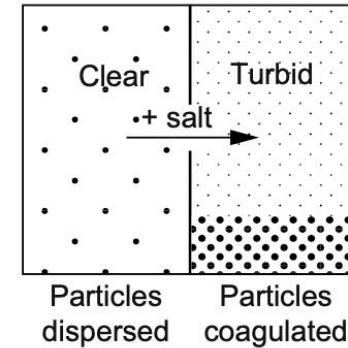
(g) Large colloidal particles (e.g., latex, viruses, DNA)



(h) Piston
Semi permeable membrane
Solution
Plate-like particles (e.g., clays, bilayers)



(i) Stable colloid
Unstable colloid
Clear
Turbid
+ salt
Particles dispersed
Particles coagulated



Direct force measurements

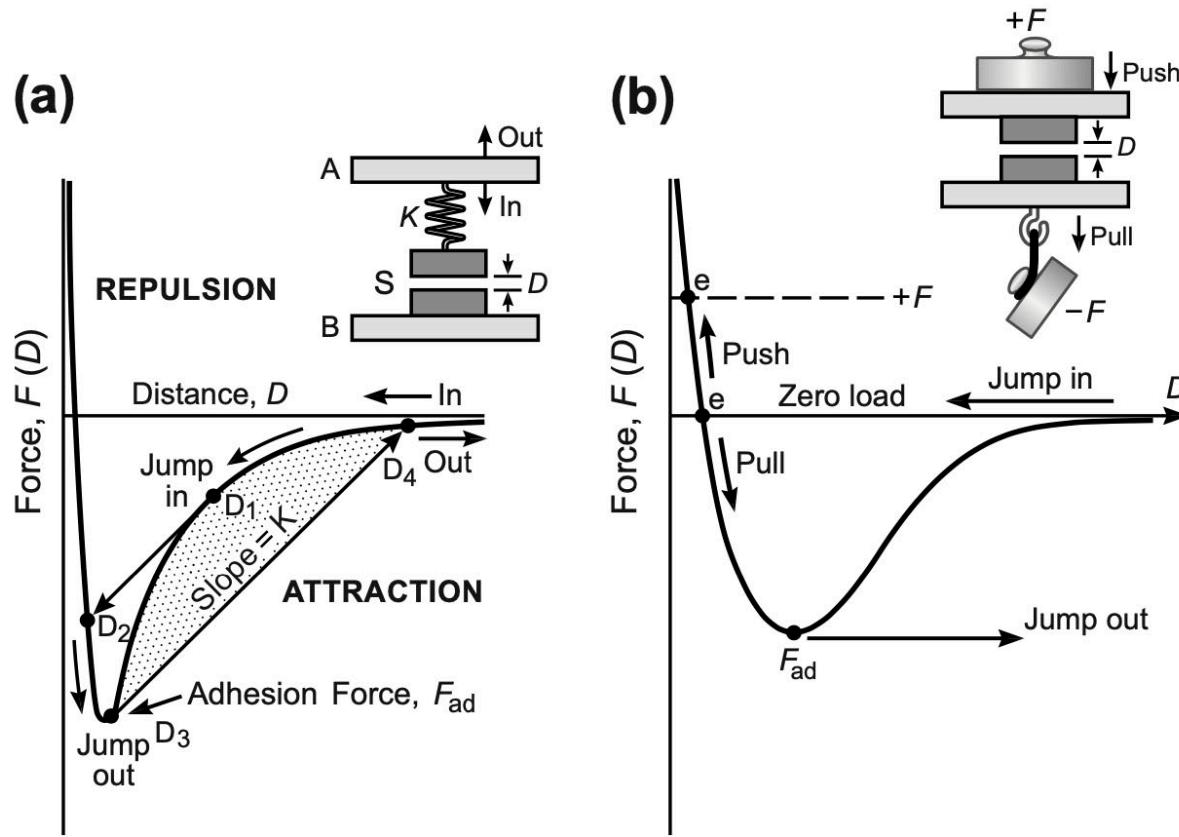


FIGURE 12.3 Measuring forces via (a) a spring (externally applied force \propto spring displacement: $(F_{\text{ext}} = K\Delta D)$, and (b) a constant force ($F_{\text{ext}} = \text{constant}$). For the typical force-law or force-distance curves $F(D)$ shown, instabilities occur at the positions shown, manifested by an inward jump on the way in (on approach) and an outward jump on the way out (on separation). (a) These jump instabilities will occur only if the effective spring constant K is lower than the maximum slope of the $F(D)$ curve. Thus, if the system is infinitely stiff ($K = \infty$) there will be no instabilities on approach or separation. In contrast, if a constant external force $+F$ is applied to one of the surfaces (b), K is effectively zero, and jumps will occur to the upper equilibrium points at e on approach, and out from F_{ad} on separation. No equilibrium is possible at surface separations between the **Jump in** and **Jump out** points.

Table 12.1 Typical Length Scales and Magnitudes for Noncovalent Interactions and the Ten Most Common Techniques for Measuring Them

Typical Parameters	Atomic (physical bonds, atoms and small molecules, submolecular groups)	Nanoscopic (macromolecules, nanoparticles, small aggregates)	Microscopic (small colloidal particles, vesicles, biological cells)	Macroscopic (large particles, flat surfaces)
Particle radius	0.1 nm	1 nm	1 μ m	1 mm
Binding or adhesion energy	$1 kT$	$10 kT$	$1000 kT$	$10^6 kT$, $100 \text{ mJ}/\text{m}^2$
Adhesion force or pressure	$10^{-11} \text{ N} = 10 \text{ pN}$	$10^{-10} \text{ N} = 100 \text{ pN}$	$10^{-7} \text{ N} = 100 \text{ nN}$	$10^{-4} \text{ N} = 100 \mu\text{N}$ $10^8 \text{ N/m}^2 = 10^3 \text{ atm}$
Range of interaction ^a	0.2 nm	2 nm	20 nm	50 nm
Suitable Experimental Techniques ^b				
Atomic Force Microscopy (AFM)	☆	☆	☆	
Micro Cantilever (MC)	with tip	with tip	with bead	
Optical Tweezers or Optical Trapping (OT)	☆	☆	☆	
Micro Pipette Aspiration (MPA) ^c and Bio Force Probe (BFP) ^c	☆	☆	☆	
Total Internal Reflection Microscopy (TIRM)			☆	
Reflectance Interference Contrast Microscopy (RICM) ^c				☆
Surface Forces Apparatus (SFA)				☆.
Osmotic Pressure (OP) or Osmotic Stress (OS)				☆
Shear Flow Detachment (SFD) ^c			☆	☆

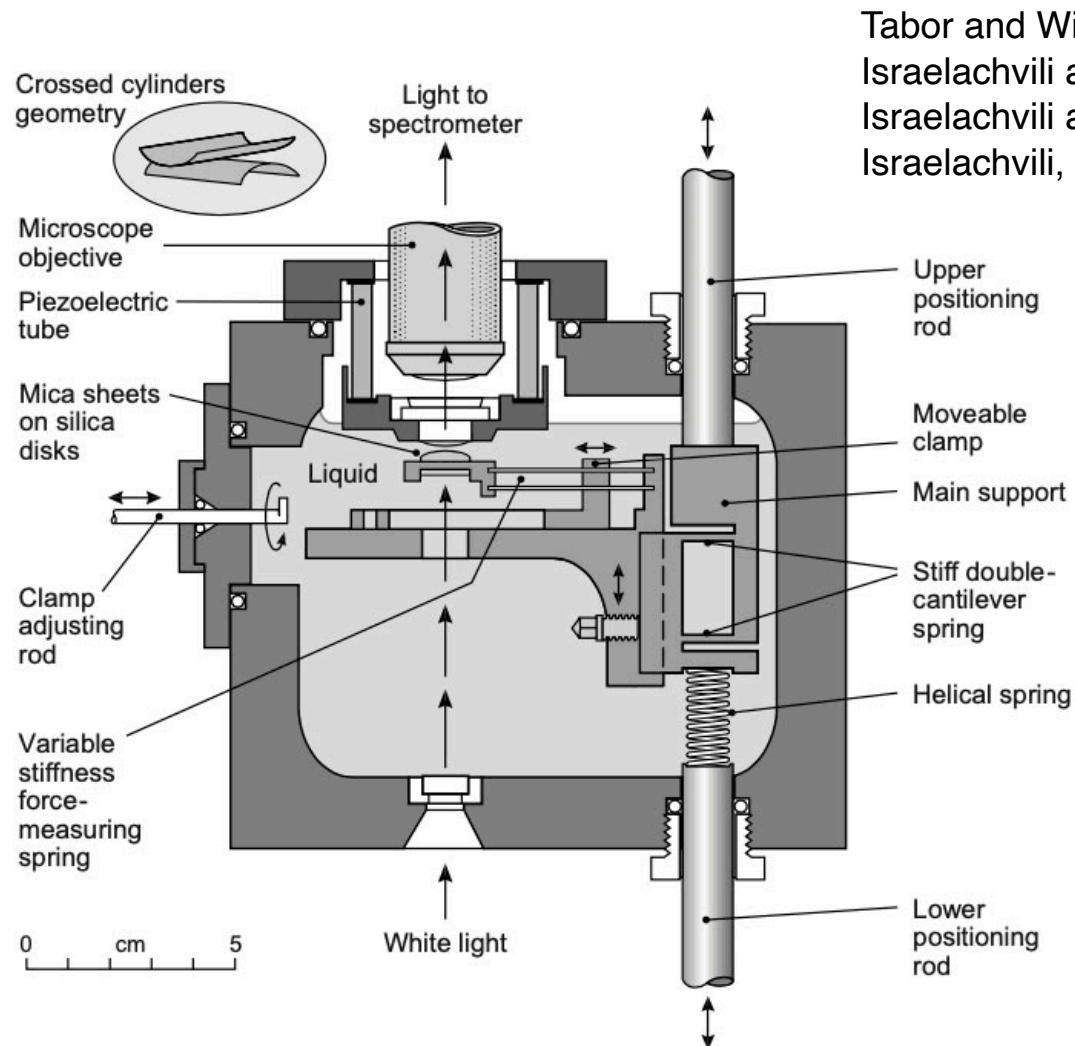
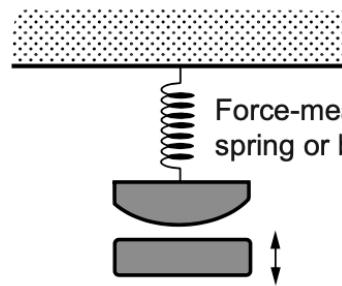
^aRange is here defined as the distance D at which the interaction energy $W(D)$ is of the same order as the thermal energy $kT \approx 4 \times 10^{-21} \text{ J}$ at 25°C .

^bSome techniques can only be used to measure attractive (e.g., adhesion) forces, some only repulsive forces, and some can measure both.

^cThe MPA, BFP, RICM and SFD techniques are devoted mainly to biological-type measurements—for example, between vesicles or living cells, described in Part III.

Table adapted from Leckband and Israelachvili (2001).

Surface forces apparatus (SFA)



Tabor and Winterton (1969)
Israelachvili and Tabor (1972, 1973)
Israelachvili and Adams, 1978
Israelachvili, 1987b

Two crossed-cylinders

Mica surfaces in air, vacuum, or liquid

Multiple Beam Interferometry (MBI) using “fringes of equal chromatic order” (FECO)

SFA measurement

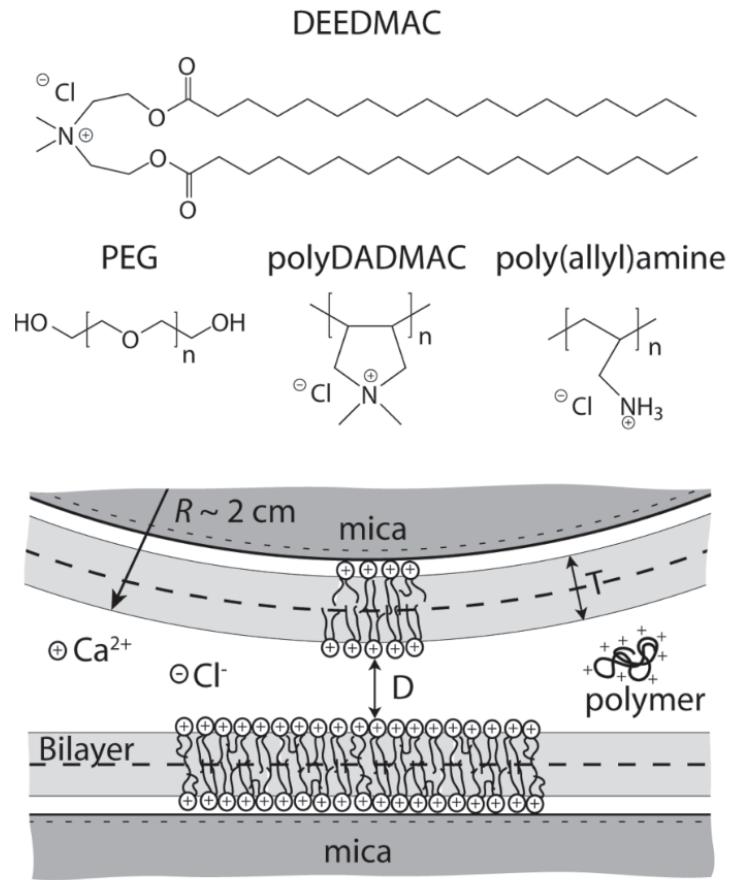


Figure 1. Top: The cationic surfactant DEEDMAC used to make the supported bilayers in the SFA studies. Middle: The polymers PEG, polyDADMAC, and poly(allyl)amine used to study the repulsive electrostatic and attractive depletion forces between DEEDMAC bilayers. Bottom: Schematic of the supported bilayers and surface geometry in the SFA experiments.

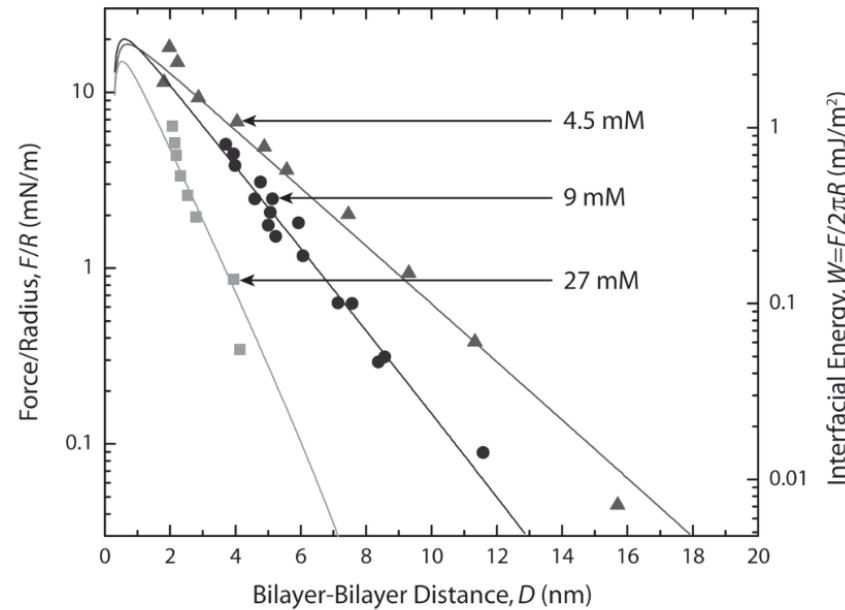
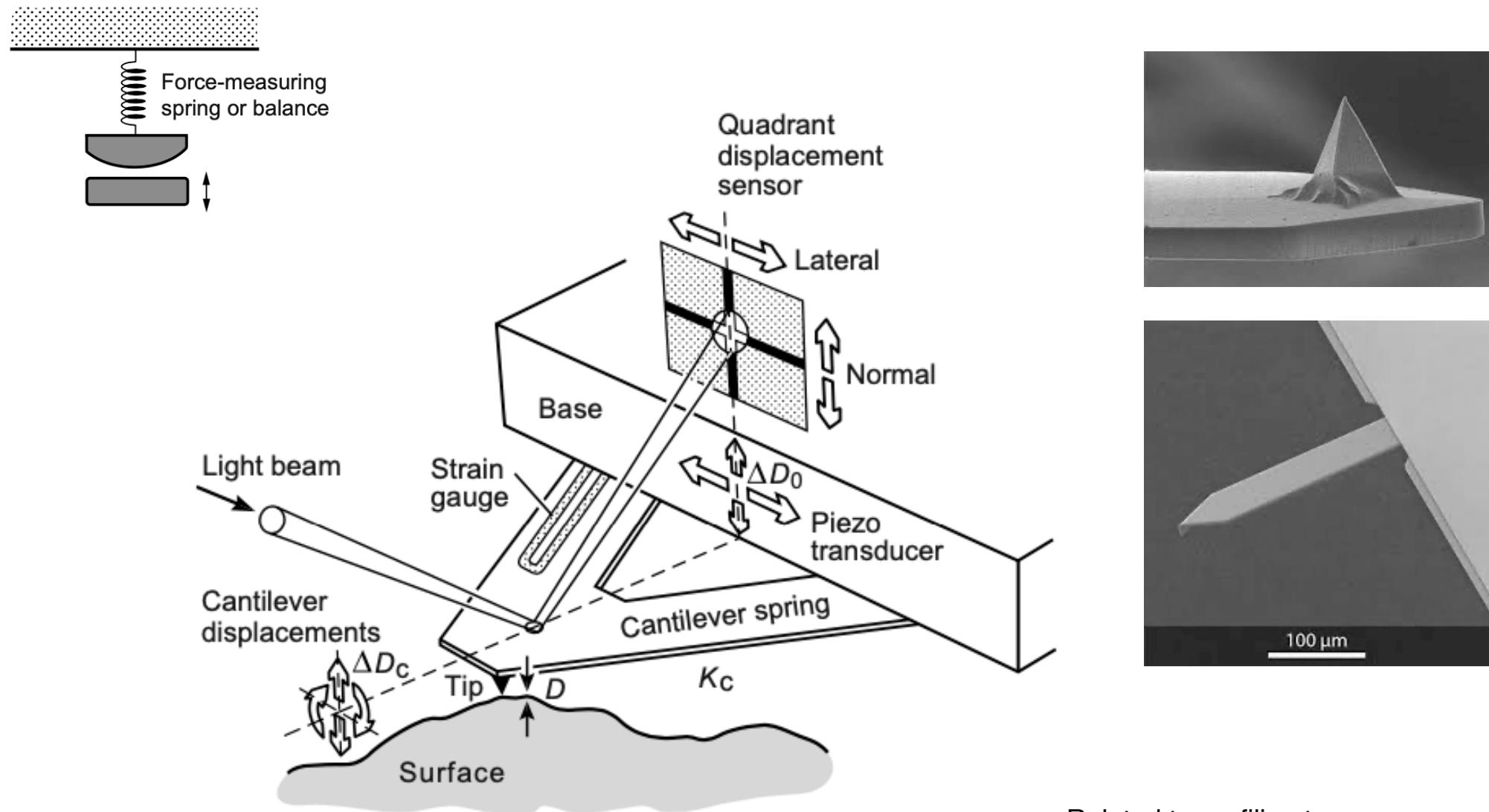


Figure 2. Measured forces between DEEDMAC bilayers in various CaCl_2 solutions at pH 4. Solid lines are DLVO fits using the parameters given in Table 1.

Atomic force microscopy (AFM)



Related to profilometry
Scanning mode or tapping mode

AFM scans of surface topology

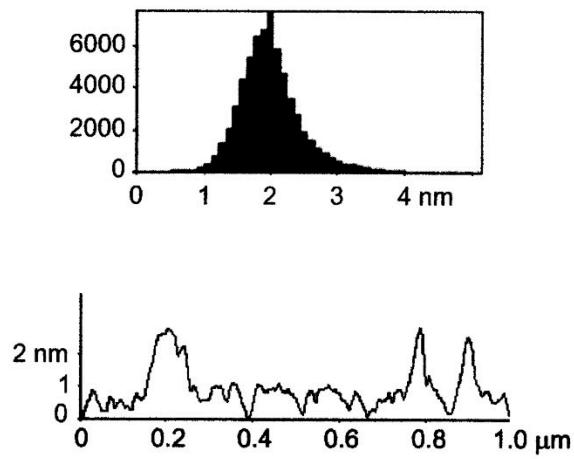
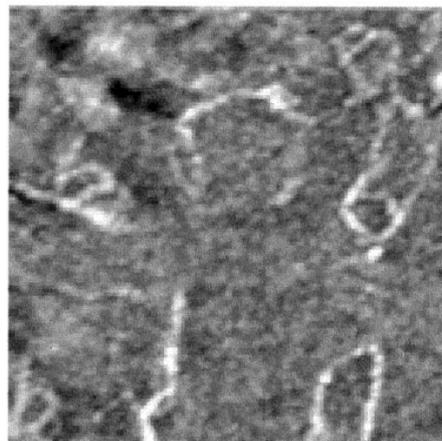


Figure 5. AFM scan of roughness on the surface of a glass slide within a typical $1\text{ }\mu\text{m}$ square. The histogram shows the distribution of elevations on the 2D image, and the line graph shows the profile of elevations along a horizontal line bisecting the 2D scan.

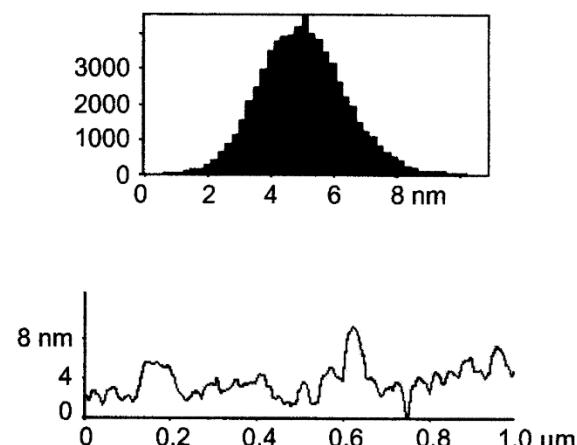
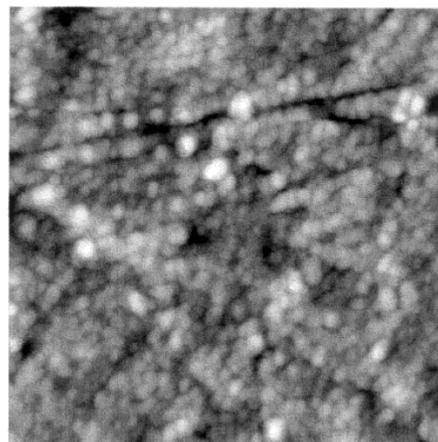


Figure 6. AFM scan of roughness on the surface of a glass slide, spin-coated with a $1\text{ }\mu\text{m}$ thick PS film, within a typical $1\text{ }\mu\text{m}$ square. The histogram shows the distribution of elevations on the 2D image, and the line graph shows the profile of elevations along a horizontal line bisecting the 2D scan.

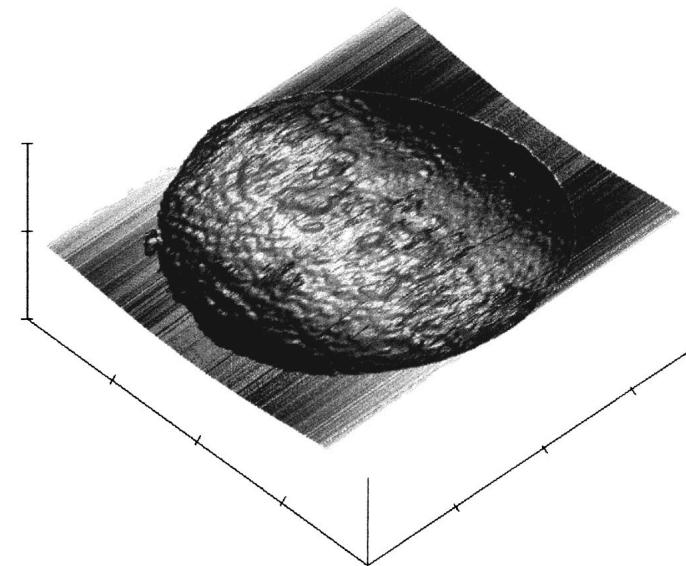


Figure 7. AFM scan of the top surface of a PS sphere (courtesy of Robert Considine). The lateral scale is $1\text{ }\mu\text{m}/\text{division}$, and the vertical scale is $1.5\text{ }\mu\text{m}/\text{division}$. The larger asperities have a height of 40 nm.

M.A. Bevan, D.C. Prieve, Direct Measurement of retarded van der Waals Attraction. *Langmuir*. 15, 7925–7936 (1999).

Colloidal probe AFM

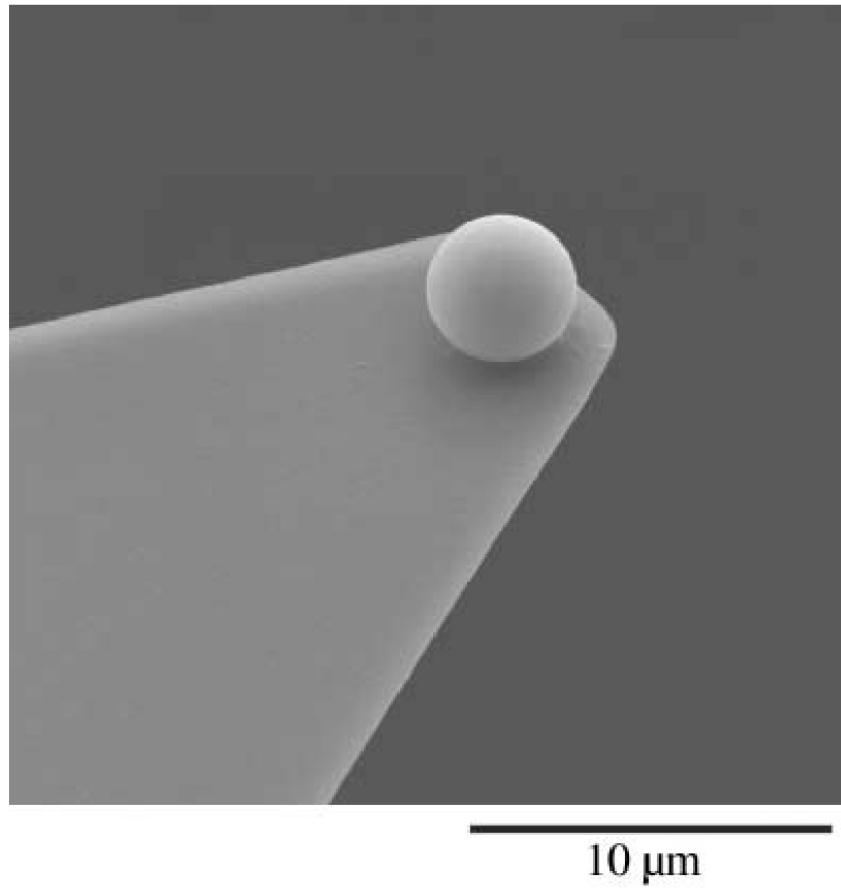
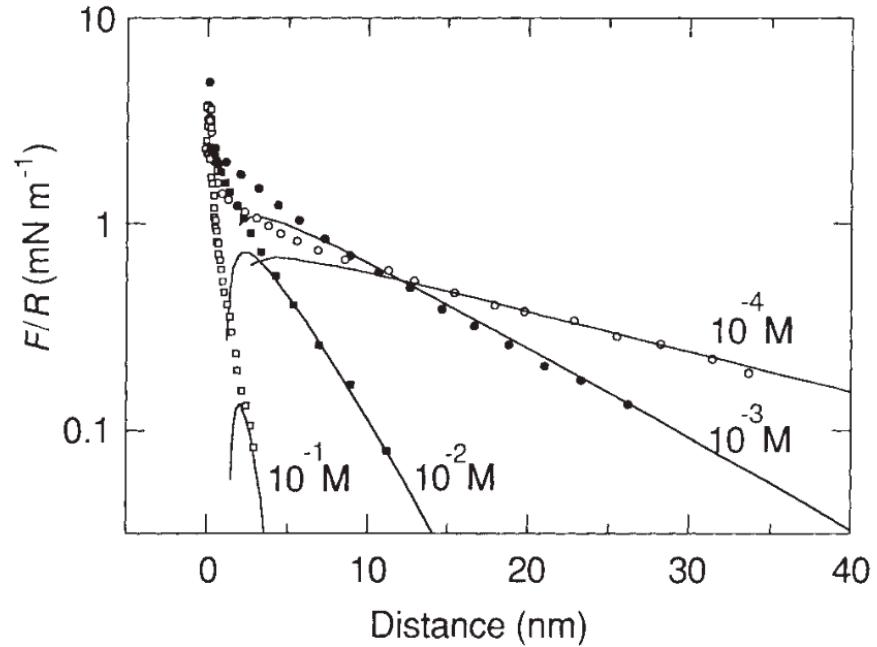


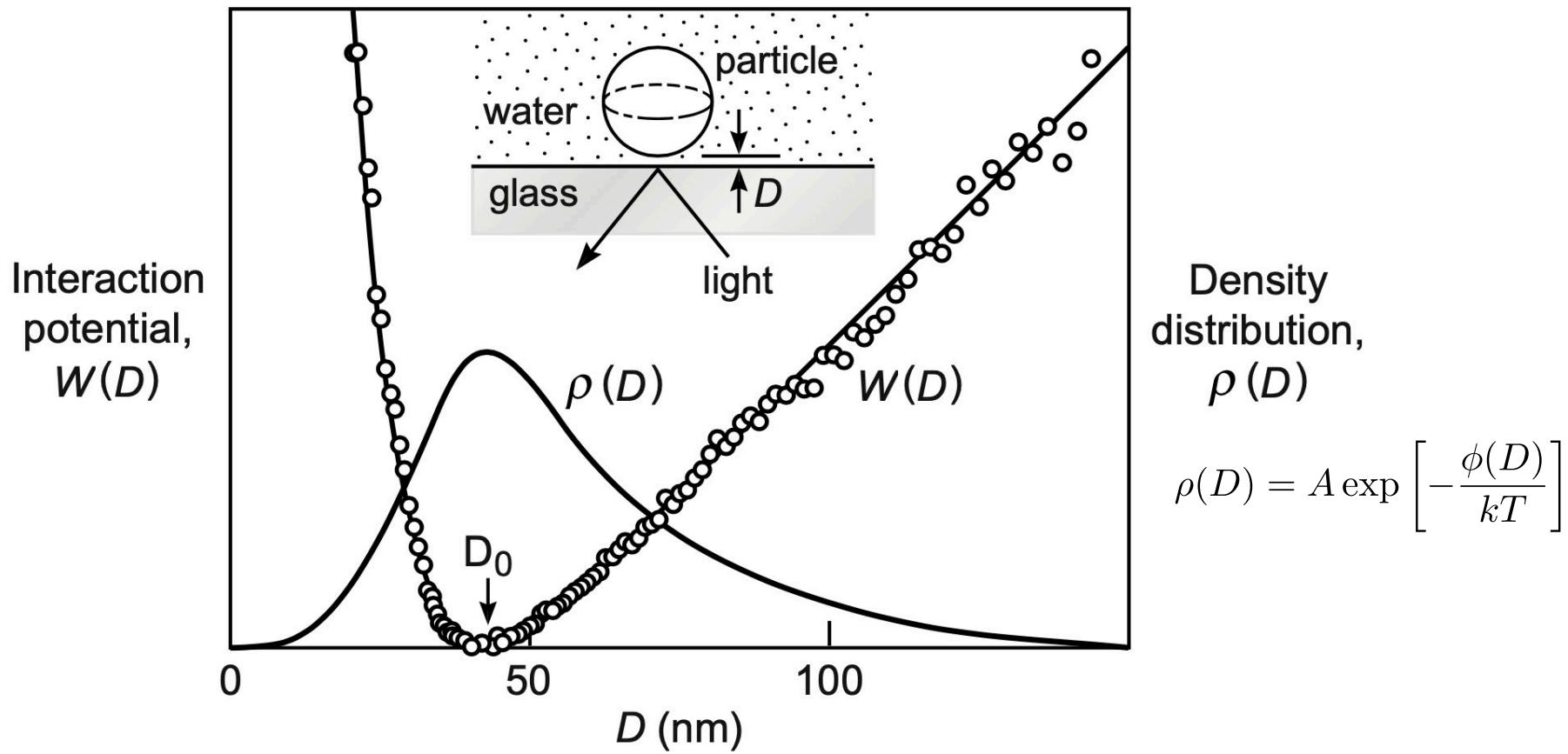
Fig. 1: Scanning electron micrograph of silanized silica microsphere glued to the end of a tipless atomic force microscope cantilever.



W.A. Ducker, T.J. Senden, R.M. Pashley, *Nature*. 353, 239–241 (1991).
M. Kappl, H.-J. Butt, *Part. Part. Syst. Charact.* 19, 129–143 (2002).

Total internal reflection microscopy (TIRM)

Prieve, 1999
Prieve and Frej, 1990
Prieve et al., 1990



TIRM measurement

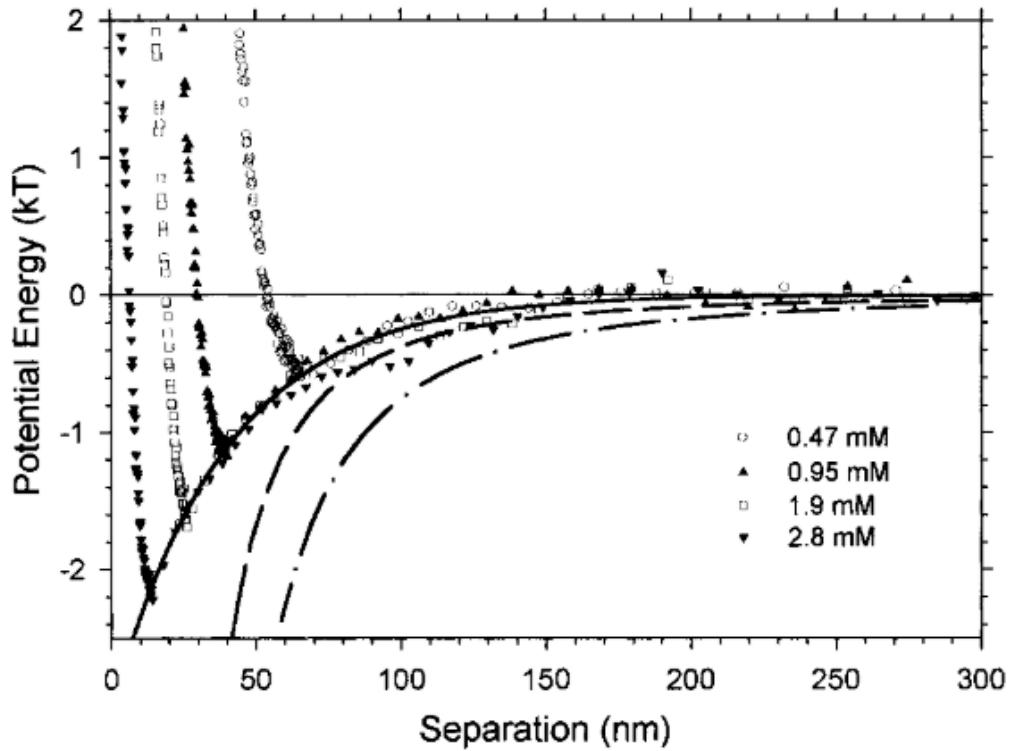


Figure 1. Profiles of the colloid interactions between the same $6\text{ }\mu\text{m}$ PS sphere and a glass slide in solutions having different ionic strengths. The solid line is an empirical fit given by eq 5, the dashed line shows the van der Waals interaction predicted from the Lifshitz–Derjaguin equation (eq 3), and the dot–dash line shows the van der Waals interaction predicted from the Lifshitz–Hamaker equation (eq 4).

$$E(l) = -\frac{A_{132}(l)}{12\pi l^2} \quad (1)$$

$$A_{132}(l) = -\frac{3}{2}kT \sum_{n=0}^{\infty}' \int_{r_n}^{\infty} x \{ \ln[1 - \Delta_{13}\Delta_{23}e^{-x}] + \ln[1 - \bar{\Delta}_{13}\bar{\Delta}_{23}e^{-x}] \} dx$$

$$\Delta_{jk} = \frac{\epsilon_j s_k - \epsilon_k s_j}{\epsilon_j s_k + \epsilon_k s_j} \quad \bar{\Delta}_{jk} = \frac{s_k - s_j}{s_k + s_j}$$

$$s_k^2 = x^2 + \left(\frac{2\xi_n l}{c} \right)^2 (\epsilon_k - \epsilon_2)$$

$$r_n = \frac{2l\xi_n \sqrt{\epsilon_2}}{c} \quad \epsilon_k = \epsilon_k(i\xi_n) \quad \xi_n = \frac{n k T}{\hbar}$$

$$\phi(h) = 2\pi a \int_h^{\infty} E(l) dl \quad (3)$$

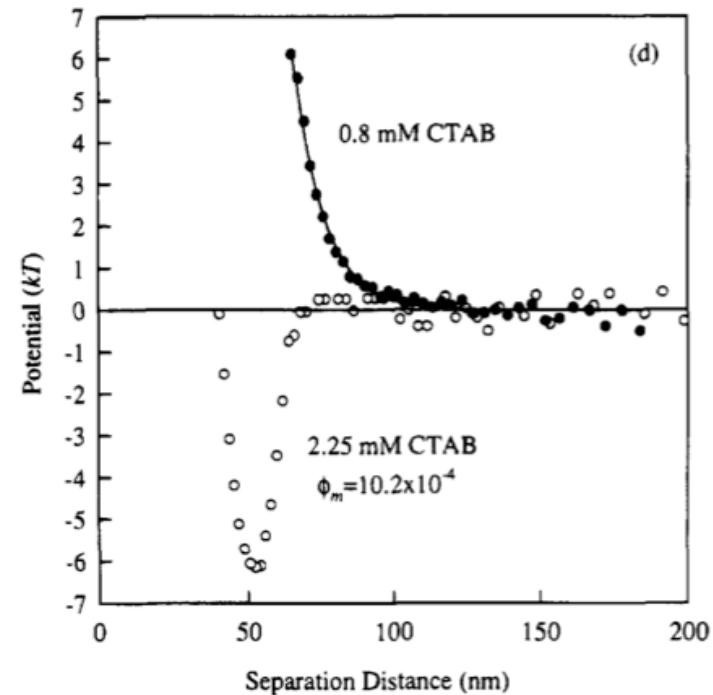
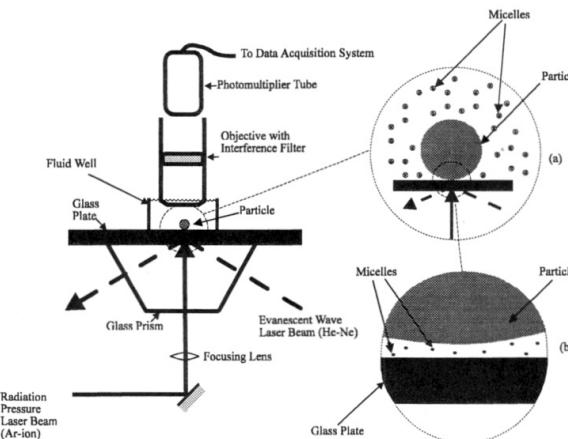
$$\phi(h) = -\frac{A(h)}{6} \left\{ \frac{2a}{h} \frac{h+a}{h+2a} - \ln \left(\frac{h+2a}{h} \right) \right\} \quad (4)$$

M.A. Bevan, D.C. Prieve, Direct Measurement of retarded van der Waals Attraction. *Langmuir*. 15, 7925–7936 (1999).

Interaction measurements

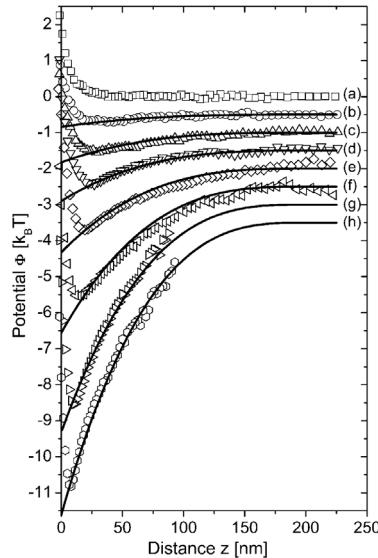
TIRM, surfactant micelles

Sober, D. L.; Walz, J. Y. *Langmuir*
1995, 11, 2352–2356.



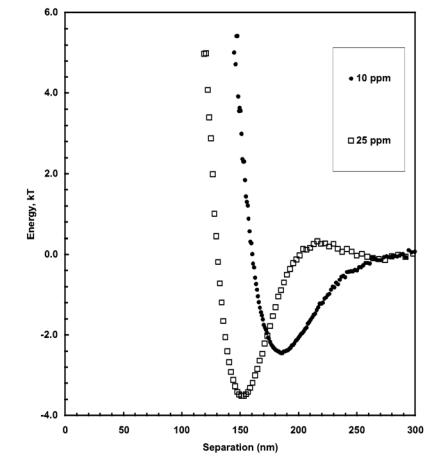
TIRM, rigid rods

Helden, L.; Roth, R.; Koenderink,
G. H.; Leiderer, P.; Bechinger, C.
Phys. Rev. Lett. 2003, 90, 48301.

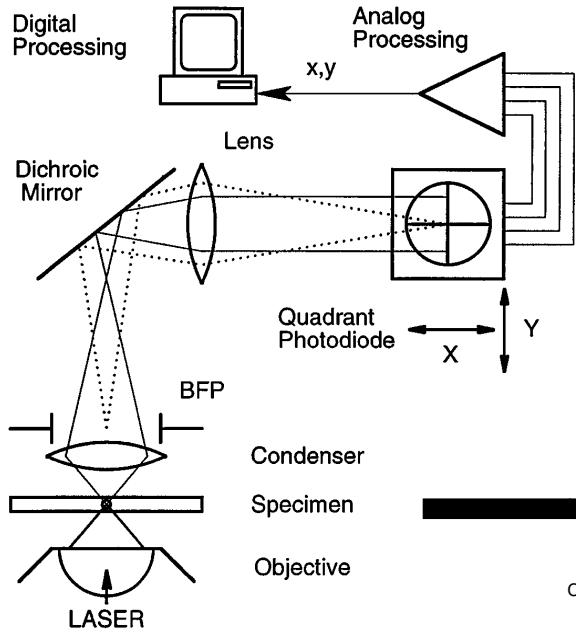


TIRM, polyelectrolytes

Biggs, S.; Dagastine, R. R.; Prieve, D. C. J.
Phys. Chem. B 2002, 106, 11557–11564.

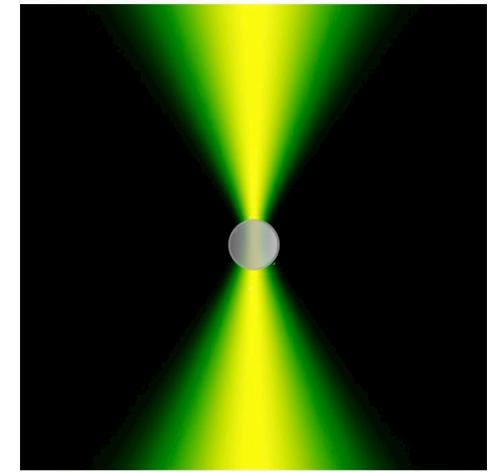
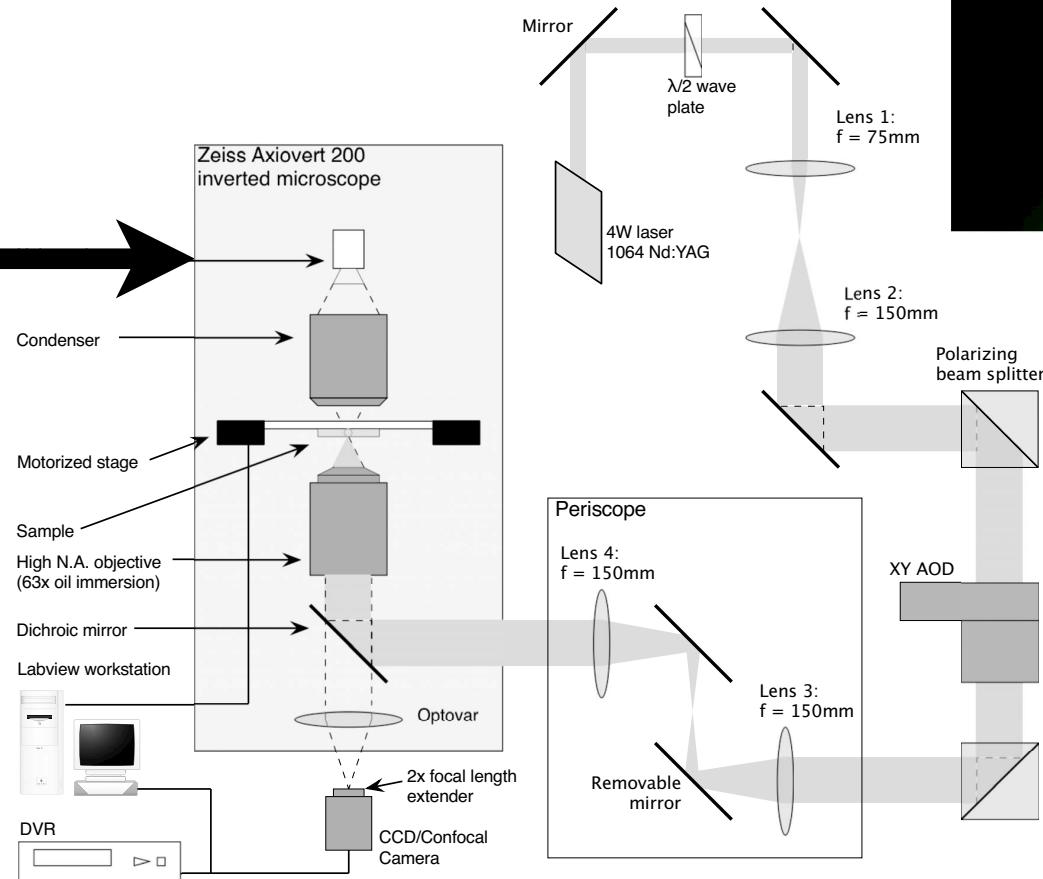


Optical tweezers

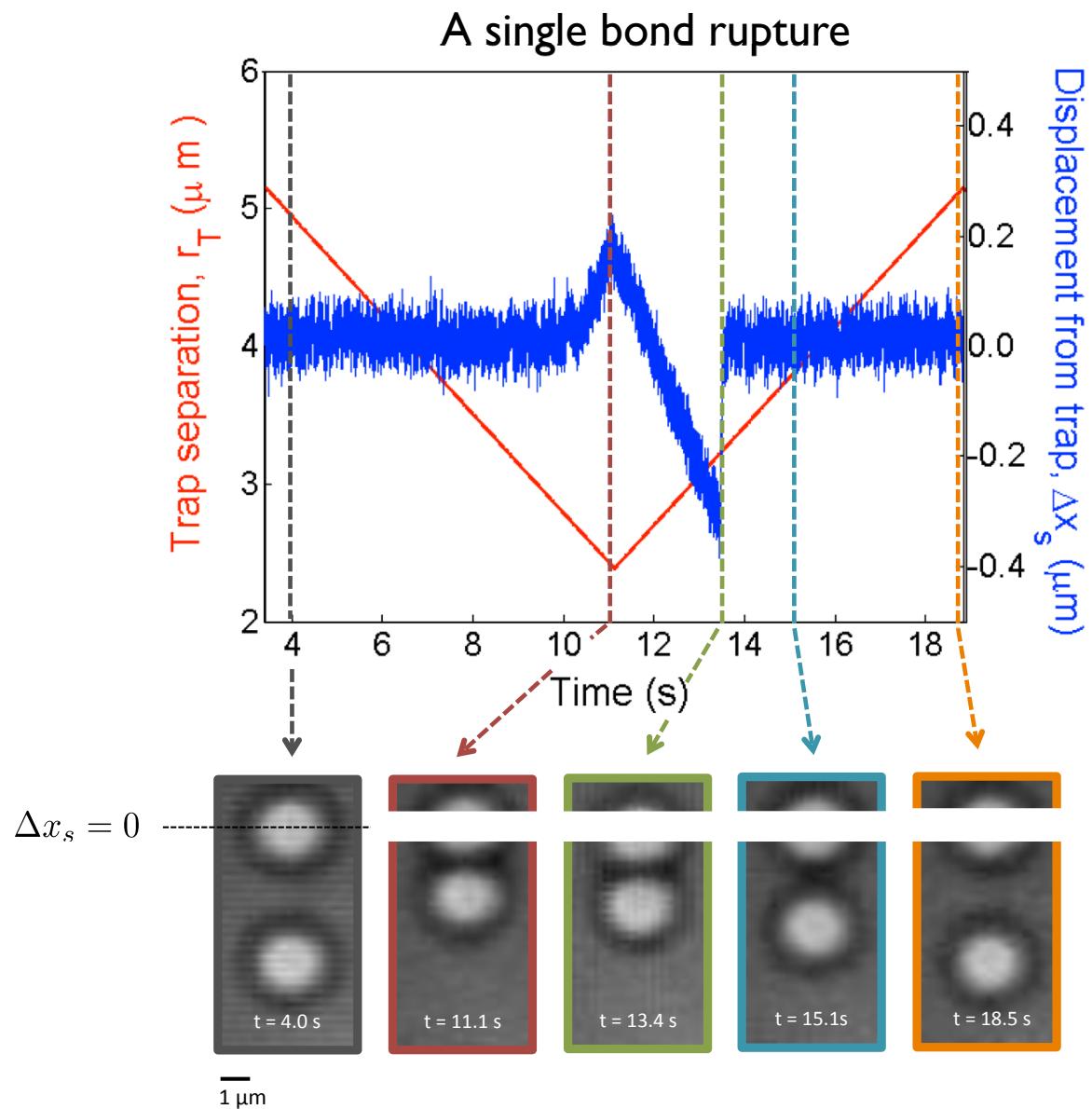
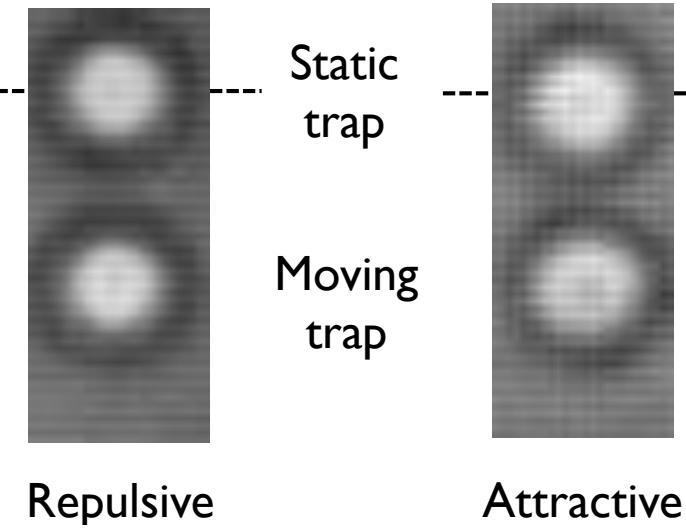


*Back focal plane
detection*

Laser tweezer apparatus

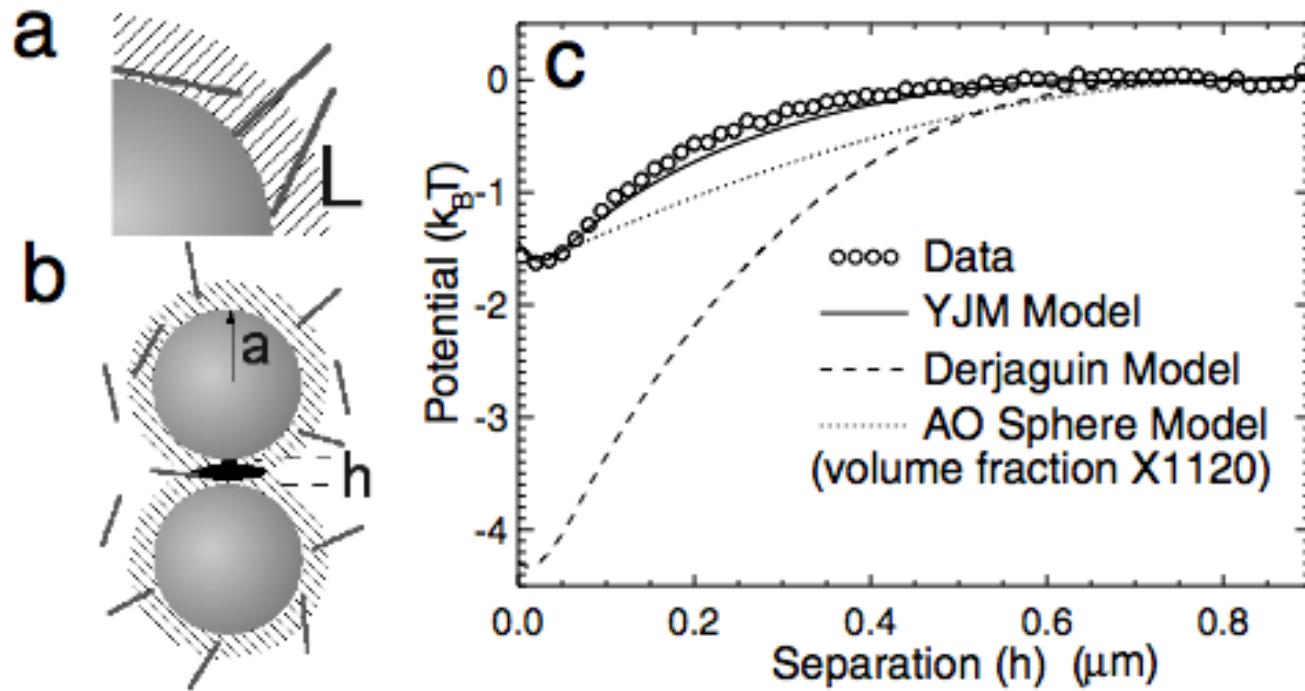


Measuring colloidal “stickiness”



“Line tweezer” interaction measurements

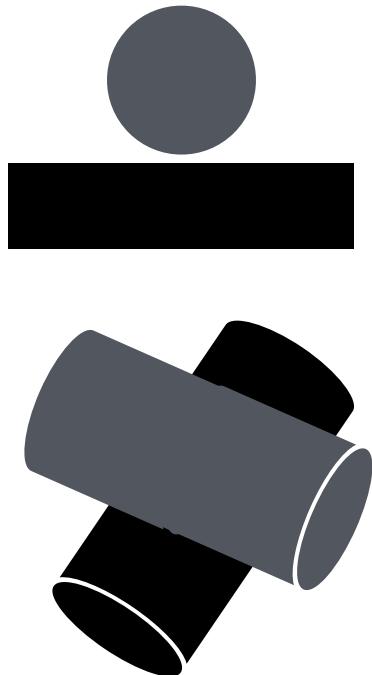
Depletion attraction between flexible rods



Lin, K.; Crocker, J. C.; Zeri, A. C.; Yodh, A. G. *Phys. Rev. Lett.* 2001, 87, 88301.

Direct force measurements and interpretation

Measurements use curved interfaces
(sphere-flat, two spheres, crossed cylinders)



$$W(D)_{\text{curved}} = - \int_{\infty}^D F(D)_{\text{curved}} dD$$

$$W(D)_{\text{flat}} = \frac{F(D)_{\text{curved}}}{2\pi R} \quad \text{Derjugin approximation}$$

$$P(D)_{\text{flat}} = \frac{dW_{\text{flat}}}{dD} = \frac{dF(D)_{\text{curved}}/dD}{2\pi R}$$

Direct force measurements and interpretation

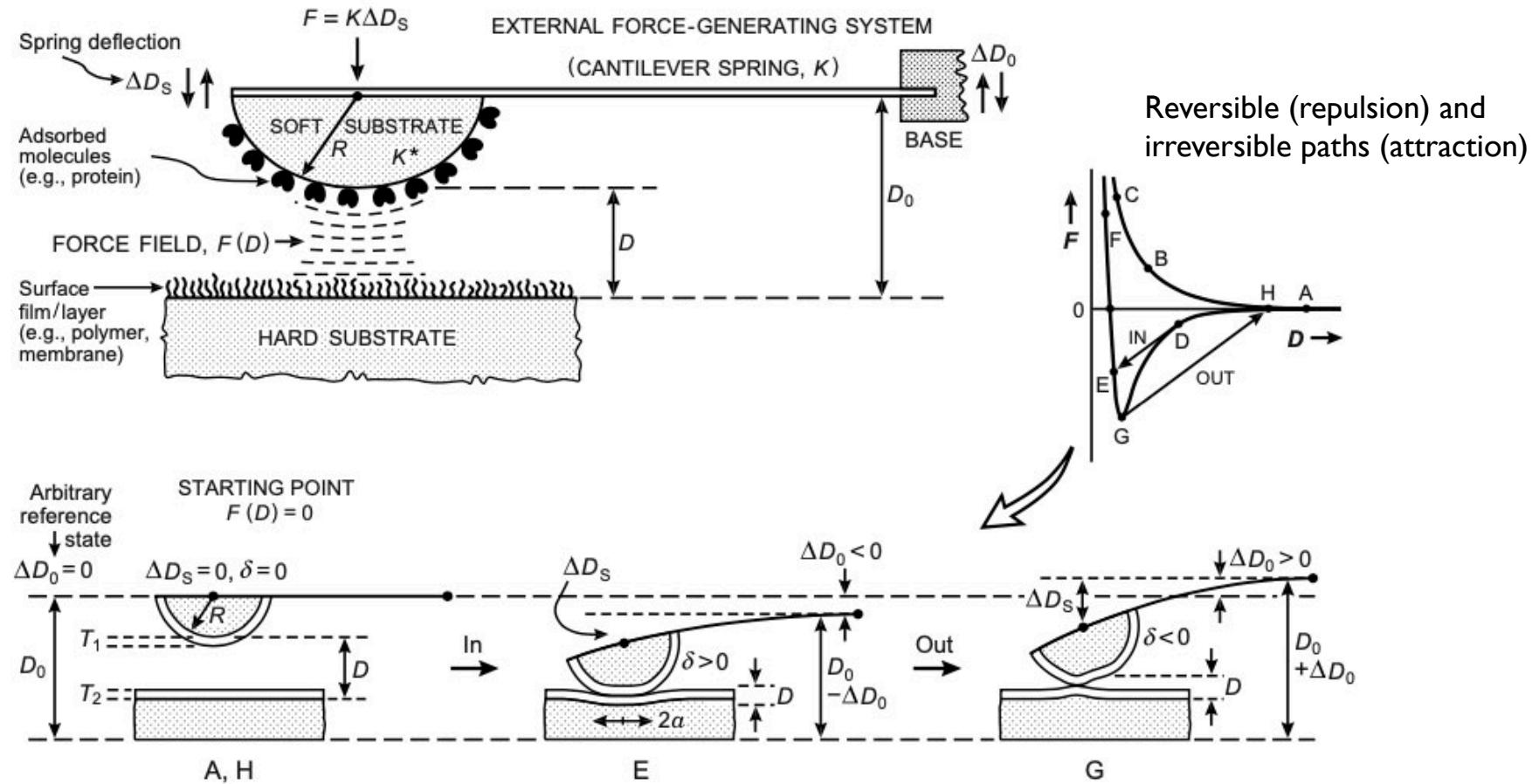
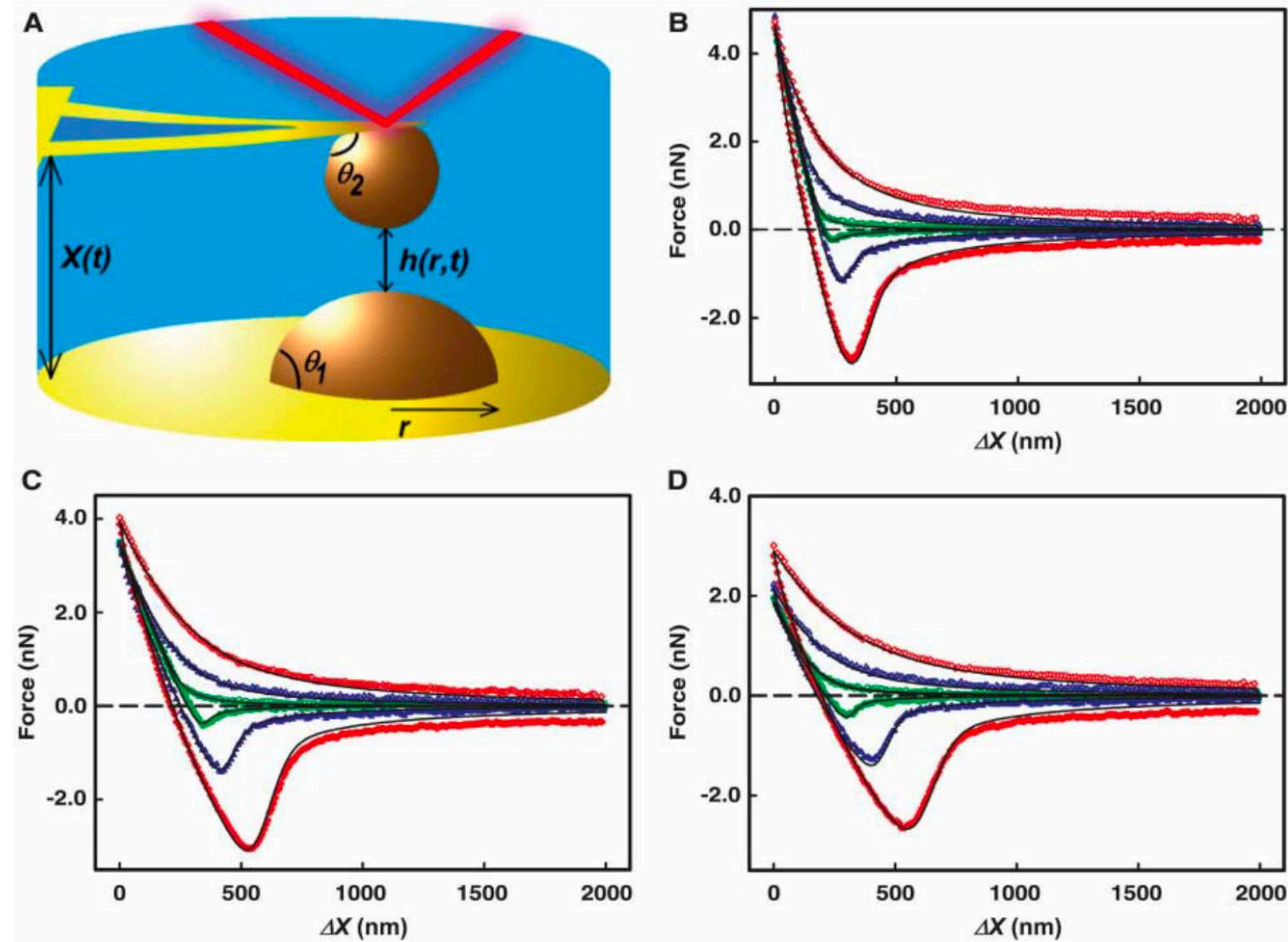


FIGURE 12.4 Generic geometry for force-measuring techniques that employ springs, showing the relevant parameters that must be controlled and/or measured, preferably as a function of time t , during a force measurement: the force F , the undeformed surface radius R , the surface separation D , the spring deflection ΔD_s , the force-measuring (cantilever) spring constant K such that $F(D) = K\Delta D_s$, the effective stiffness of the material K^* and its elastic deformation δ (see JKR theory in Chapter 17 and Figure 12.5), the flattened area $A = \pi a^2$ of the surfaces when in contact, and the film thicknesses T_1 and T_2 . One can define a laboratory reference distance D_0 such that when the base is moved the change in the surface separation ΔD is given by Eq. (12.6): $\Delta D = \Delta D_0 \pm \Delta D_s \pm \delta$.

Two oil droplets interacting

R. R. Dagastine et al., Dynamic Forces Between Two Deformable Oil Droplets in Water. *Science*. 313, 210–213 (2006).

Fig. 1. (A) A schematic of the experiment between two oil droplets, one immobilized on the cantilever and the other immobilized on the substrate of an atomic force microscope. (B to D) The dynamic interaction force F versus piezo drive motion ΔX between two decane droplets in aqueous solution in the presence of SDS at a series of approach (open symbols) and retract (filled symbols) velocities (green circles, $2 \mu\text{m/s}$; blue triangles, $9.3 \mu\text{m/s}$; red diamonds, $28 \mu\text{m/s}$) over a range of SDS concentrations: (B) 0.1 mM , (C) 3 mM , and (D) 10 mM . The points refer to the experimental data, and the solid lines are the calculated force curves from a comprehensive model of the dynamic droplet interactions.



JKR apparatus

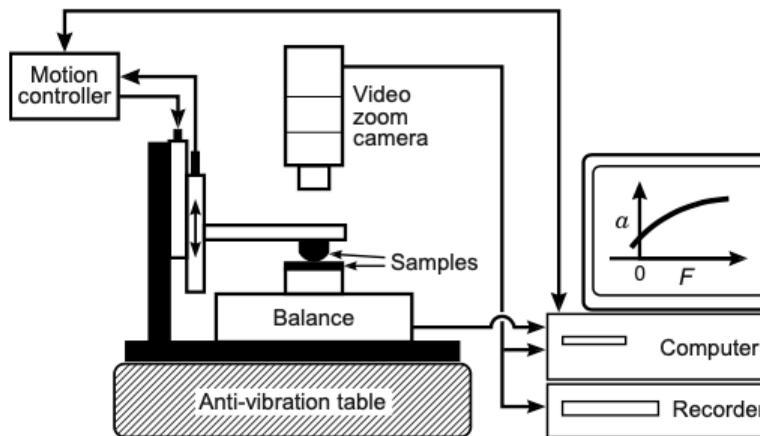
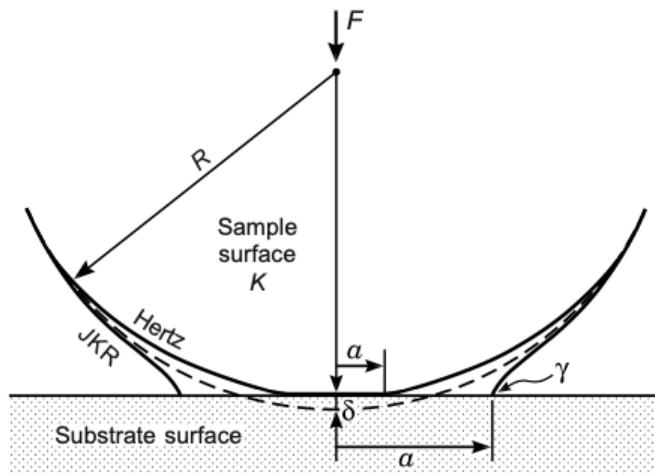


FIGURE 12.5 A JKR apparatus named after the Johnston-Kendall-Roberts theory of adhesion mechanics (Johnson et al., 1971), developed by Tirrell and coworkers (Mangipudi et al., 1995; Tirrell, 1996) to study the adhesion forces and deformations of macroscopic solids. The device allows for the measurement of the molecular contact radius a as a function of the applied load F , including negative loads. The results can then be used to test theories of adhesion and contact mechanics such as the JKR and Hertz theories (Chapter 17). This technique also enables the measurement of surface energies γ via Eq. (11.17) or Eq. (12.10).

Adhesion between elastic particles

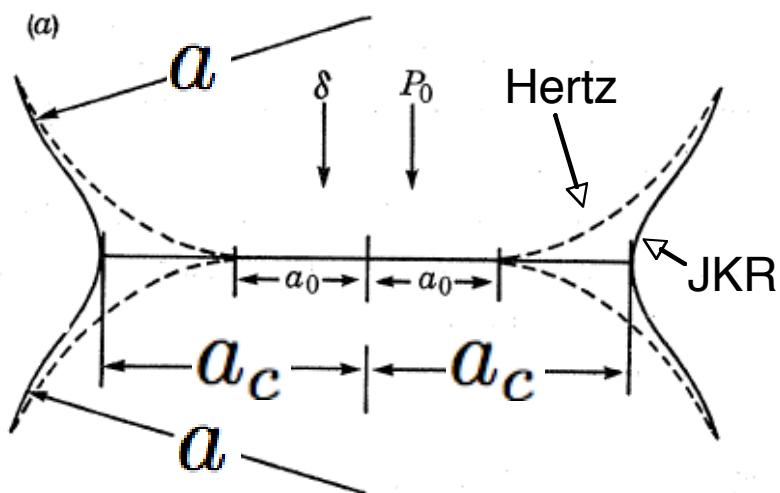
JKR Theory

Johnson, Kendall and Roberts, Proc. R. Soc. London, Ser.A 324, 301 (1971).

$$P_0 = 0 \quad a_c = \left[\frac{3\pi a^2 W_{SL}}{2K} \right]^{1/3}$$

Surface adhesion energy:

$$W_{SL} = \gamma_S + \gamma_L - \gamma_{SL}$$



$$W_{SL} = \gamma_L(1 + \cos \theta_0) \approx 93 \text{ mJ/m}^2$$

where¹ $\theta_0 = 73.7 \pm 0.3^\circ$

$$\gamma_L \approx 72 - 73 \text{ mN/m}$$

Elastic modulus: $K = \frac{2E}{3(1 - \nu^2)}$
 $K \approx 2.5 \text{ GPa}$

- 1 Kwok et al. J. Colloid Interface Sci. 206, 44 (1998).
2 Jarvis and Scheiman, J. Phys. Chem. 72, 74 (1968).

Instabilities

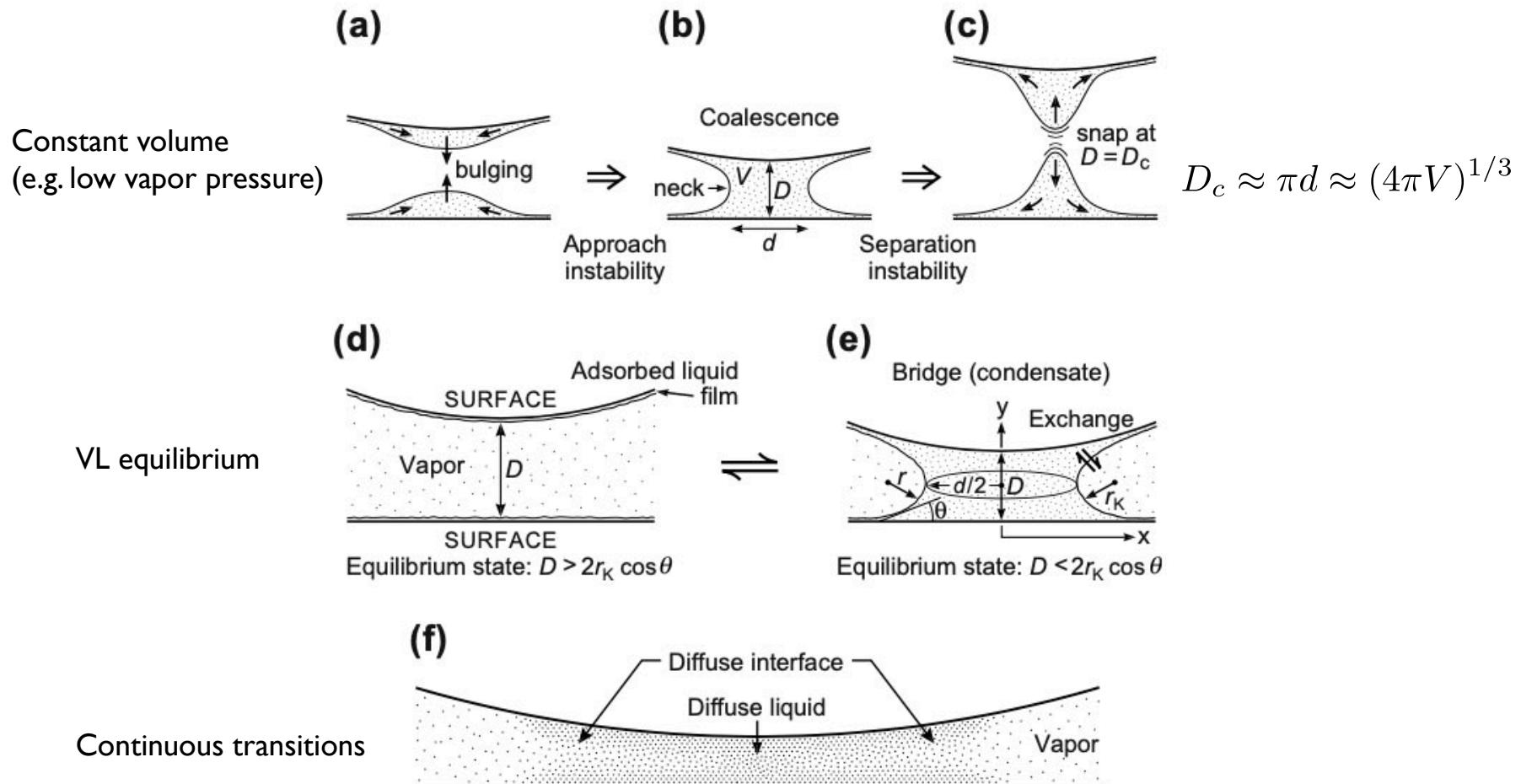
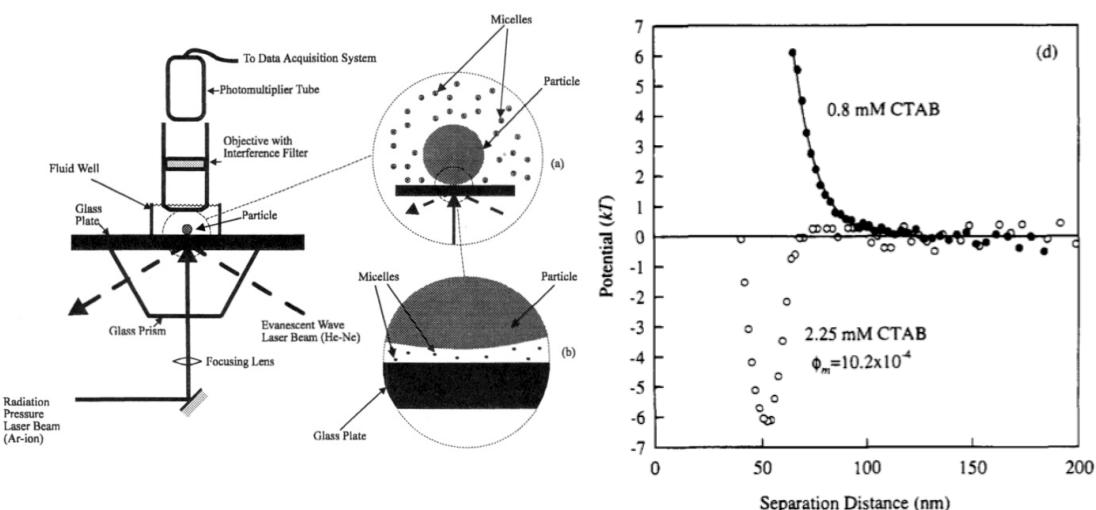


FIGURE 12.6 The different geometries of liquid bridges can give rise to instabilities in the “capillary forces” measured on approach or separation of two surfaces. Shown here are the rapid *mechanical* instabilities and slower *thermodynamic transitions* that can occur on approach ($a \rightarrow b$) and separation ($b \rightarrow c$) of two surfaces with a liquid film on each. At (c) a mechanical instability (rapid snapping) occurs at a critical distance $D_c \propto V^{1/3}$, where V is the (constant) volume of the liquid. At thermodynamic equilibrium (d and e), V is no longer constant; instead, the mean curvature of the liquid-vapor interface is now constant and equal the Kelvin radius r_K (Ch. 17). Some transitions in confined geometries and thin films can be slow and/or continuous rather than abrupt (Stroud et al., 2001).

Interaction measurements

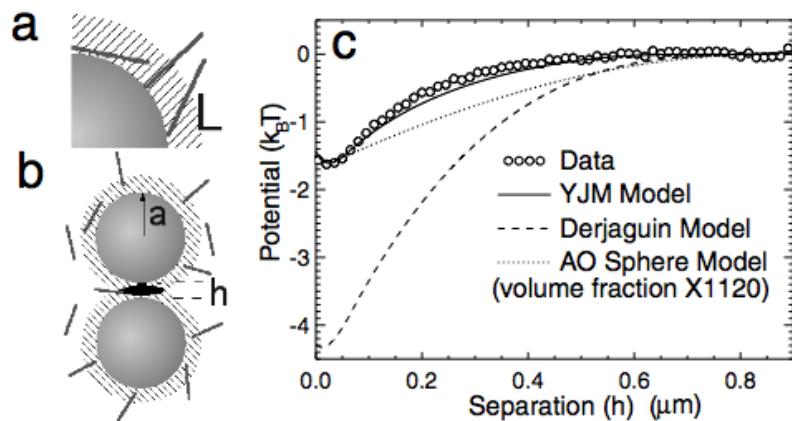
TIRM, surfactant micelles

Sober, D. L.; Walz, J. Y. *Langmuir* 1995, 11, 2352–2356.



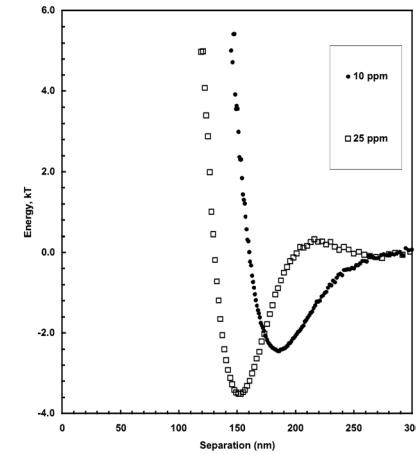
Line tweezers, flexible rods

Lin, K.; Crocker, J. C.; Zeri, A. C.; Yodh, A. G. *Phys. Rev. Lett.* 2001, 87, 88301.



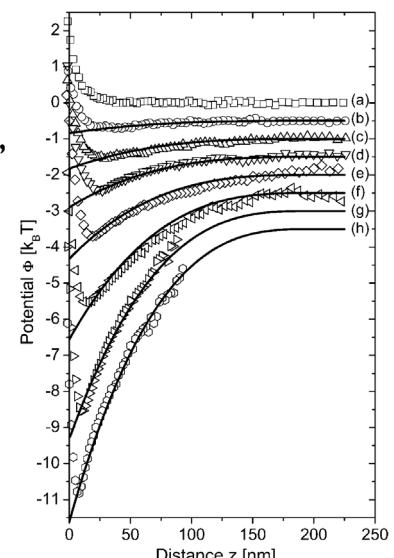
TIRM, polyelectrolytes

Biggs, S.; Dagastine, R. R.; Prieve, D. C. *J. Phys. Chem. B* 2002, 106, 11557–11564.

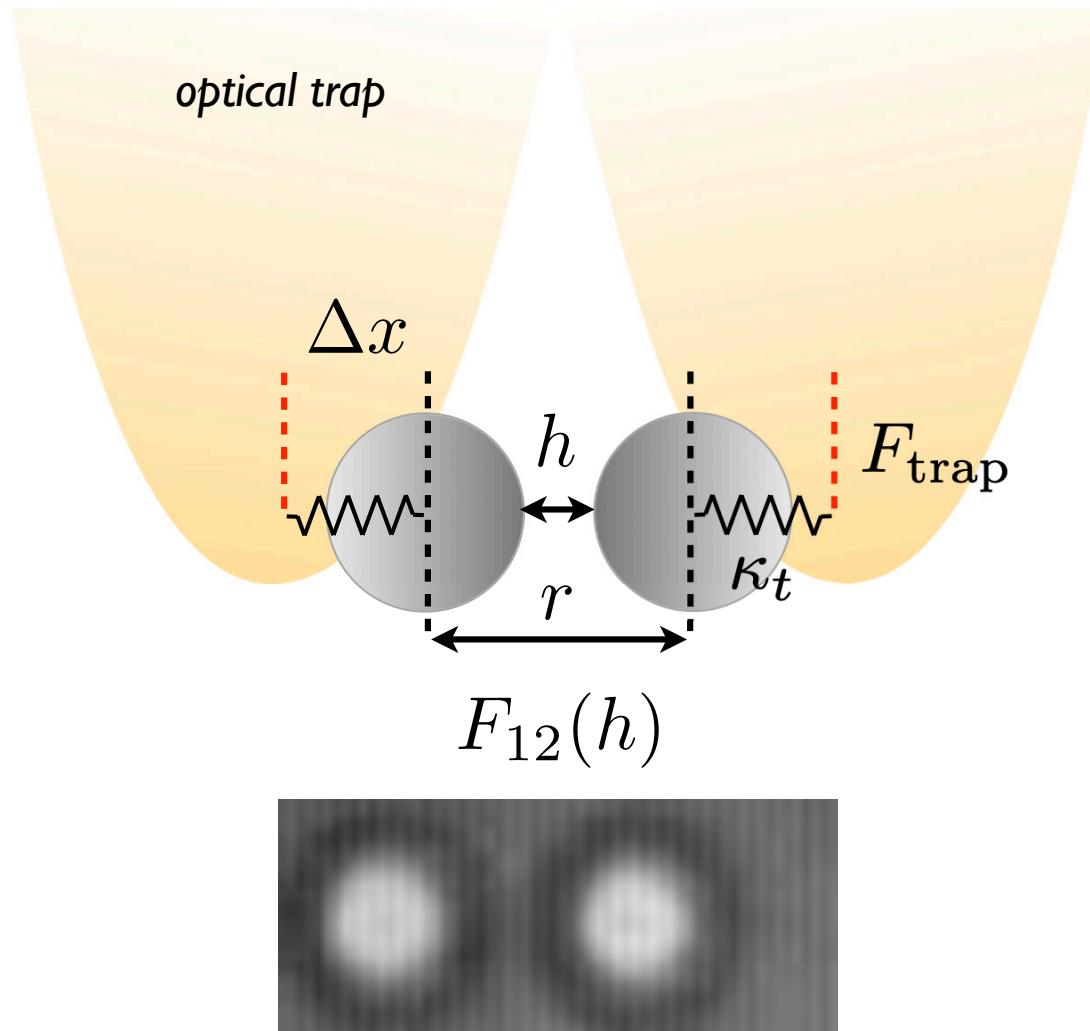


TIRM, rigid rods

Helden, L.; Roth, R.; Koenderink, G. H.; Leiderer, P.; Bechinger, C. *Phys. Rev. Lett.* 2003, 90, 48301.



Optical micromanipulation



Radiation forces

Photon momentum $\mathbf{p} = \hbar\mathbf{k}$

$$|\mathbf{k}| = 2\pi/\lambda$$

Radiation pressure acting on macroscopic surface

$$P_{rad} \sim 2qI_0/c$$

$$I_0 = 1000 \text{W/m}^2 \quad \approx 10^{-8} \text{N/m}^2$$

$$q = 0.1$$

Radiation force exerted on micrometer diameter particle

Particle diameter $2a = 1\mu\text{m}$

$$F_{rad} \approx 10^{-11} \text{N}$$

Laser power 10mW

$$kT/a \approx 10^{-14} \text{N}$$

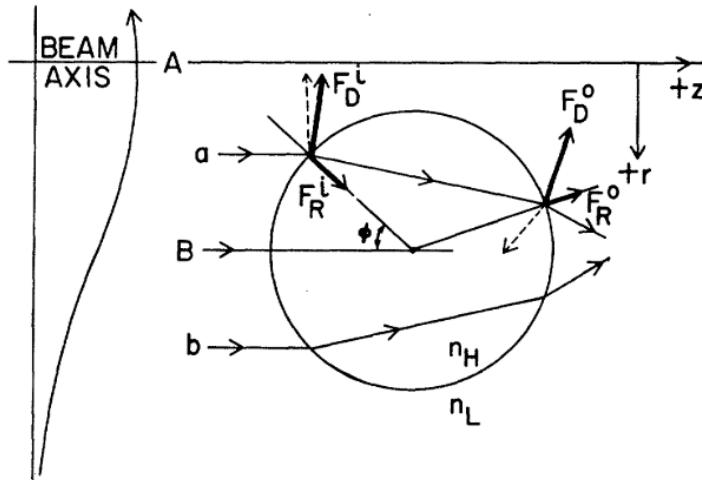
$$F_{rad} \sim 10^3 kT/a$$

Radiation forces in a Gaussian beam

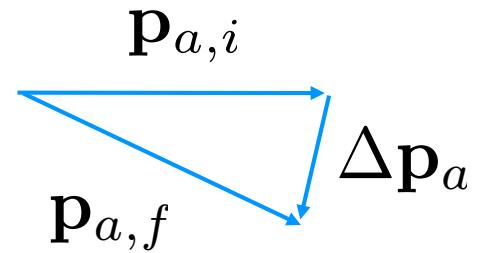
A. Ashkin. Phys. Rev. Lett., 24:156–159, 1970.

A. Ashkin. Proc. Natl. Acad. Sci. USA, 94:4853–4860, 1997.

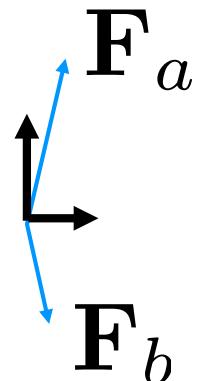
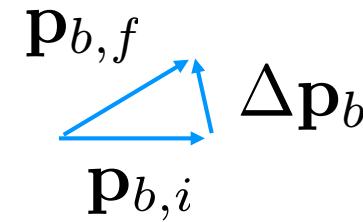
Collimated TEM₀₀



Ray *a*

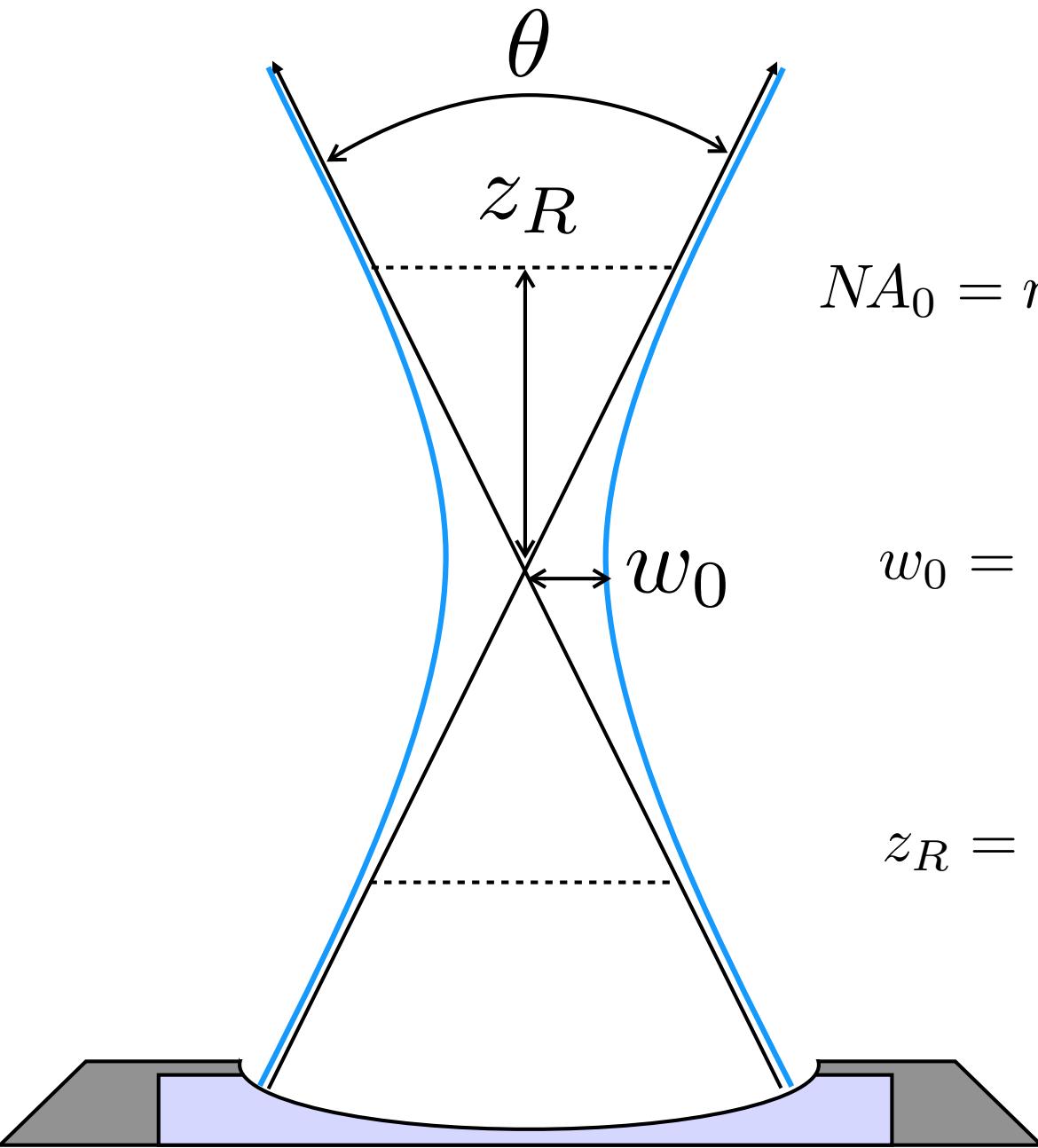


Ray *b*



*Resultant force
imparted on particle*

Focused Gaussian beam



$$NA_0 = n \sin \theta$$

Numerical aperture

$$w_0 = \frac{\lambda}{2NA_0}$$

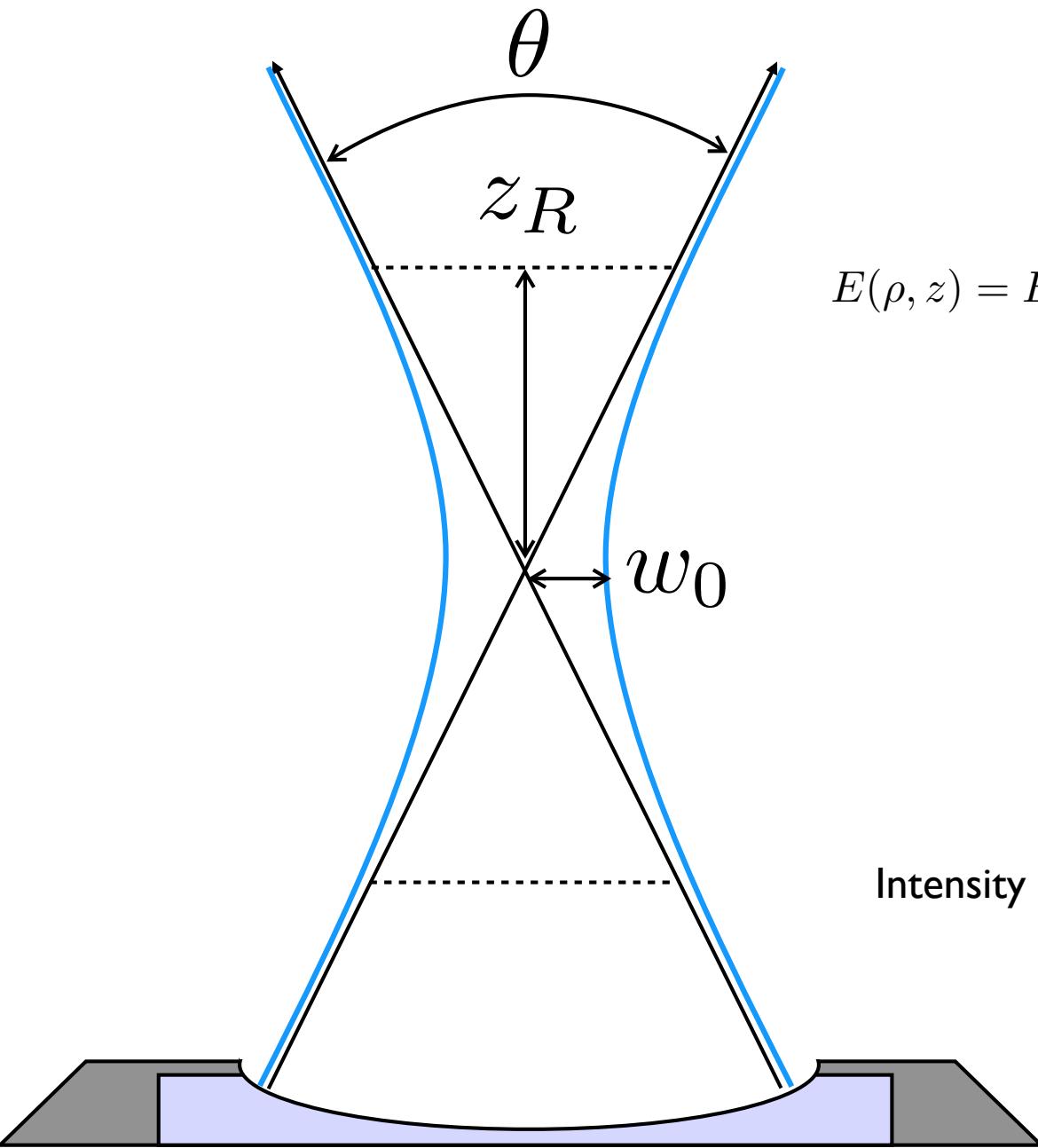
Diffraction limited
beam width

$$z_R = \frac{\pi n w_0^2}{\lambda}$$

Rayleigh length

J. Mertz. *Introduction to Optical Microscopy*.
Roberts, Greenwood Village, Colorado, 2010.

Radiant field



Radiant field

$$E(\rho, z) = E_0 \frac{w_0}{w(z)} e^{i2\pi\kappa z} e^{-\rho^2/w(z)^2} e^{i\pi\kappa\rho^2/R(z)} e^{-i\eta(z)}$$

$$w(z) = w_0 \sqrt{1 + z^2/z_R^2}$$

$$R(z) = z (1 + z_R^2/z^2)$$

$$\eta(z) = \tan^{-1}(z/z_R)$$

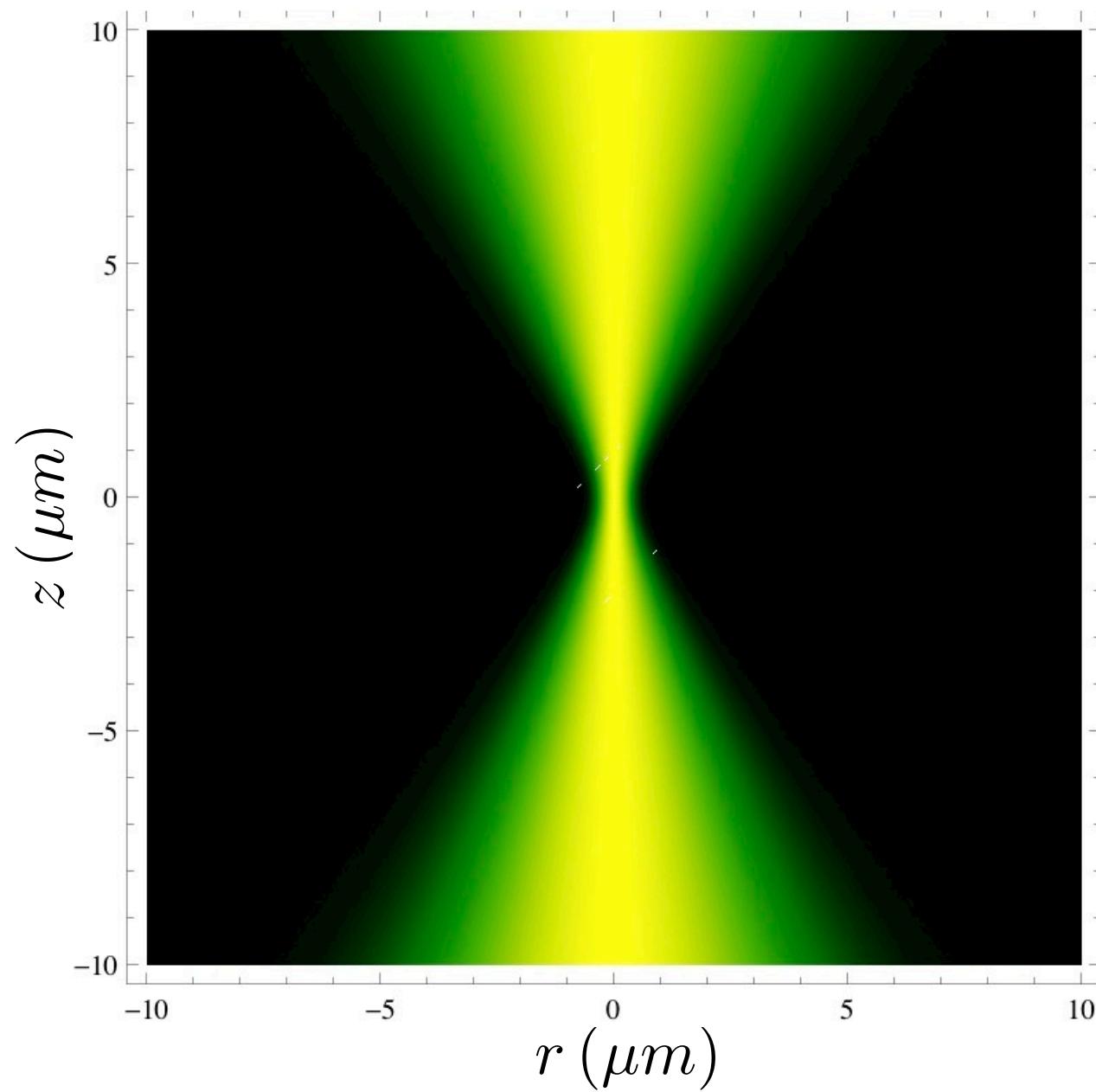
$$\kappa = n/\lambda$$

Intensity (irradiance) $I(\rho, z) = \langle E(\rho, z) E^*(\rho, z) \rangle$

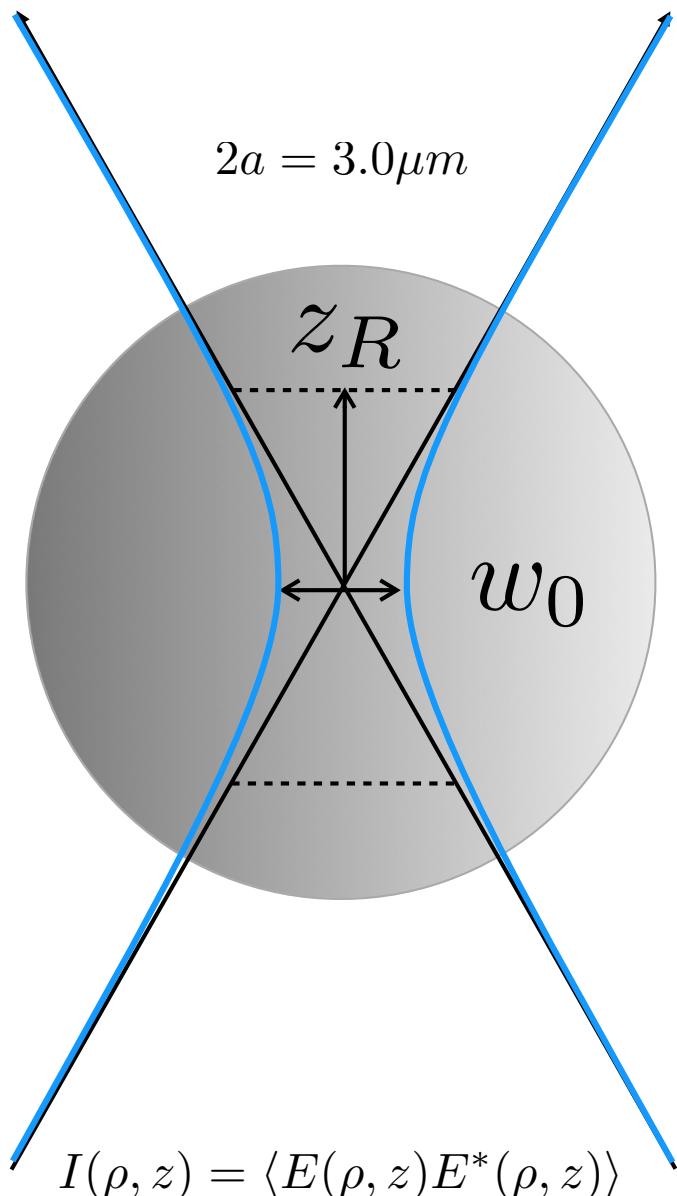
$$[=] W/m^2$$

J. Mertz. *Introduction to Optical Microscopy*.
Roberts, Greenwood Village, Colorado, 2010.

Density plot of beam intensity



Typical values



$$2a = 3.0\mu m$$

$$\lambda_{vacuum} = 1064\text{nm}$$

$$NA_0 = 1.1$$

$$\theta = 58^\circ$$

$$w_0 = 0.48\mu m$$

$$z_R = 0.92\mu m$$

$$a > z_R$$

Beam diverges with full angle

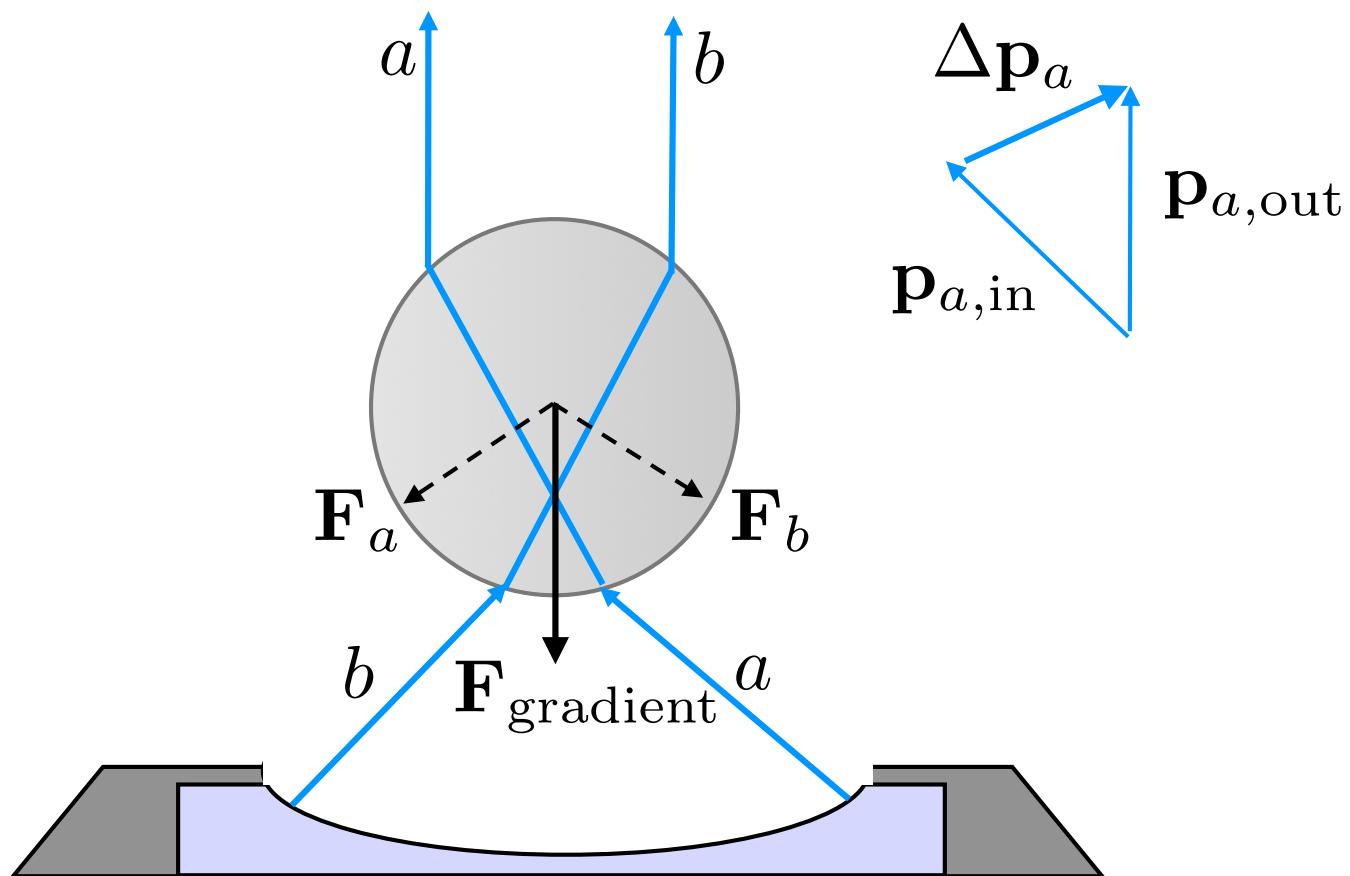
Analyze in ray-optic regime

Gradient force trap

A. Ashkin, et al. Opt. Lett., 11:288–290, 1986.

Ray optic regime

$$d \gg \lambda$$

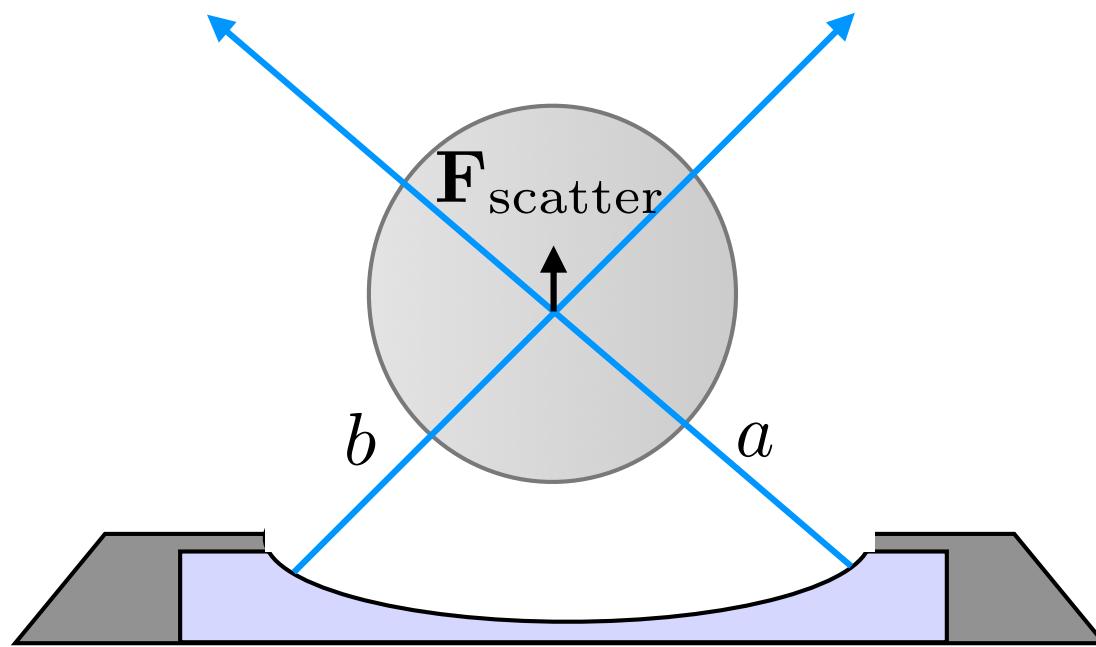


High NA objective

Gradient force trap

Ray optic regime

$$d \gg \lambda$$

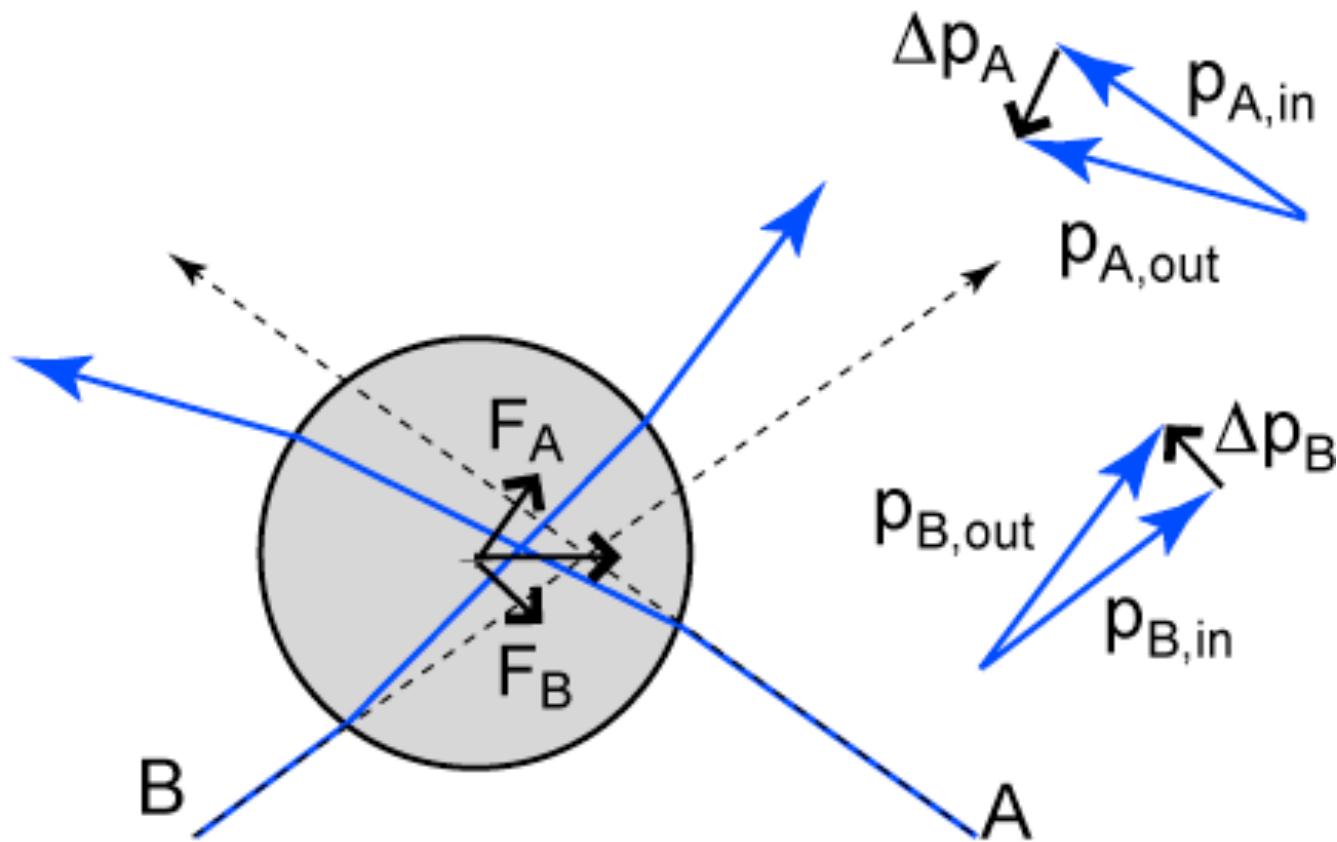


High NA objective

Lateral displacement

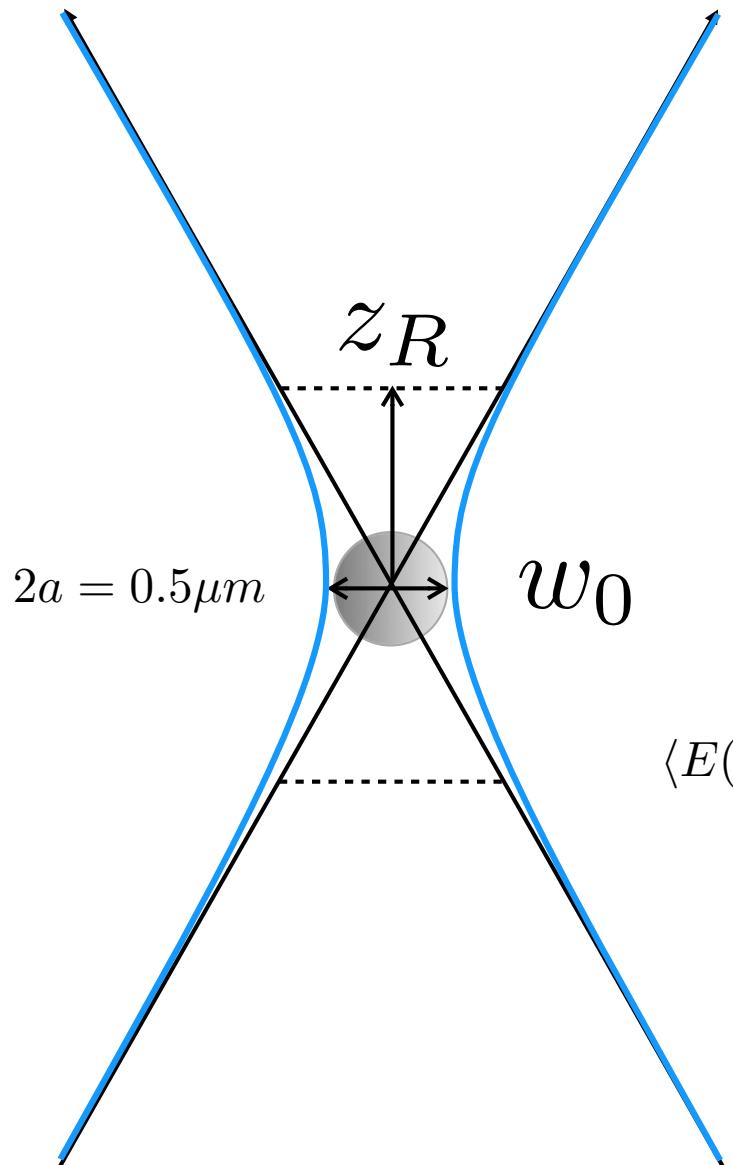
Ray optic regime

$$d \gg \lambda$$



Rayleigh regime

$$d \ll \lambda$$



Induced dipole $\mathbf{p} = \alpha \mathbf{E}$

Energy and force $U = -\mathbf{p} \cdot \mathbf{E}$ $\mathbf{F} = \nabla(U)$

$$\mathbf{F}_{gradient} = 2\pi\epsilon\epsilon_0 \Re\{\alpha(\omega)\} a^3 \nabla \langle E(\rho, z) E^*(\rho, z) \rangle$$

$$\mathbf{F}_{gradient} = \frac{4\pi\epsilon_0 a^3}{c} \left[\frac{n_p^2 - n_s^2}{n_s(n_p^2 + 2n_s^2)} \right] \nabla I$$

$$\langle E(\rho, z) E^*(\rho, z) \rangle = \frac{2I(\rho, z)}{cn_s\epsilon_0} = \left(\frac{2}{cn_s\epsilon_0} \right) \left(\frac{I_0 e^{-2\rho^2/w(z)^2}}{1 + z^2/z_R^2} \right)$$

$$w(z) = w_0 \sqrt{1 + z^2/z_R^2}$$

Important parameters for optical trapping

The ability to optically trap a particle depends on:

The particle size and shape

The refractive index contrast between the particle and medium

The amount of light absorbed by the particle and sample

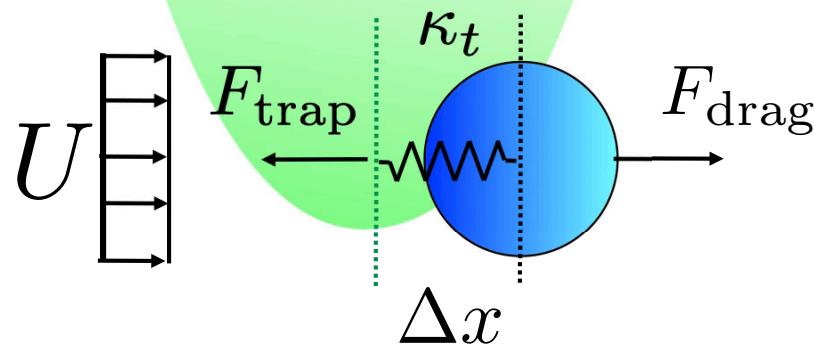
The laser wavelength

The laser power

The quality of the beam and focus

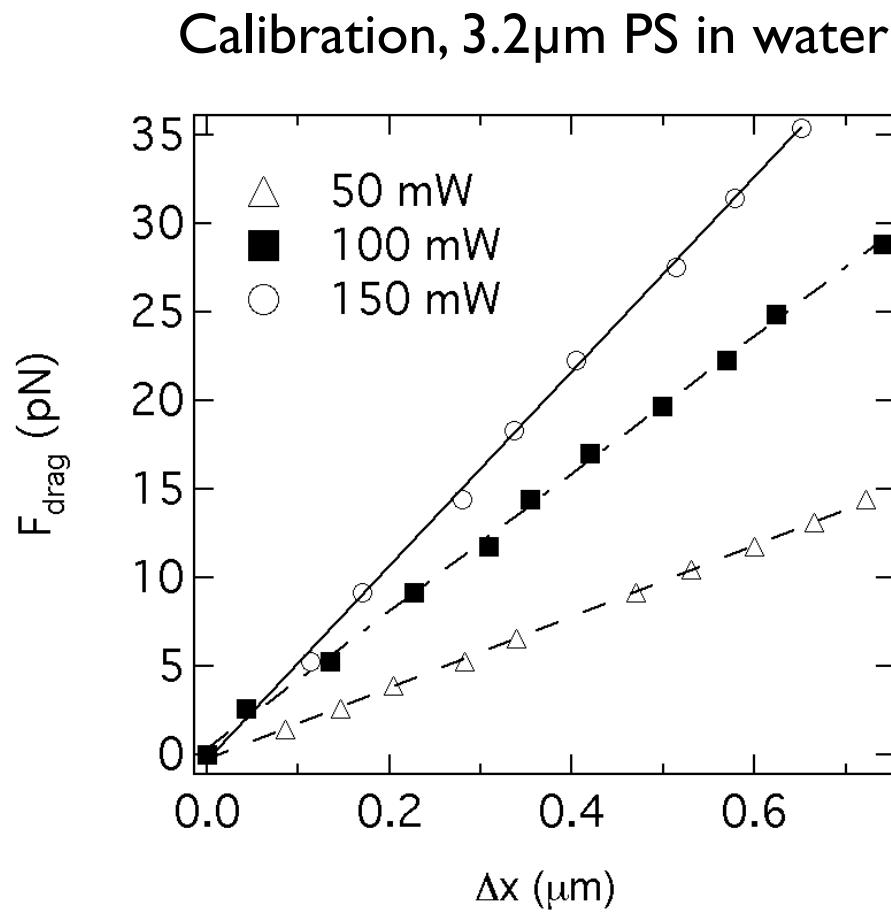
The laser mode (Gaussian, Laguerre-Gaussian, etc.)

“Steady drag” measurements



$$F_{\text{drag}} = 6\pi a \eta U = \kappa_t \Delta x$$

$$F_{\text{drag}} > F_{t,\max}$$



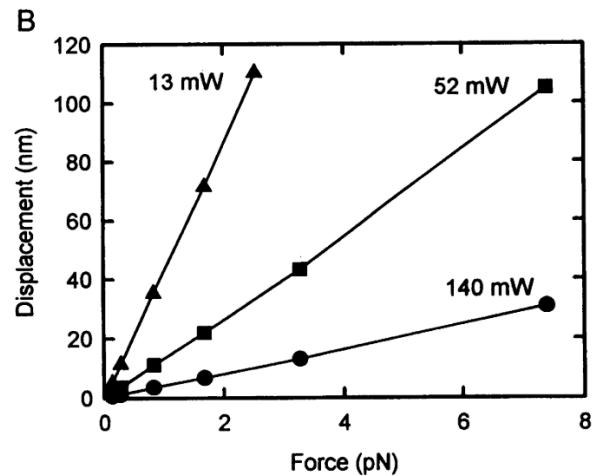
$$\kappa_t = 10 - 60 \text{ pN}/\mu\text{m}$$

Optical tweezer calibration

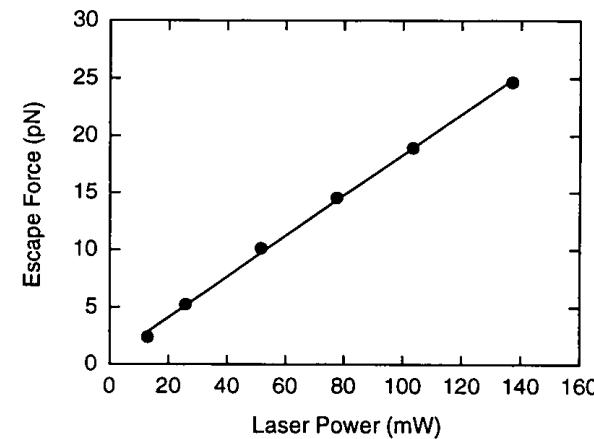
1 μm polystyrene, Nd:YAG ($\lambda=1064\text{nm}$)

R. M. Simmons, J. T. Finer, S. Chu, and J. A. Spudich. Biophys. J., 70:1813–1822, 1996.

Trap stiffness



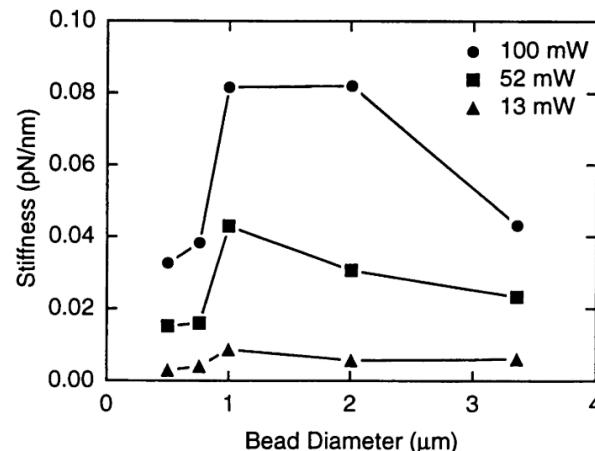
Escape force



Particle size dependence

$$\kappa_T \sim a^3 \quad \text{Rayleigh regime}$$

Plateaus in ray-optic



Trap calibration—power spectral method

- Use power spectral density (PSD) from photodiode detector
- Fit corner frequency to find trap stiffness, α
- Langevin equation

$$\beta \dot{y} + \alpha y = F(t)$$

- PSD & corner frequency

$$S_{yy} = \frac{kT}{2\pi\beta(f_0^2 + f^2)}$$

$$f_0 = \alpha(2\pi\beta)^{-1}$$

Force calibration, $0.6\mu\text{m}$ silica particle

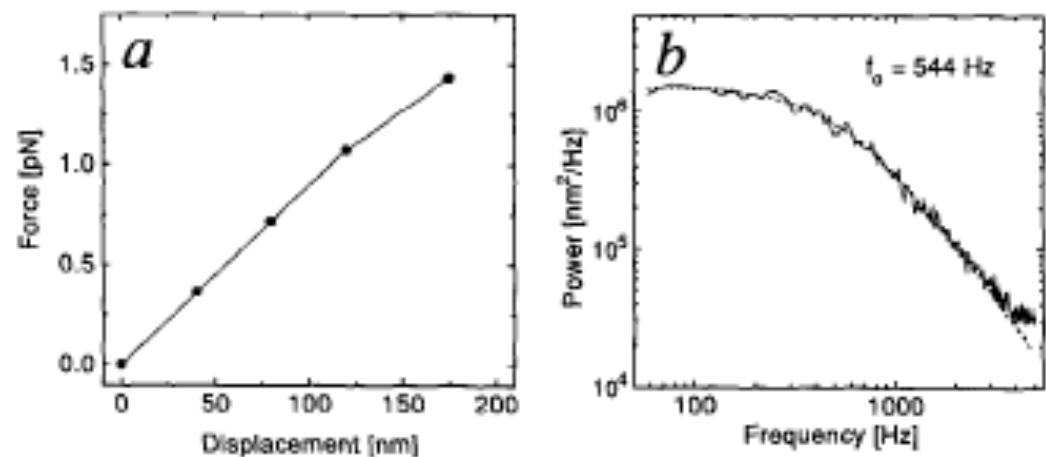
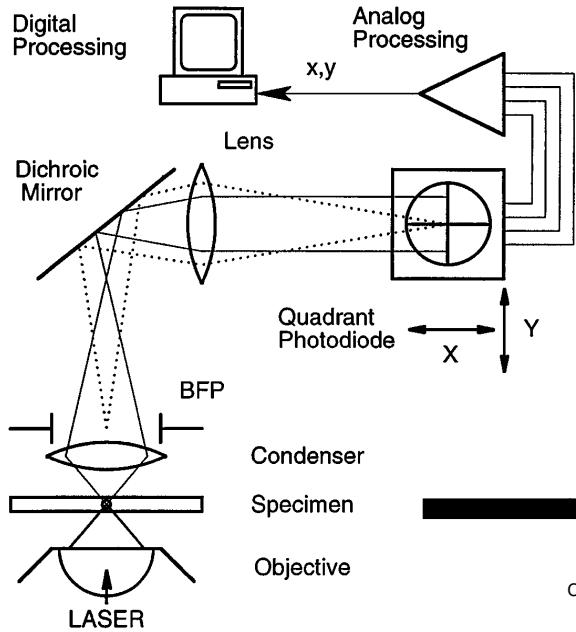


Figure 10 Force calibration of an optical trap. (a) Force versus displacement for a trapped silica bead (diameter $\sim 0.6 \mu\text{m}$; power $\sim 14 \text{ mW}$), calibrated according to Equations 14 and 15 ($x_0 = 2 \mu\text{m}$; $\beta = 5.7 \times 10^{-6} \text{ pN s/nm}$). Stiffness is constant out to 150 nm ($\alpha = 9.0 \times 10^{-3} \text{ pN/nm}$), beyond which it decreases. (b; Solid line) The thermal noise spectrum of a trapped silica bead measured with the optical trapping interferometer (diameter $\sim 0.6 \mu\text{m}$; power $\sim 28 \text{ mW}$). (Dotted line) The fit by a Lorentzian. The corner frequency of the Lorentzian (544 Hz) implies $\alpha = 1.9 \times 10^{-2} \text{ pN/nm}$.

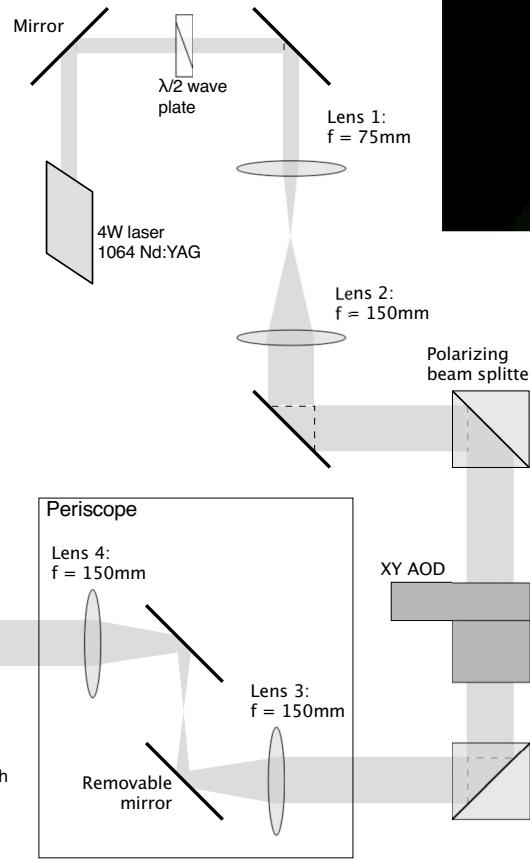
Adapted from K. Svoboda and S.M. Block, 1994.

Optical tweezer experiment



*Back focal plane
detection*

Laser tweezer apparatus



Measurement of particle position: Back focal plane detection

Scattered light collected by condenser is imaged onto a quadrant photodiode

Spatial resolution \sim nm,

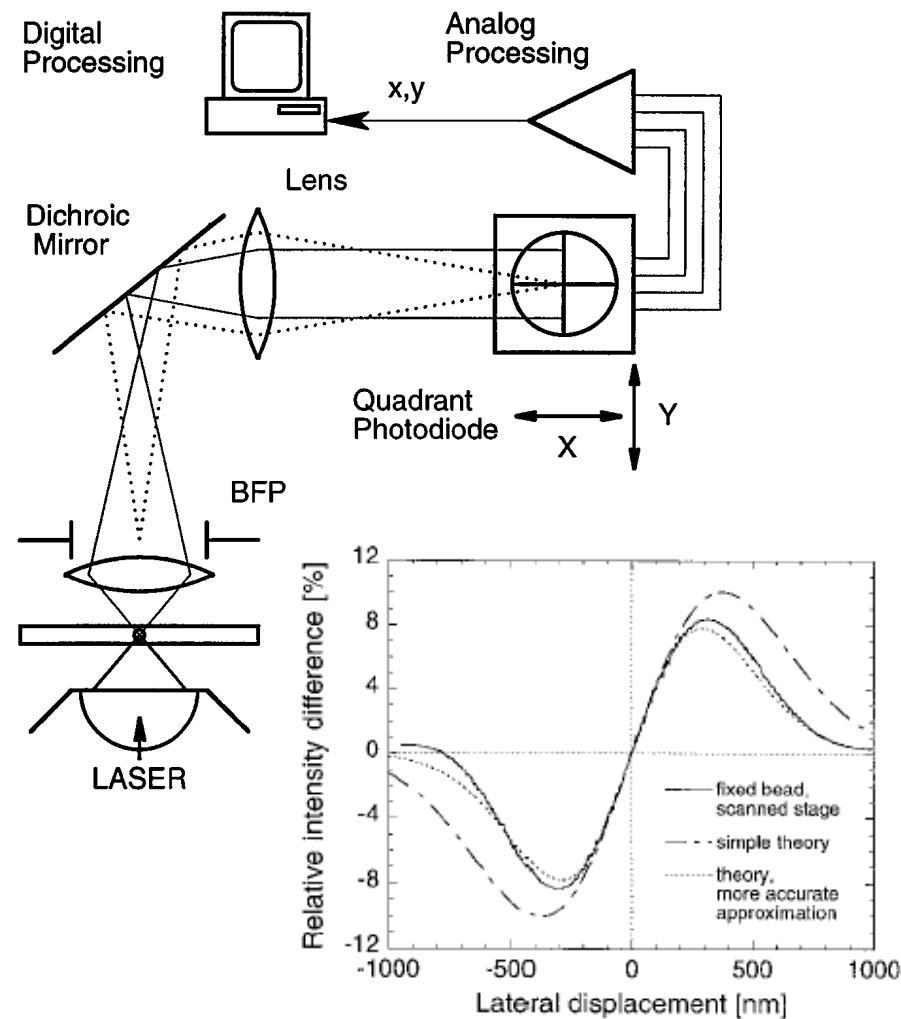
Frequency response \sim 10kHz

Source can be trap laser or co-aligned laser

Fixed vs. moving reference frame

Can be extended to two independent beams by splitting polarization

Bandwidth improved using non-Si detectors

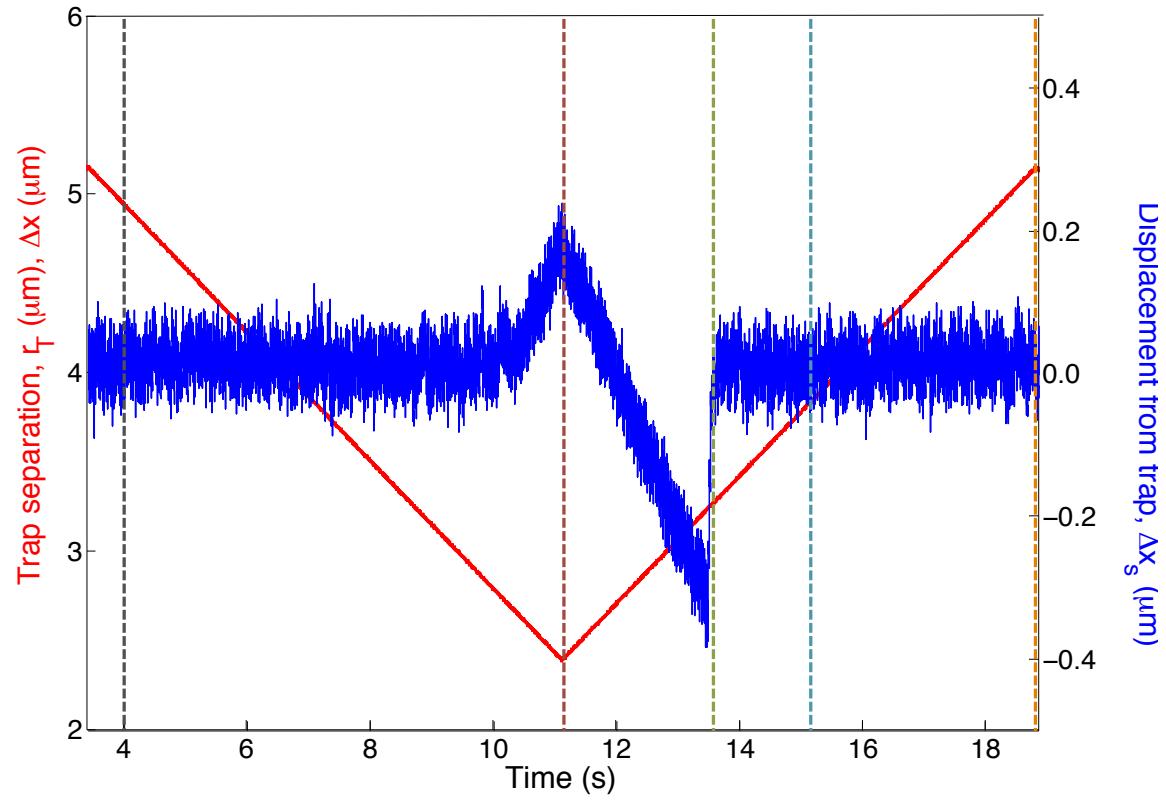
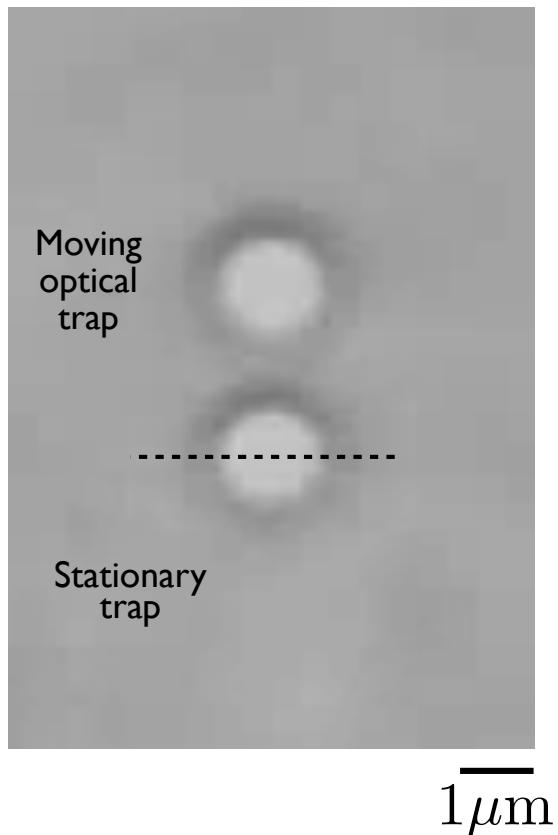


Gittes and Schmidt, Optics Lett. 23:7, 1998.

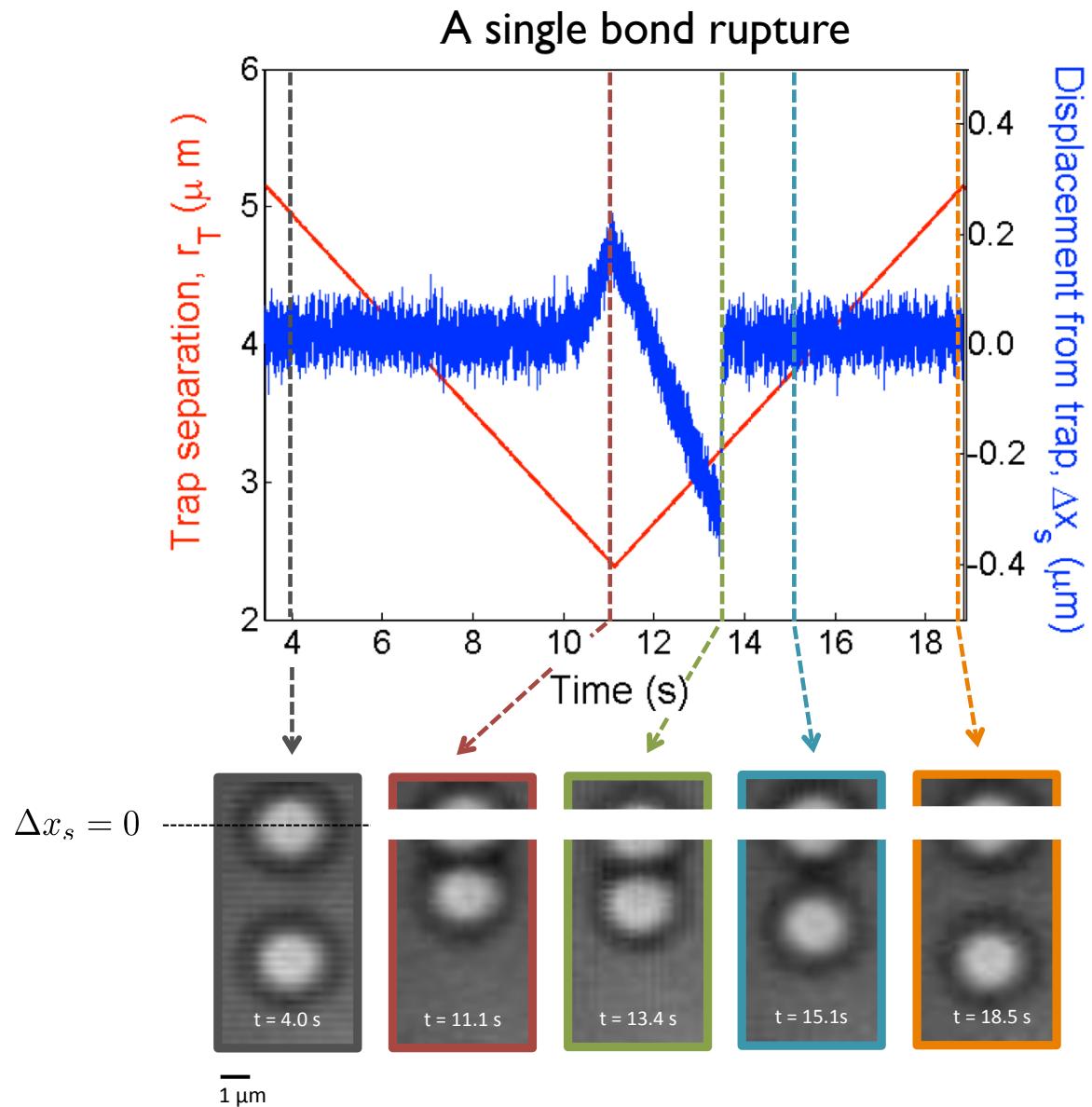
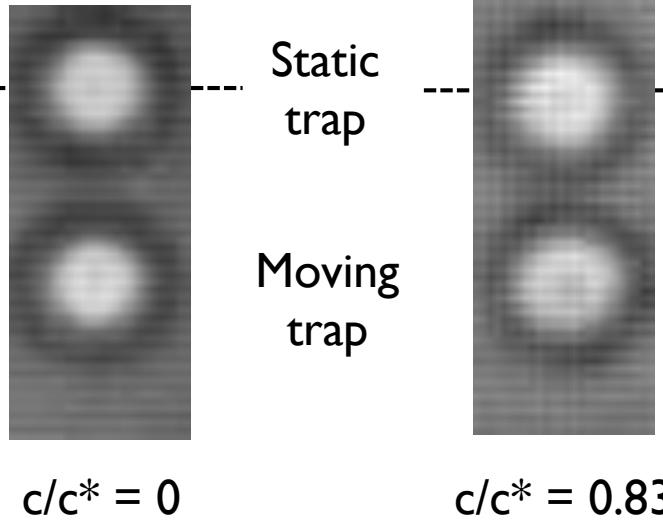
Peterman et al., Rev. Sci. Instrum. 74:3246, 2003.

Rupturing “bonds” between colloids

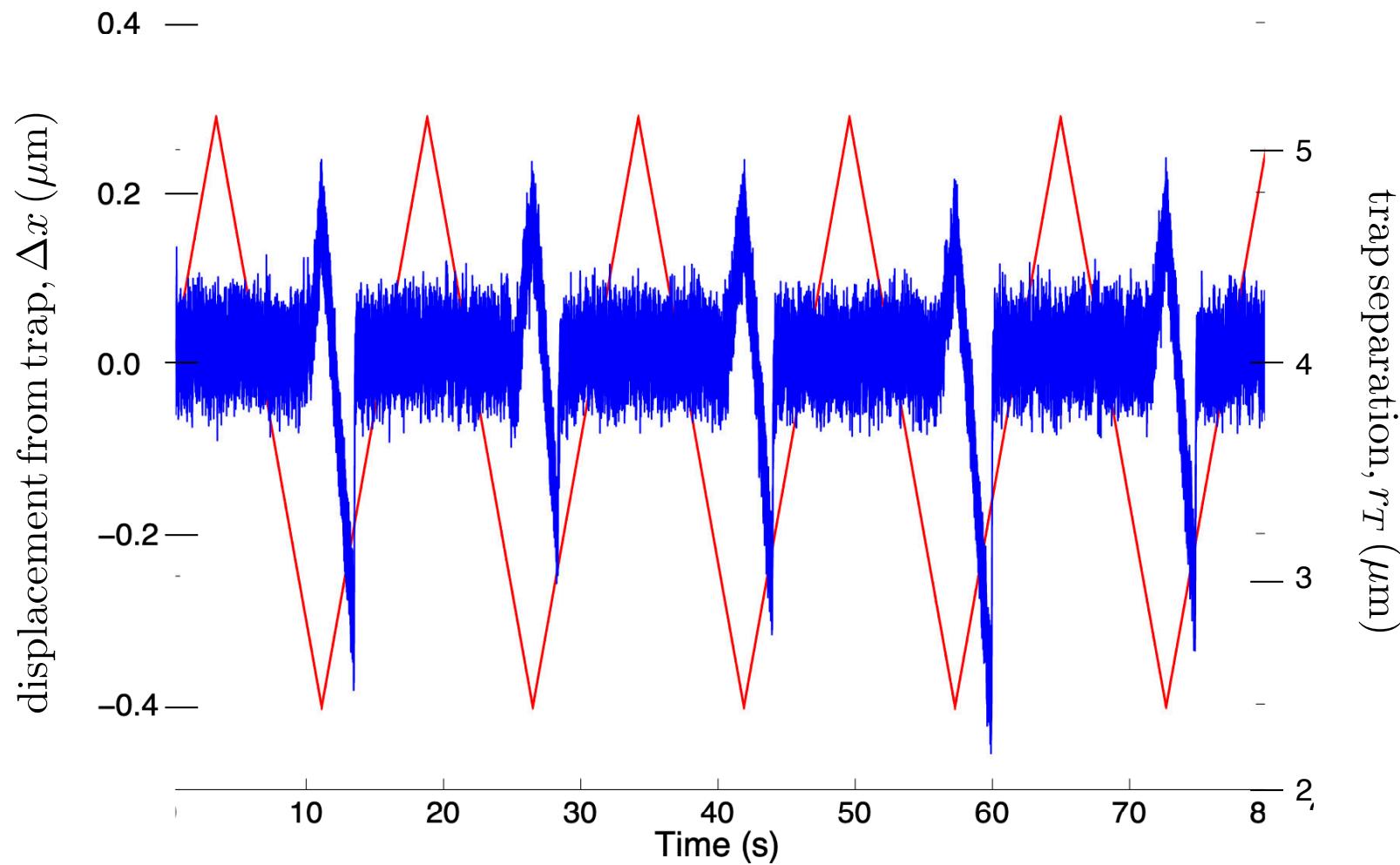
Swan, J.W., Shindel, M. & Furst, E. M. *Phys. Rev. Lett.* 109, 198302 (2012).



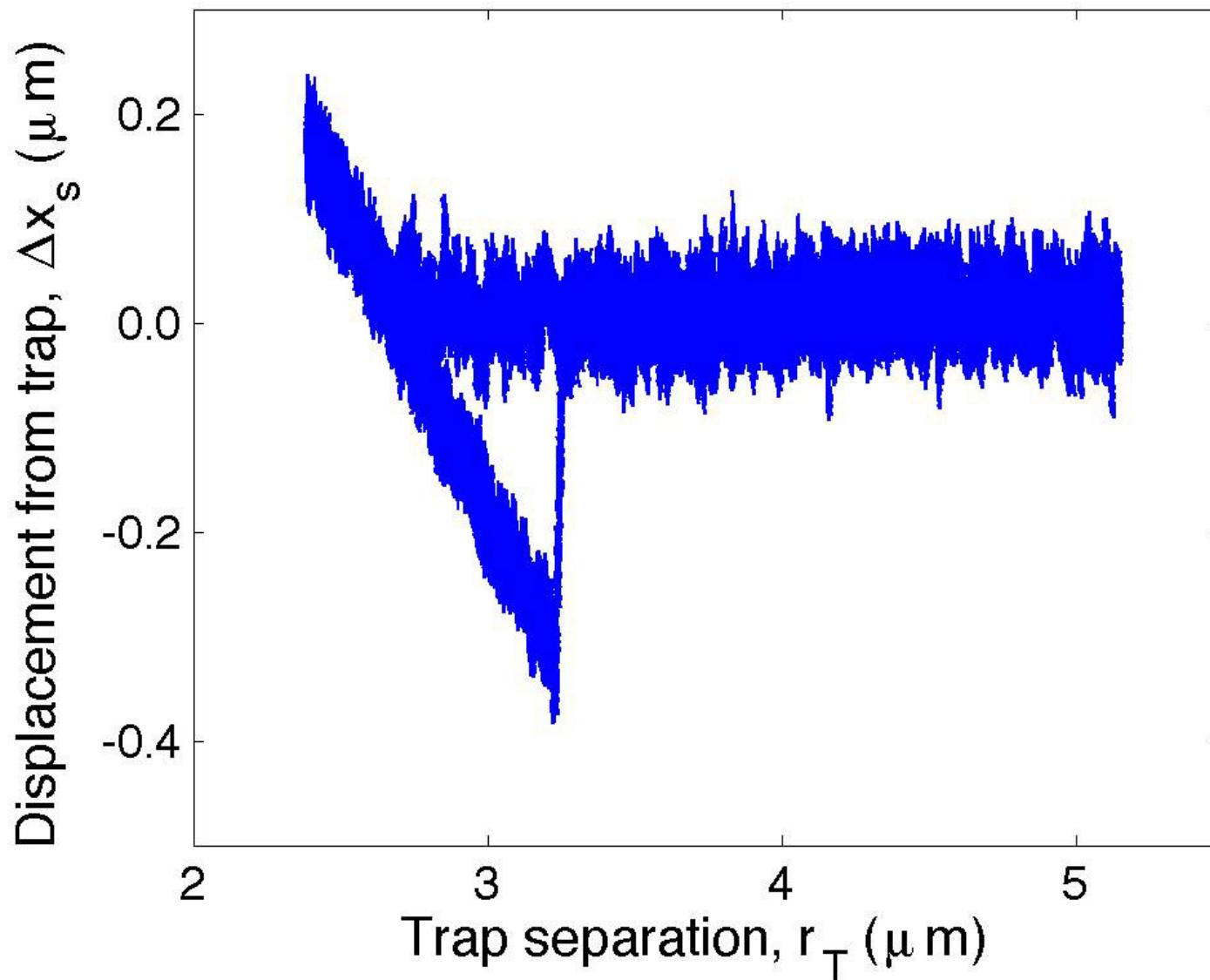
Measuring thermal rupture forces

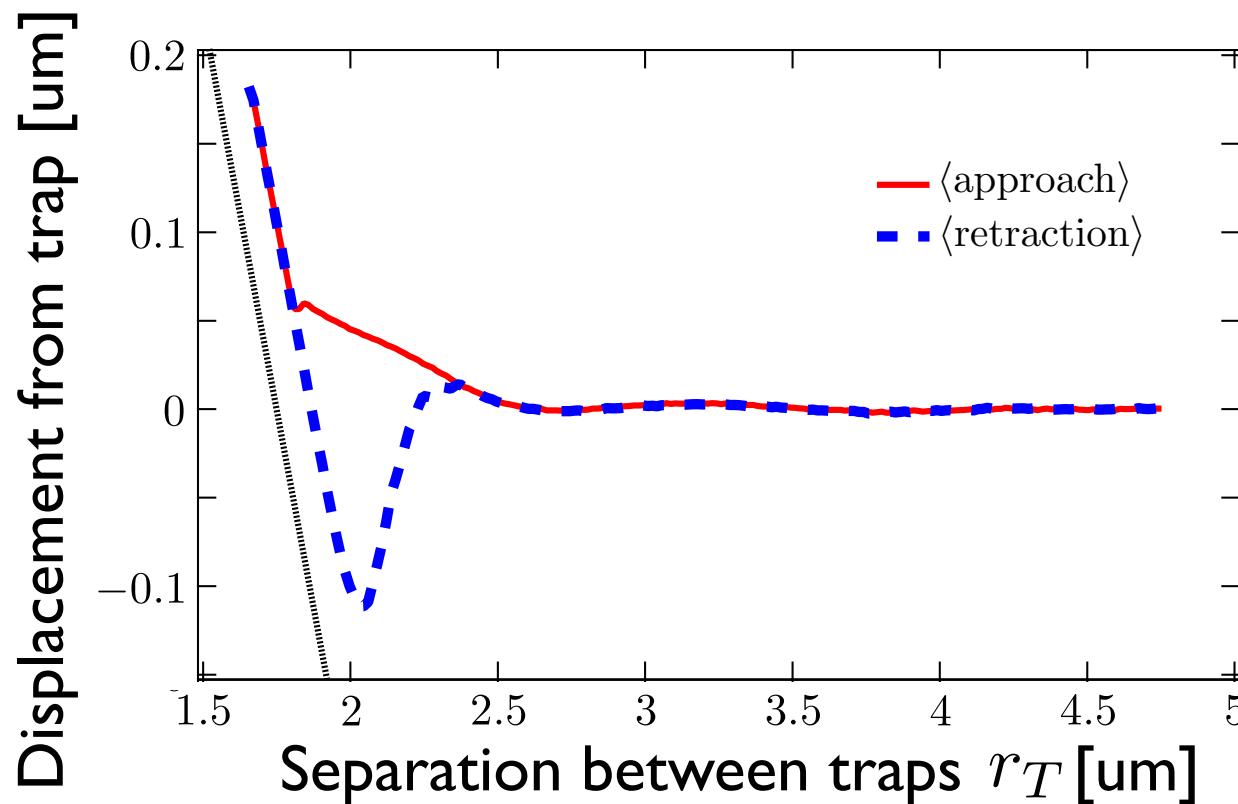


Repeated cycles



Multiple trajectories





Swan, J.W., Shindel,
M. & Furst, E. M.
Phys. Rev. Lett. 109,
198302 (2012).

Two state trajectory model

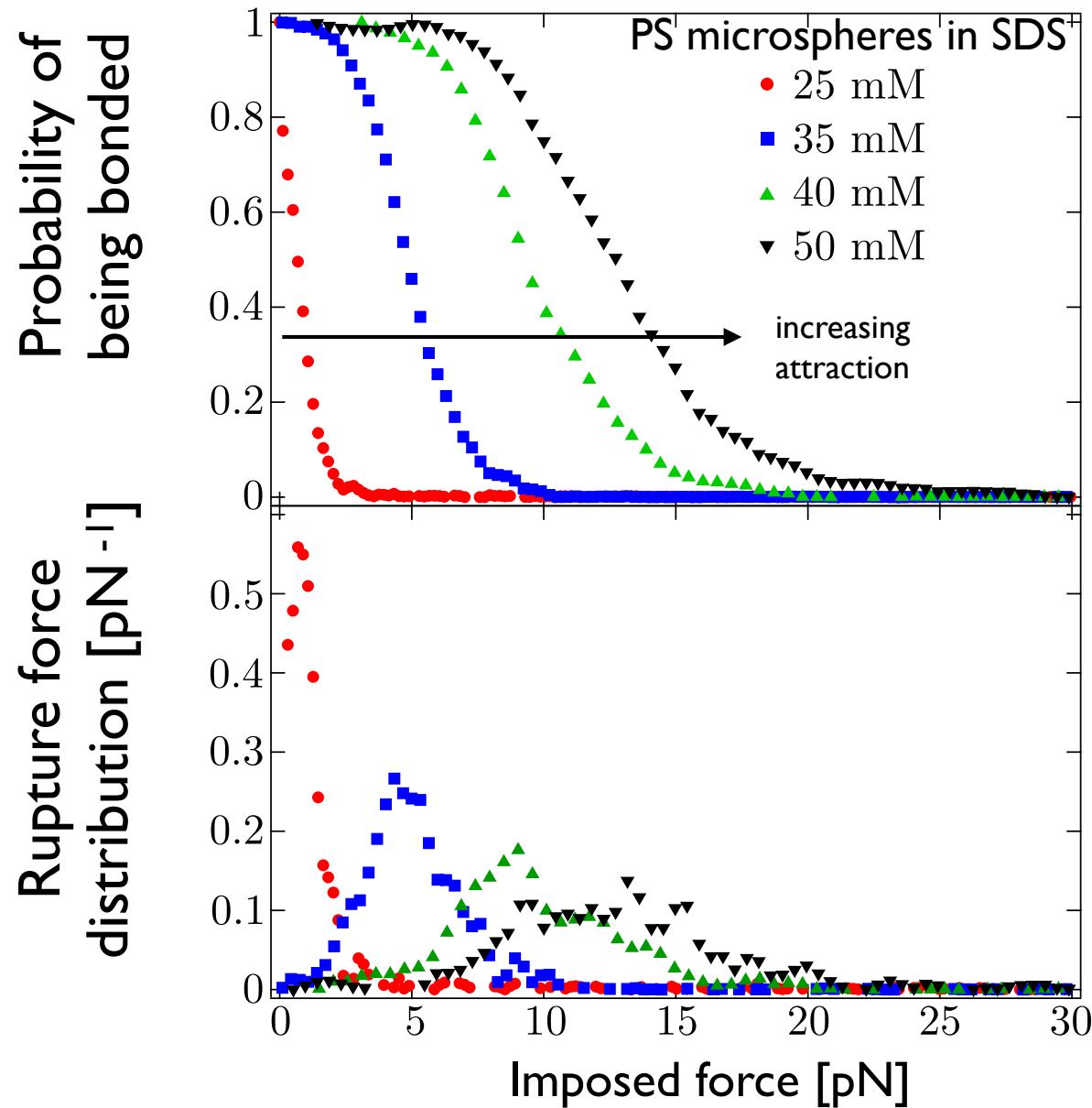
$$\text{retraction} = \begin{cases} -r_T & \text{with } P(r_T) \\ \langle \text{approach} \rangle & \text{with } 1 - P(r_T) \end{cases} \quad \begin{matrix} \text{bonded} \\ \text{ruptured} \end{matrix}$$

Probability of being bonded

$$P(r_T) = \frac{\langle \text{retraction} \rangle - \langle \text{approach} \rangle}{(-r_T) - \langle \text{approach} \rangle}$$

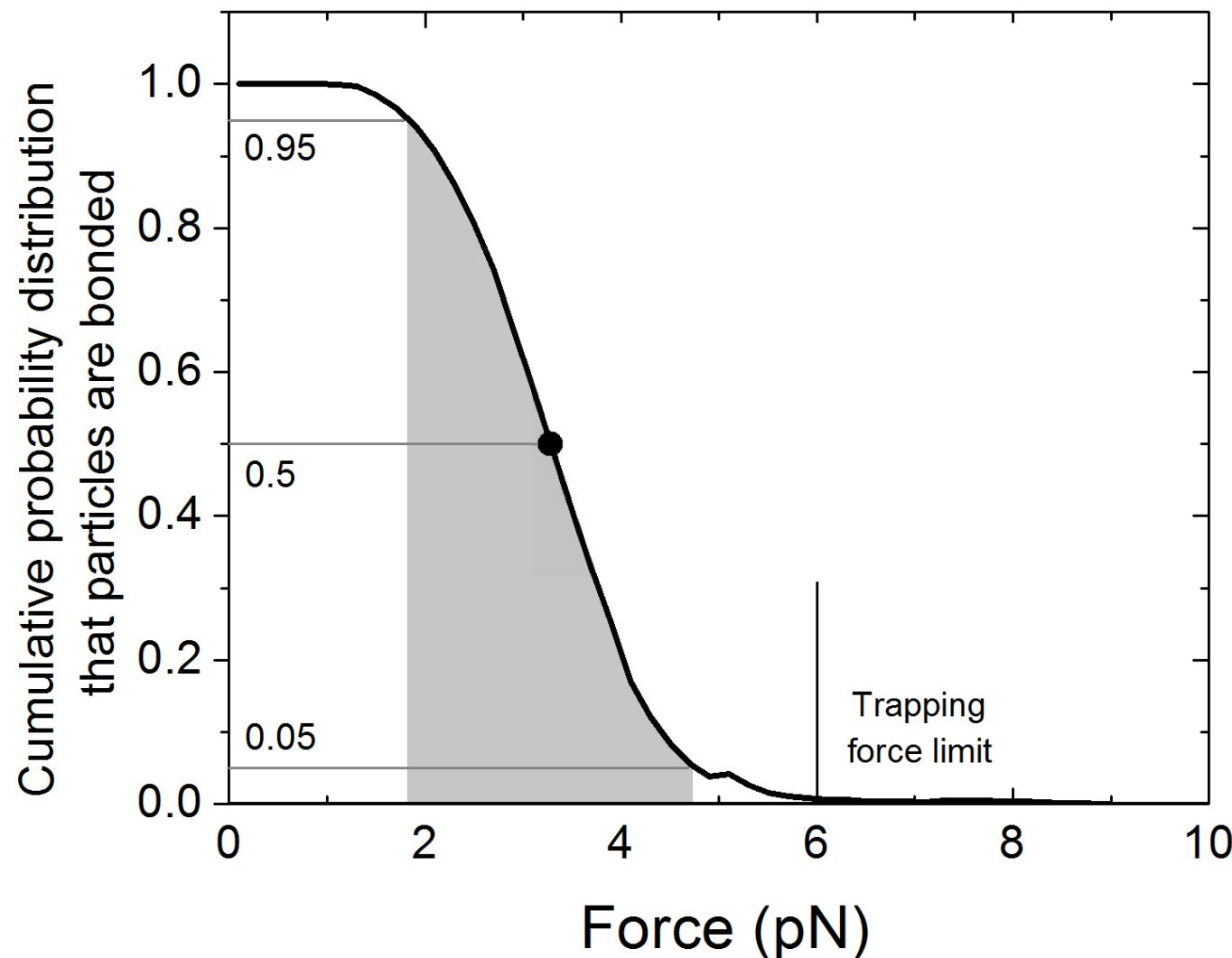
Thermal rupture force distribution

Swan, J. W., Shindel, M. & Furst, E. M. *Phys. Rev. Lett.* 109, 198302 (2012).



PMMA rupture distribution $c/c^* = 0.81$

Hsiao, L. C.; Solomon, M. J.; Whitaker, K. A.; Furst, E. M., *J. Rheology* 2014, 58, 1485–1504.



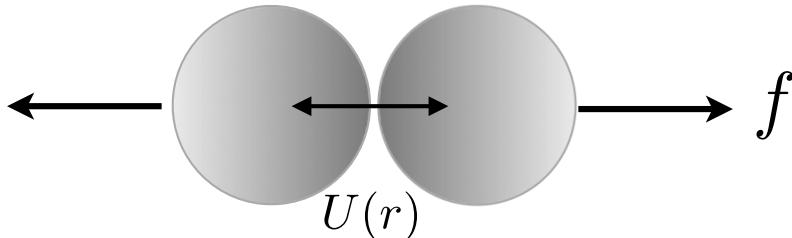
PHSA-PMMA in 75%/25% (v/v) cyclohexane/cyclohexyl bromide

Thermal model of bond rupture

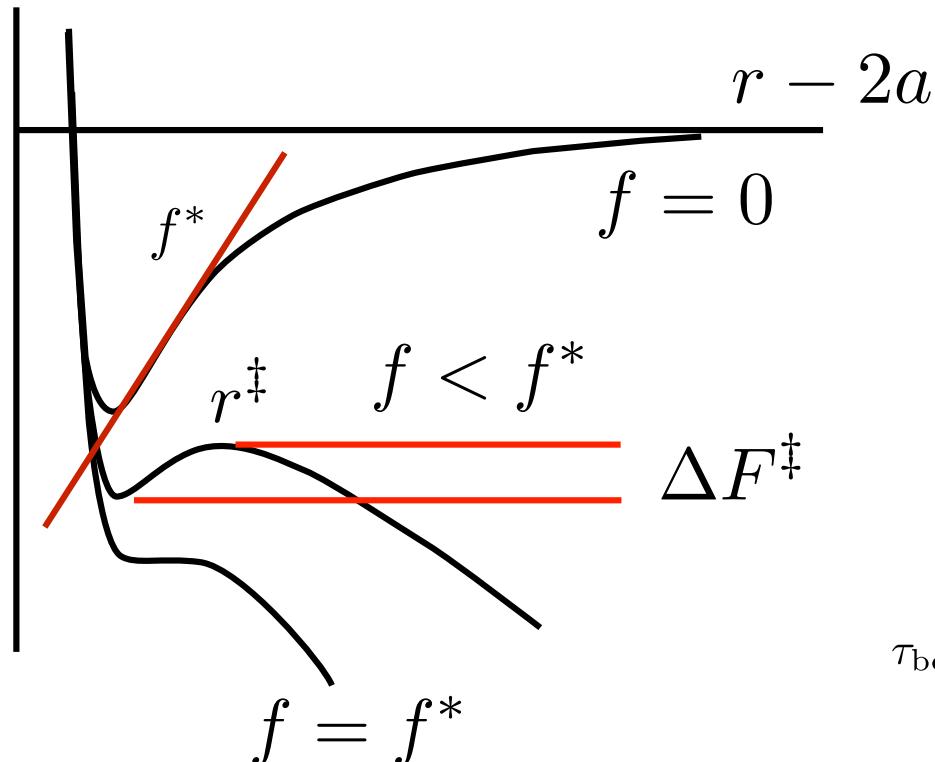
Whitaker, K.A. & Furst, E. M. J. Rheology 60, 517–529 (2016).

H.A. Kramers. *Physica*, 7(4):284–304, 1940.

P. Hanggi, P. Talkner and M. Borkovec. *Reviews of Modern Physics*, 62(2):251–341, 1990.



$$F = U(r) - fr$$



Mean first-passage time

$$\partial P/\partial t = -\nabla \cdot \mathbf{D} \cdot [(\mathbf{f} - \nabla U)P/kT - \nabla P]$$

$$\frac{1}{\tau_{\text{bond}}} = j_+ = D_r \left[\left(f - \frac{dU}{dr} \right) \frac{P}{kT} - \frac{dP}{dr} \right]$$

$$j_+ = -D_r \left[\frac{dP}{dr} + \frac{dF(r)}{dr} \frac{P}{kT} \right]$$

$$\tau_{\text{bond}} = \frac{1}{j_+} = \int_{2a}^{r^\ddagger} e^{-F(r)/kT} \left[\int_r^{r^\ddagger} \frac{1}{D_r(y)} e^{F(y)/kT} dy \right] dr$$

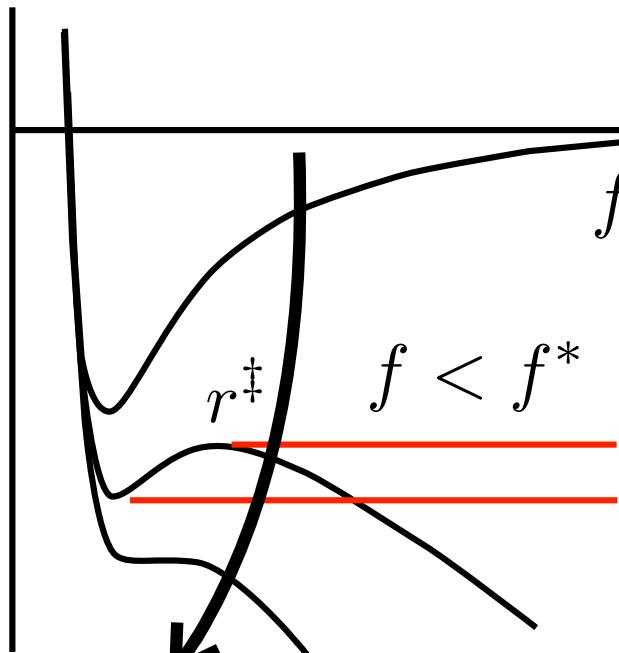
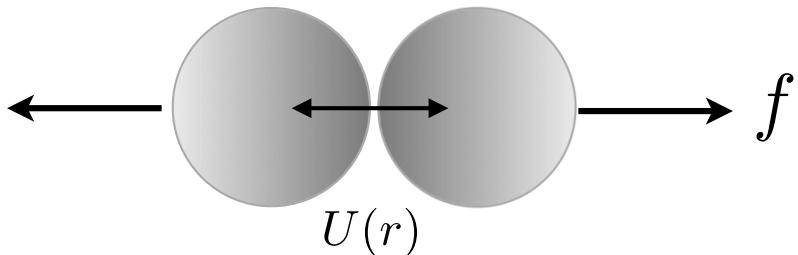
$$D_r = \frac{2k_B T(r - 2a)}{3\pi a^2 \eta}$$

Thermal model of bond rupture

Whitaker, K.A. & Furst, E. M. J. Rheology 60, 517–529 (2016).

H.A. Kramers. *Physica*, 7(4):284–304, 1940.

P. Hanggi, P. Talkner and M. Borkovec. *Reviews of Modern Physics*, 62(2):251–341, 1990.



$$r = 2a$$

$$f = 0$$

$$f < f^*$$

$$\Delta F^\ddagger$$

$$r_f \quad f = f^*$$

Bond lifetime

$$\tau_{\text{bond}} = \frac{1}{j_+} = \int_{2a}^{r^\ddagger} e^{-F(r)/kT} \left[\int_r^{r^\ddagger} \frac{1}{D_r(y)} e^{F(y)/kT} dy \right] dr$$

Rate of bond rupture

$$\frac{dS_1}{dt} \approx -j_+ S_1$$

Rate of loading

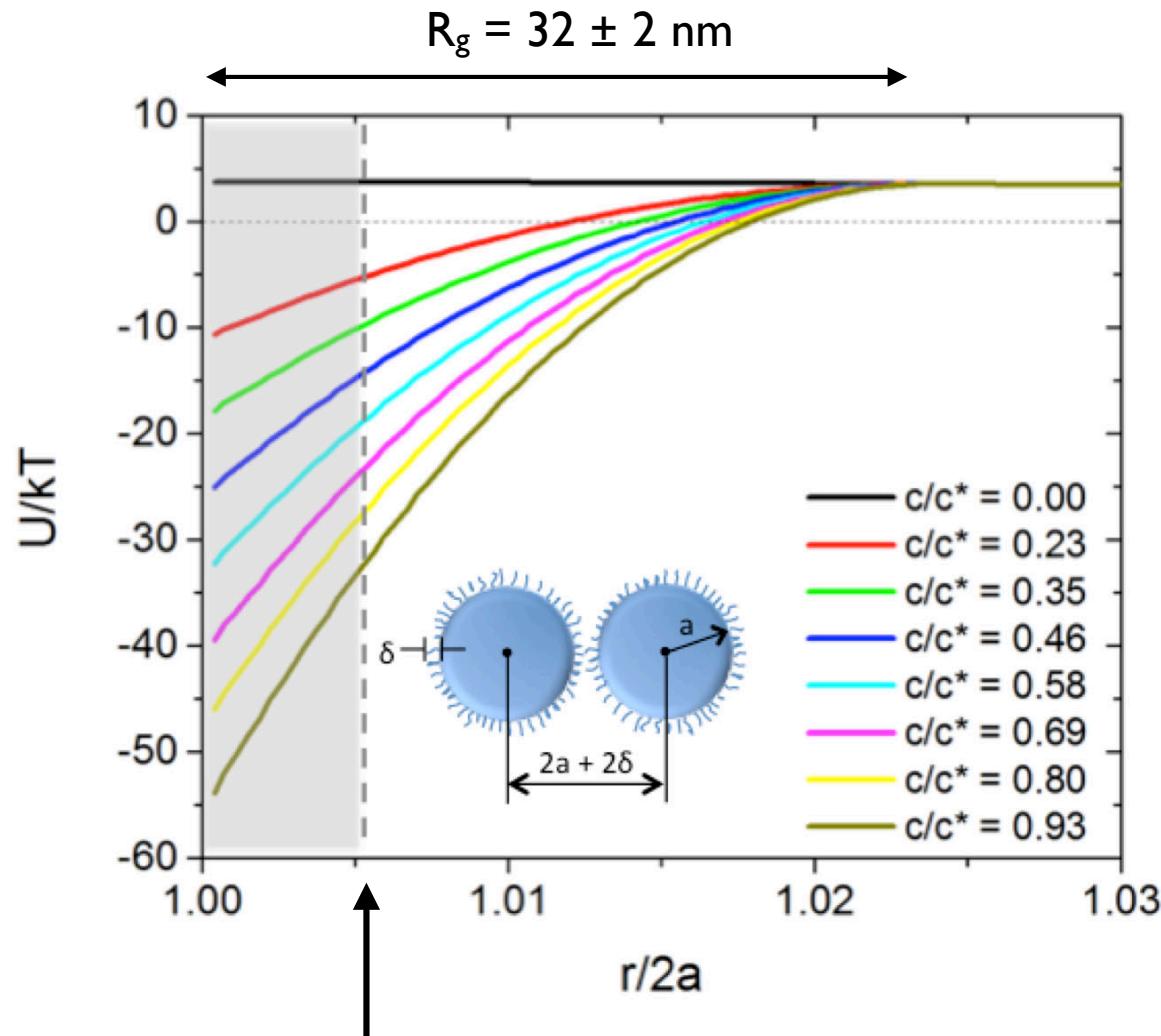
$$r_f \frac{dS_1}{df} \approx -j_+ S_1$$

Cumulative bond probability distribution

$$S_1(f) = \exp \left(-\frac{1}{r_f} \int_0^f j_+(f) df \right)$$

DLVO+AO interaction potential

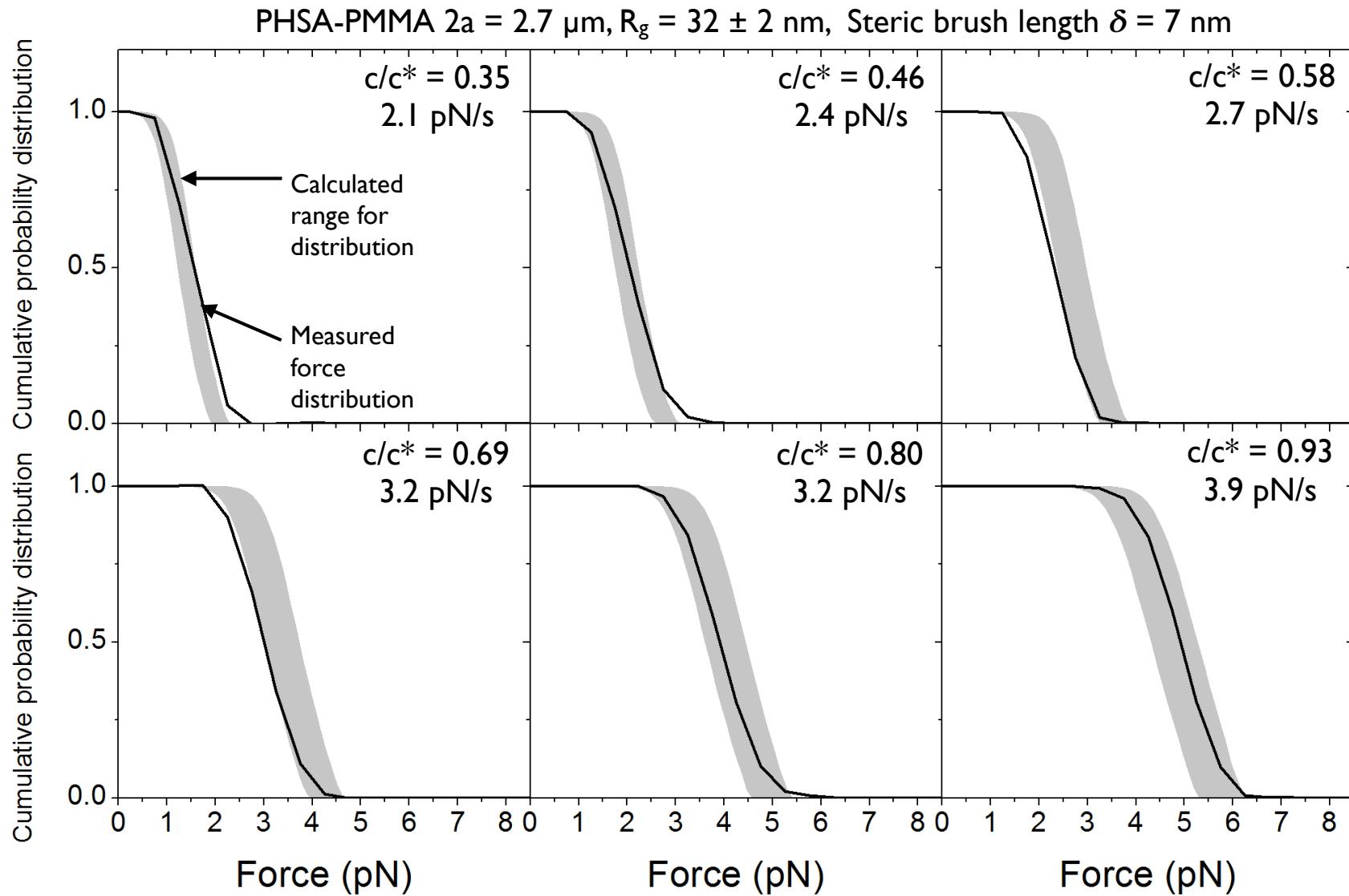
PHSA-PMMA $2a = 2.7 \mu\text{m}$, $R_g = 32 \pm 2 \text{ nm}$, Steric brush length $\delta = 7 \text{ nm}$



PHSA brush cutoff, 7nm

Rupture force distribution calculated from theoretical AO potential

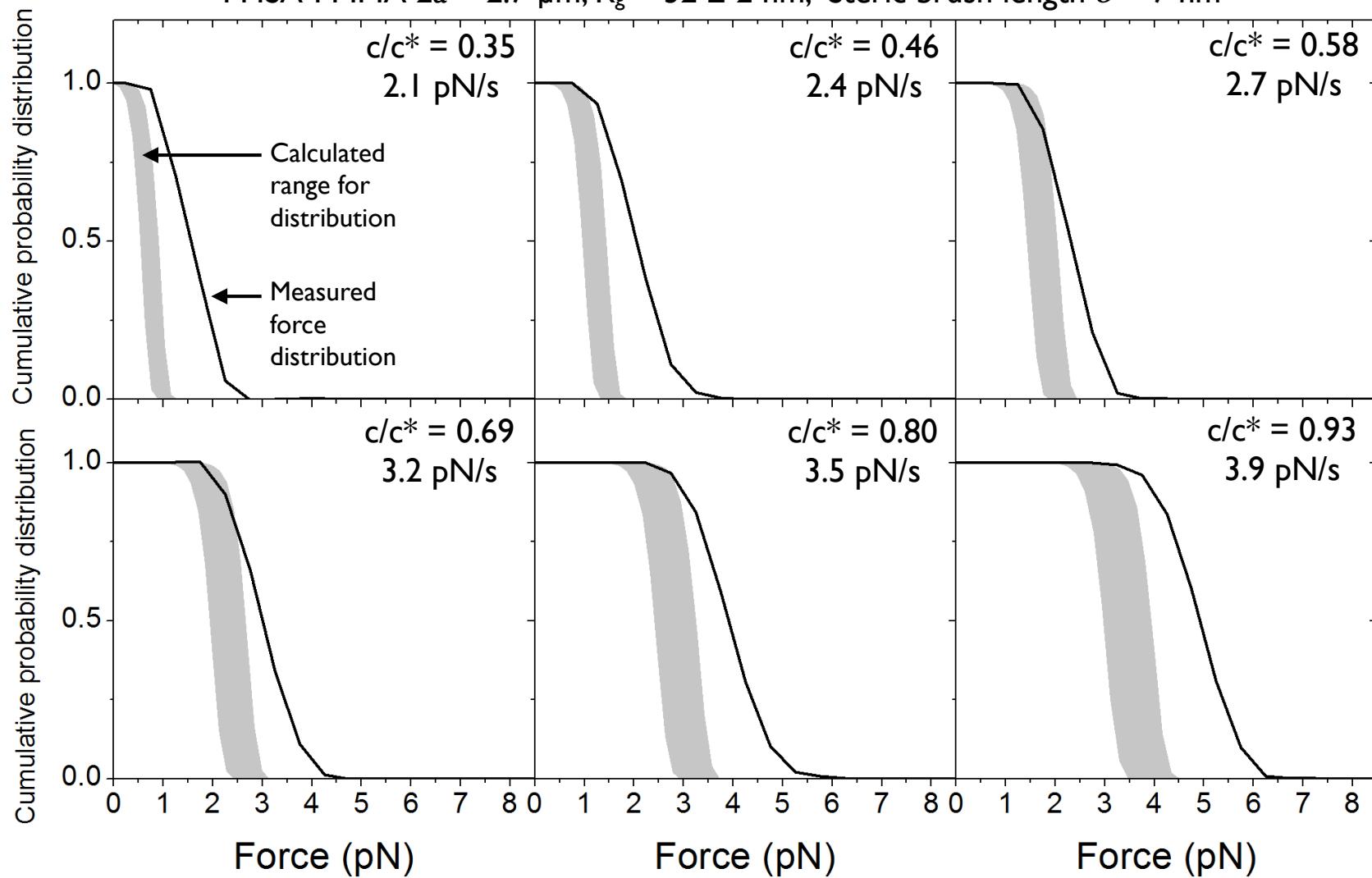
Whitaker, K.A. & Furst, E. M.J. *Rheology* 60, 517–529 (2016).



Rupture forces calculated from AO potential

Whitaker, K.A. & Furst, E. M. J. Rheology 60, 517–529 (2016).

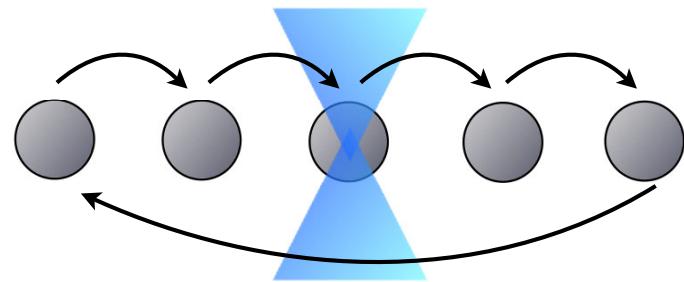
PHSA-PMMA $2a = 2.7 \mu\text{m}$, $R_g = 32 \pm 2 \text{ nm}$, Steric brush length $\delta = 7 \text{ nm}$



Directed assembly of aggregates

J. P. Pantina and E. M. Furst. Langmuir, 20:3940–3946, 2004.

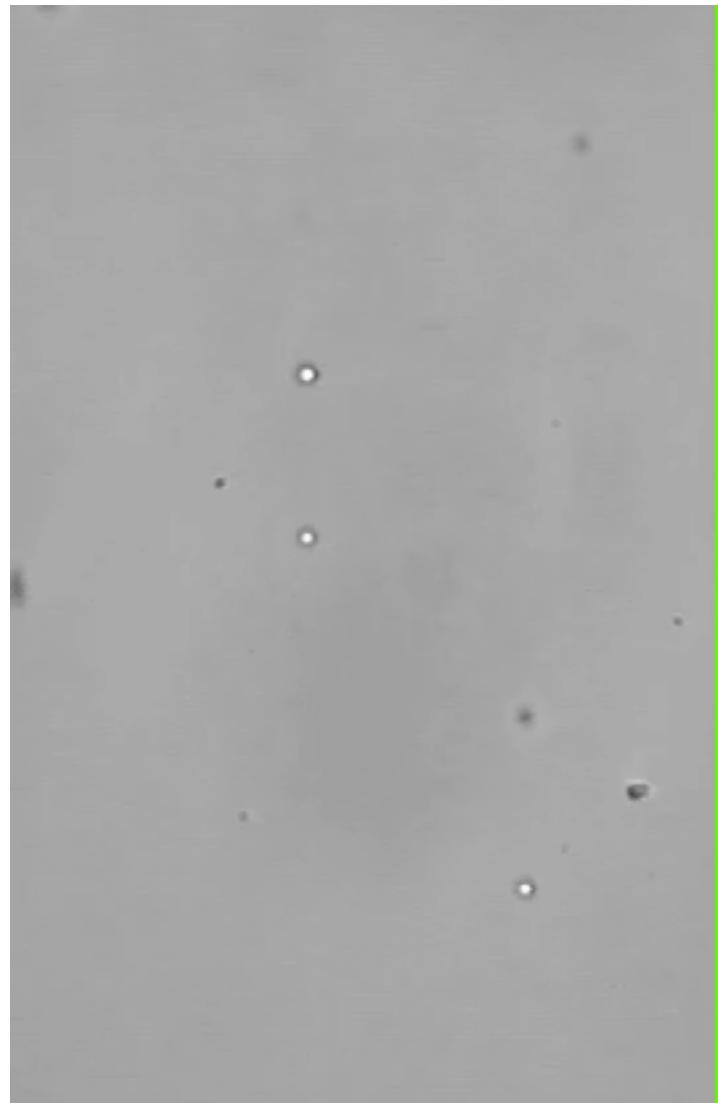
Time shared and scanning
optical traps



$$\nu = \frac{2k_B T [\operatorname{erf}^{-1}(\gamma)]^2}{3\pi a^3 b^2}$$

Mio and Marr, Langmuir. 15, 8565 (1999)
Faucheux et al., Phys. Rev. E. 51, 5239 (1995)
Crocker et al, Phys. Rev. Lett. 82, 4152 (1999)

High bandwidth positioning using AOD
Assembly in salt solutions



10 μm

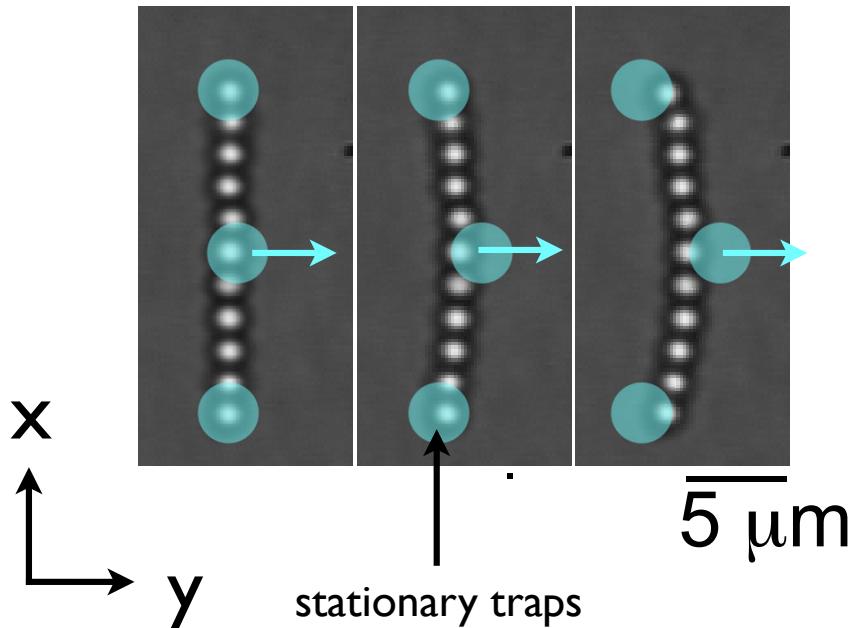
2X real-time

Linear elastic behavior of aggregates

Pantina and Furst, Phys. Rev. Lett. 94:138301, 2005.

1.47 μm PMMA, I I particles, 250mM MgCl₂

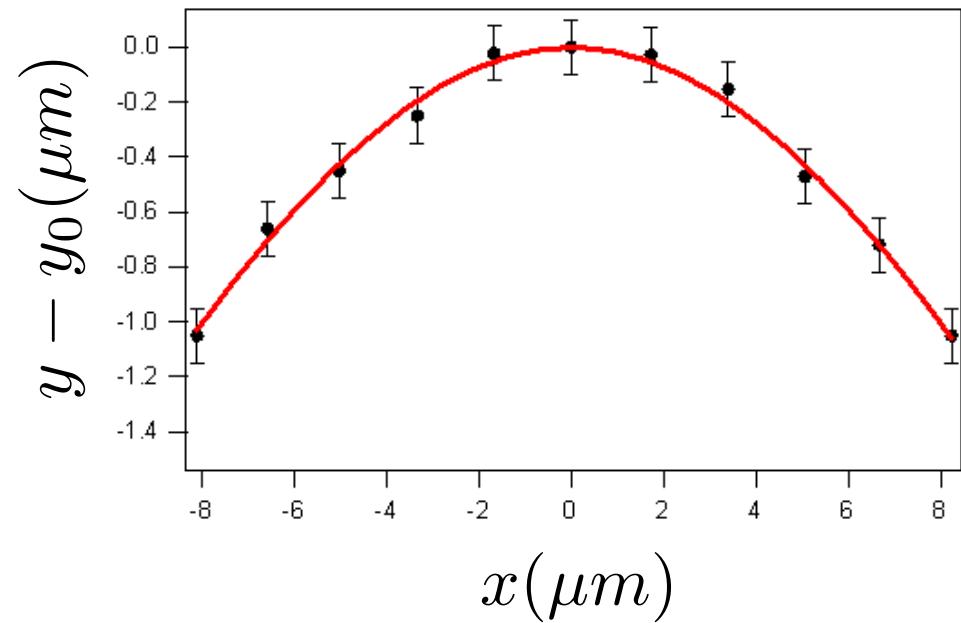
3-point bending geometry



$$F = \kappa_T \Delta x$$

Note: aggregates > 100 μm from interface

Comparison to beam equation

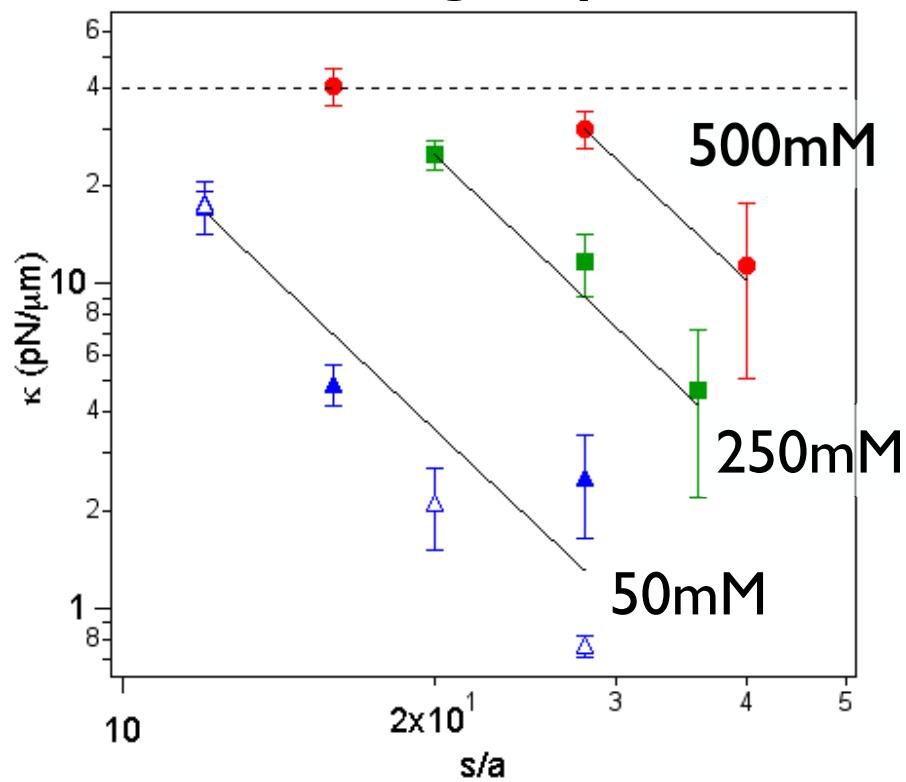


$$y_i(x_i) = -\frac{F}{EI} \left(\frac{Lx_i^2}{4} - \frac{|x_i^3|}{6} \right)$$

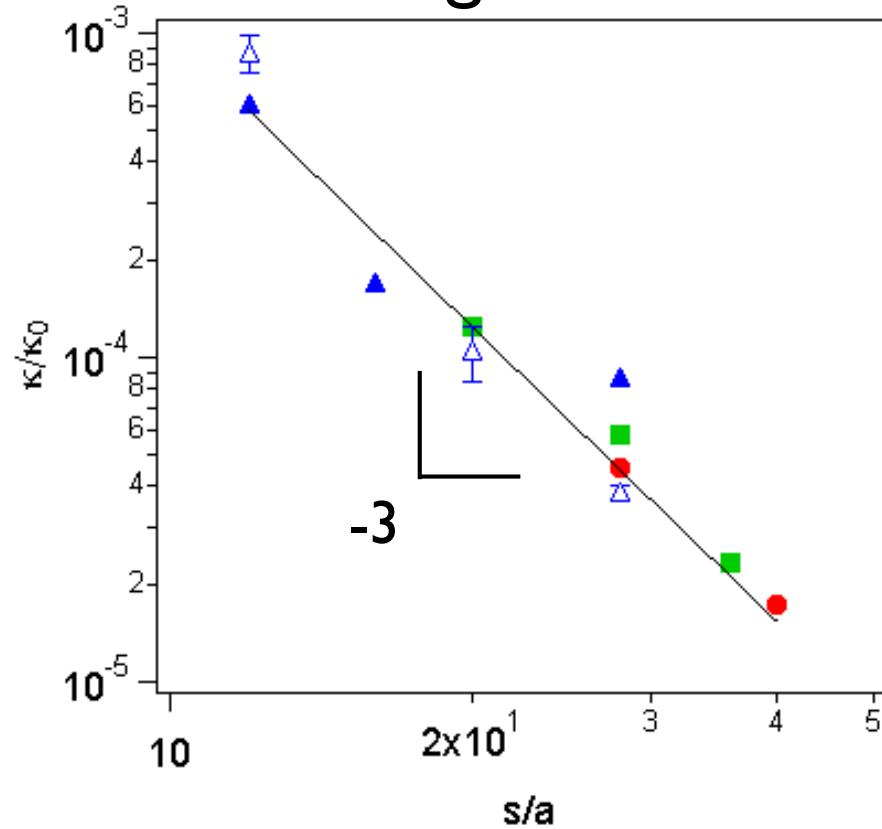
Linear bending mechanics in MgCl_2

Pantina and Furst, Phys. Rev. Lett. 94:138301, 2005.

Rigidity



Scaling behavior

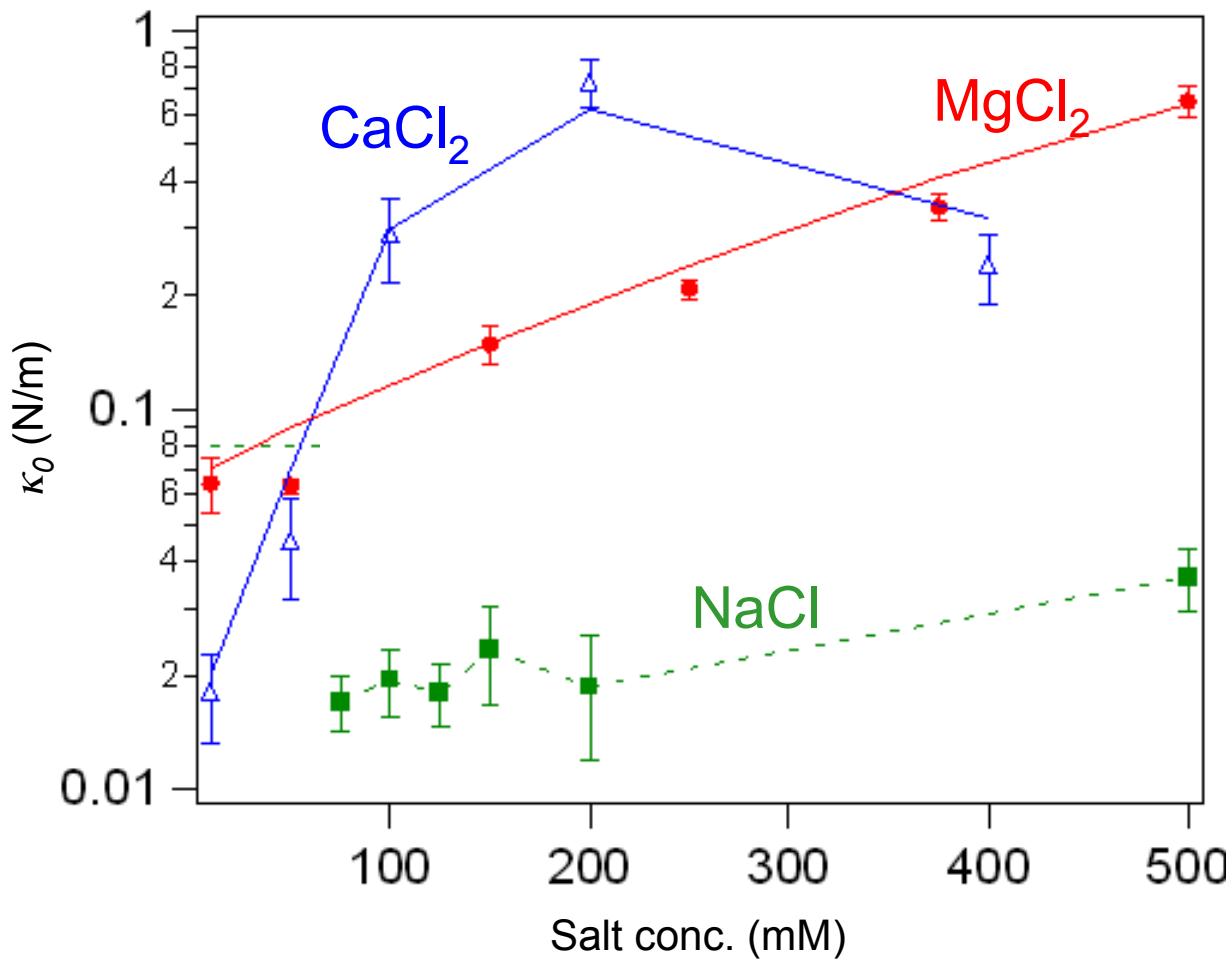


Open symbols: PS
Closed symbols: PMMA

Aggregate rigidity displays surprising dependence on salt concentration

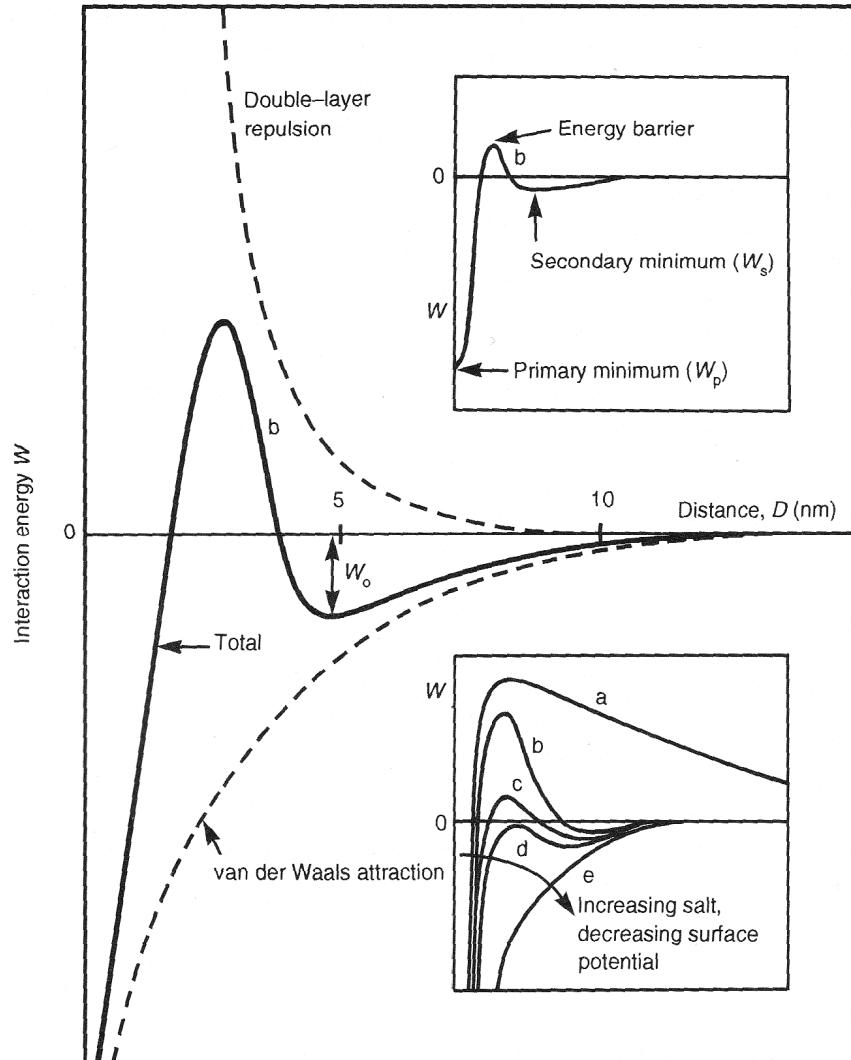
Single bond bending rigidity

1.47 μm PMMA



Typical colloidal interactions

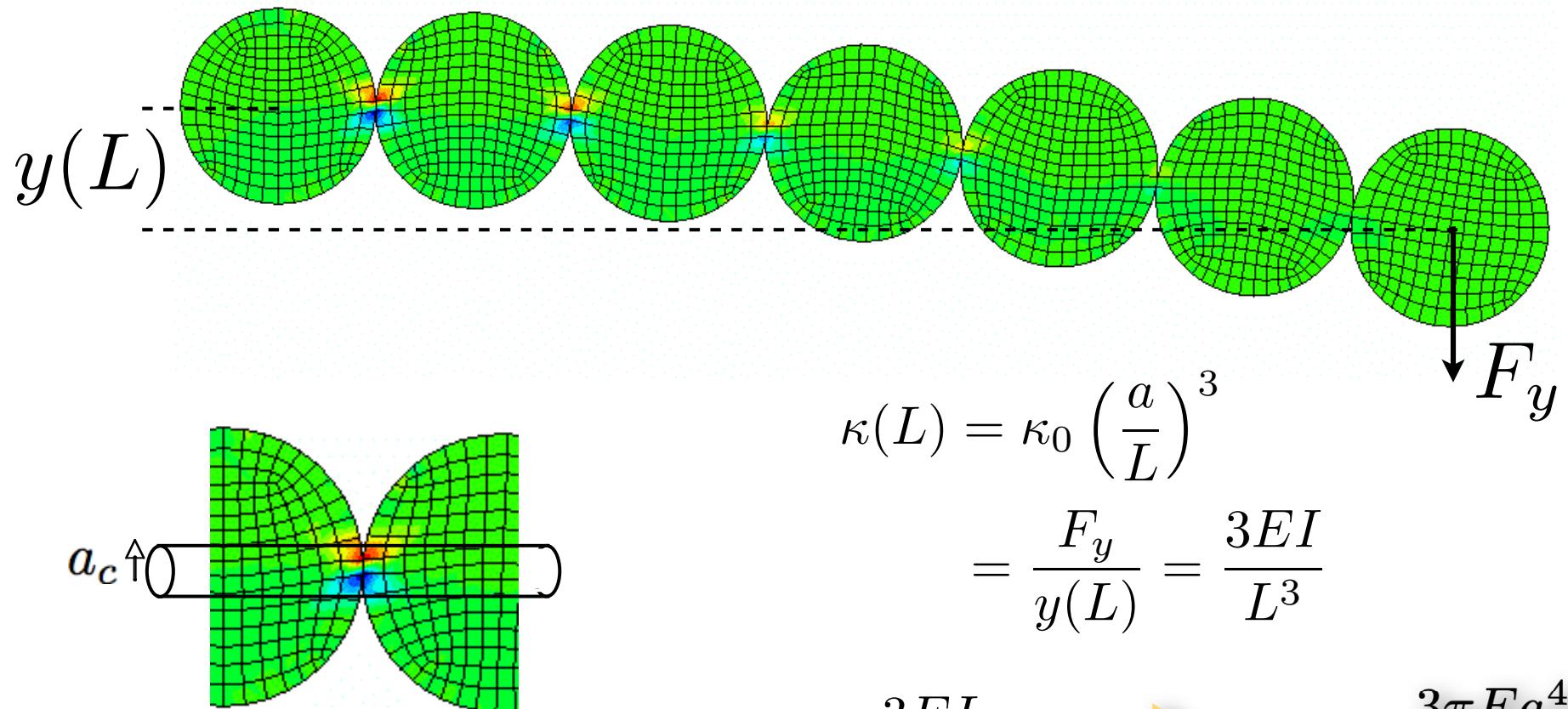
DLVO-van der Waals + double layer



DLVO cannot account for tangential forces or salt dependence

Israelachvili, Intermolecular and Surface Forces

Bond rigidity



Area moment of inertia

$$I = \frac{\pi a_c^4}{4}$$

$$\begin{aligned}\kappa(L) &= \kappa_0 \left(\frac{a}{L} \right)^3 \\ &= \frac{F_y}{y(L)} = \frac{3EI}{L^3}\end{aligned}$$

$$\kappa_0 = \frac{3EI}{a^3} \quad \rightarrow \quad \kappa_0 = \frac{3\pi E a_c^4}{a^3}$$

Bending rigidity in terms of contact radius

JKR Theory

Johnson, Kendall and Roberts, Proc. R. Soc. London, Ser. A 324, 301 (1971).

$$P_0 = 0 \quad a_c = \left[\frac{3\pi a^2 W_{SL}}{2K} \right]^{1/3}$$

Surface adhesion energy:

$$W_{SL} = \gamma_S + \gamma_L - \gamma_{SL}$$

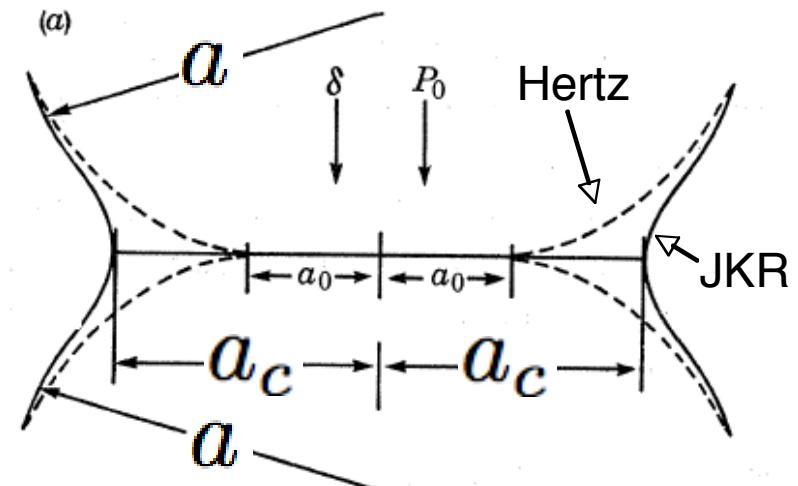
$$W_{SL} = \gamma_L(1 + \cos \theta_0) \approx 93 \text{ mJ/m}^2$$

where¹ $\theta_0 = 73.7 \pm 0.3^\circ$

$$\gamma_L \approx 72 - 73 \text{ mN/m}$$

Elastic modulus: $K = \frac{2E}{3(1 - \nu^2)}$

$$K \approx 2.5 \text{ GPa}$$



Single-bond rigidity:

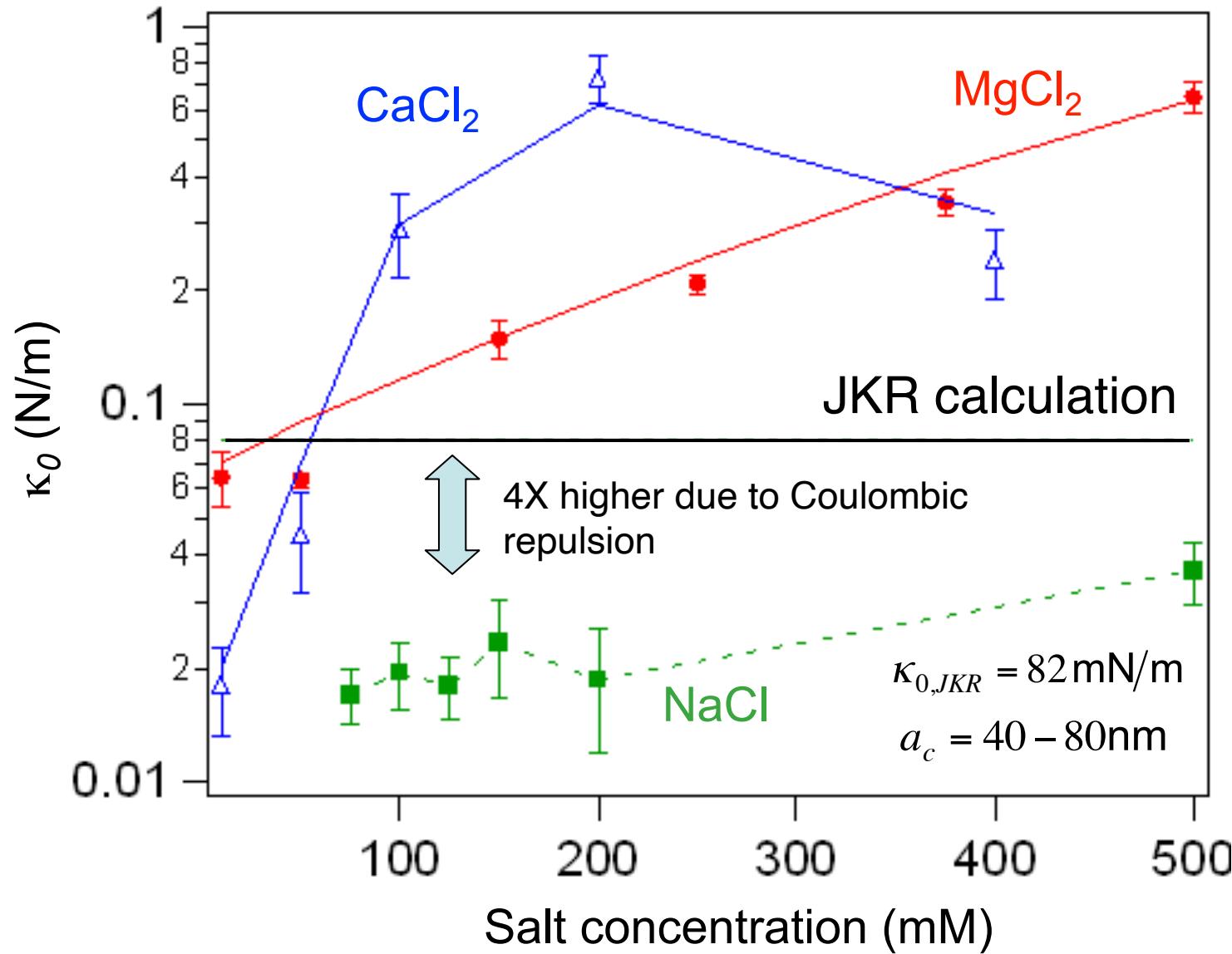
$$\kappa_0 = \frac{3\pi E a_c^3}{a^3} = \frac{3\pi E}{4a^3} \left(\frac{3\pi a^2 W_{SL}}{2K} \right)^{4/3}$$

Relates mechanics to adhesion energy
(particle interfacial phenomena)

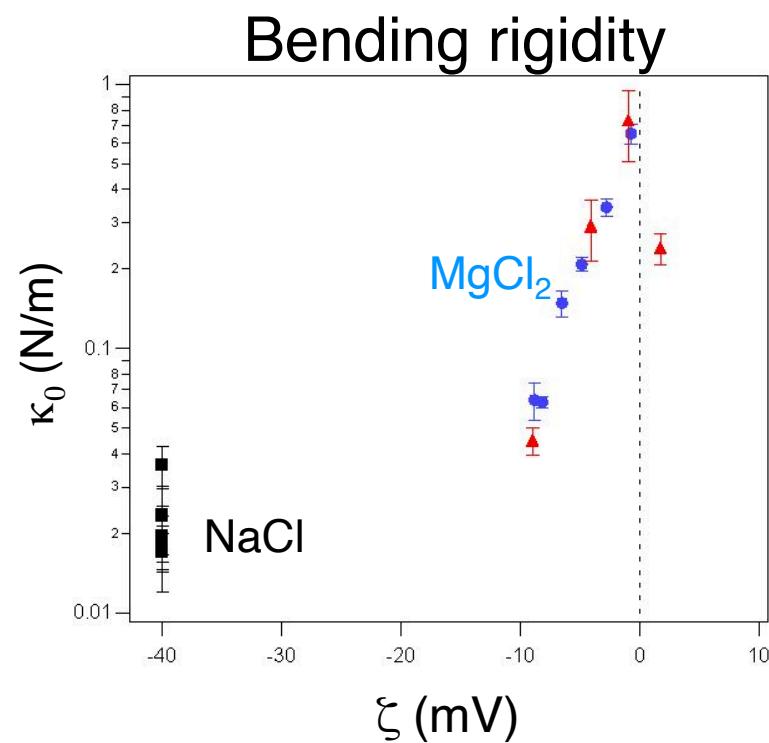
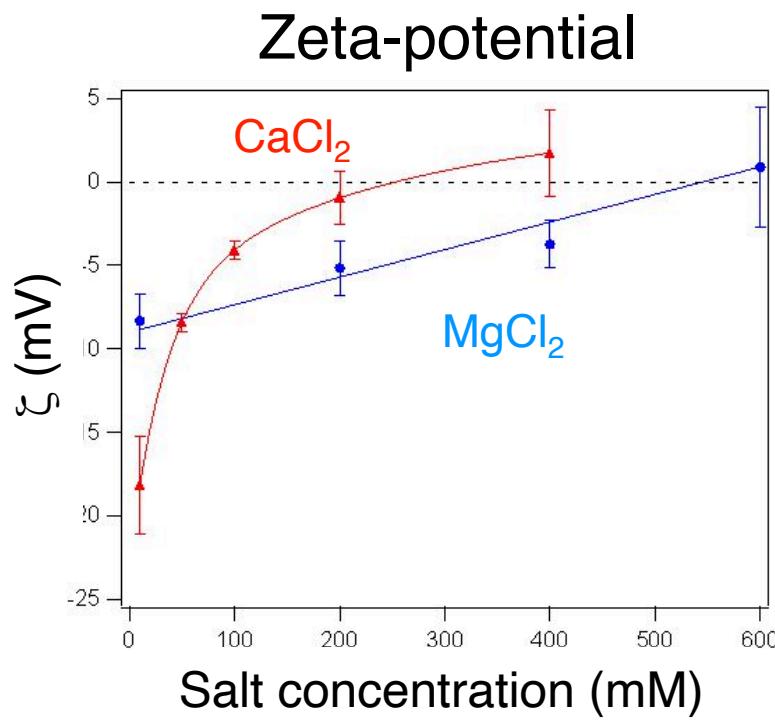
¹ Kwok et al. J. Colloid Interface Sci. 206, 44 (1998).

² Jarvis and Scheiman, J. Phys. Chem. 72, 74 (1968). Chemical & Biomolecular Engineering, University of Delaware — furst@udel.edu

Calculated bending rigidity



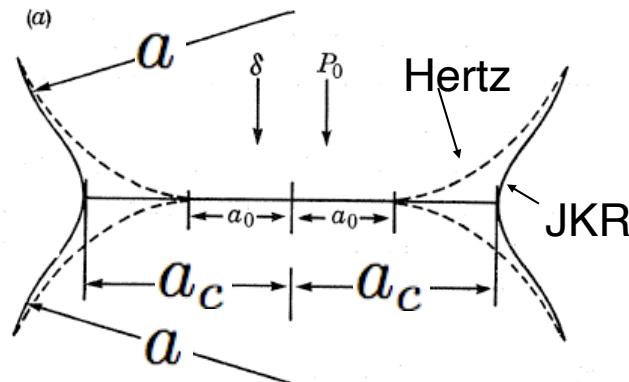
Divalent ion adsorption to PMMA



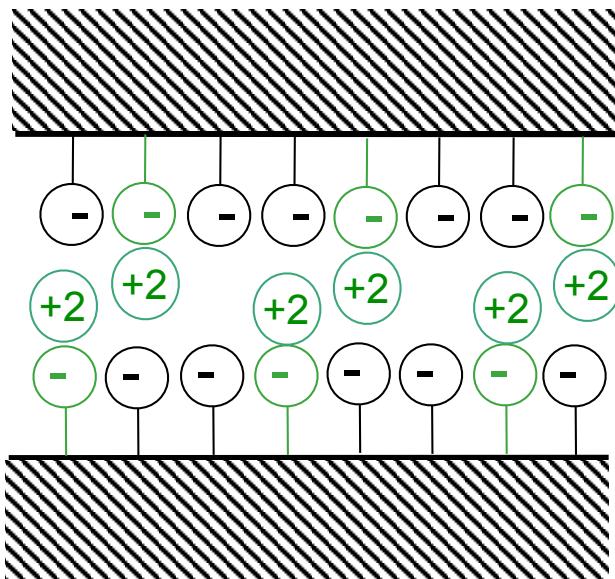
- Bond rigidity collapses with zeta potential
- Maximum at point of zero charge
- Calculate fractional coverage of divalent ion

$$\theta = \frac{[C^{++}S^{-1}]}{[S]^{Tot}}$$

Ion-bridge model



Divalent ions “bridge” charged adsorption sites



Total adhesion energy

$$W_{SL} = W_{SL}^0 + W_{SL}^{ES} + W_{SL}^{div}$$

vdw **Coulomb** **ion
bridge**

$$W_{SL}^{div} \sim \theta(1 - \theta)$$

$$W_{SL}^{ES} \sim -(\sigma/e)^2 \pi a_c^2$$

$$\sim -(1 - 2\theta)^2 \pi a_c^2$$

$$\kappa_0 = \frac{3\pi E a_c^3}{a^3} = \frac{3\pi E}{4a^3} \left(\frac{3\pi a^2 W_{SL}}{2K} \right)^{4/3}$$

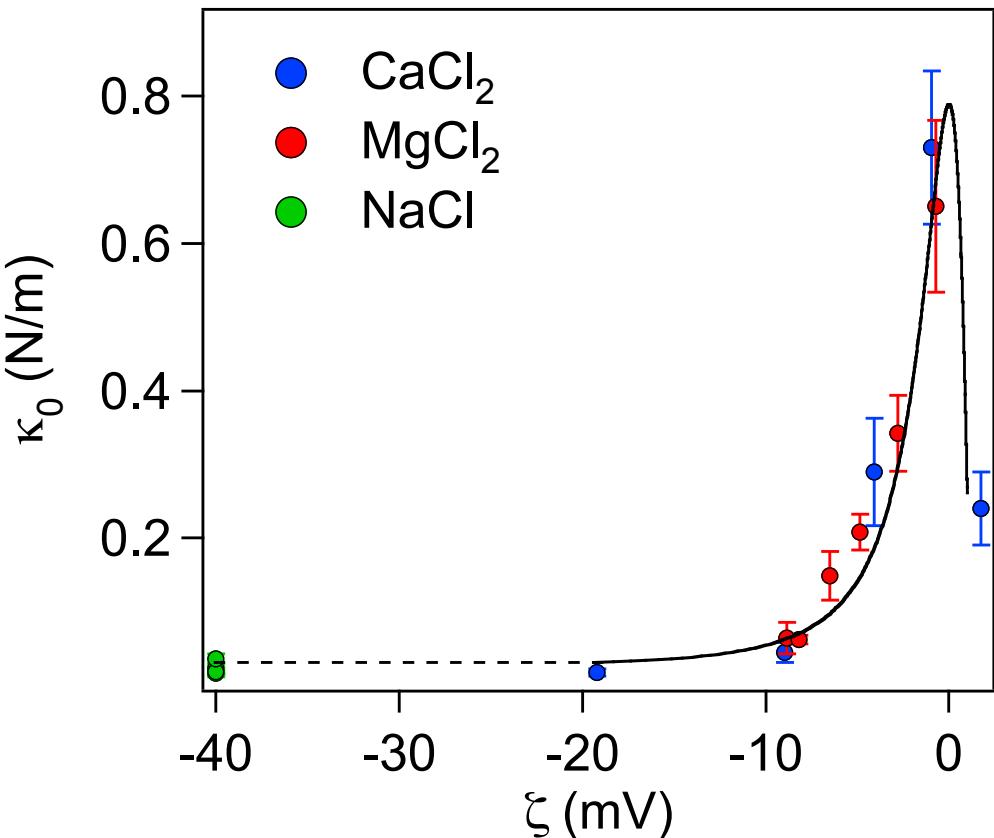
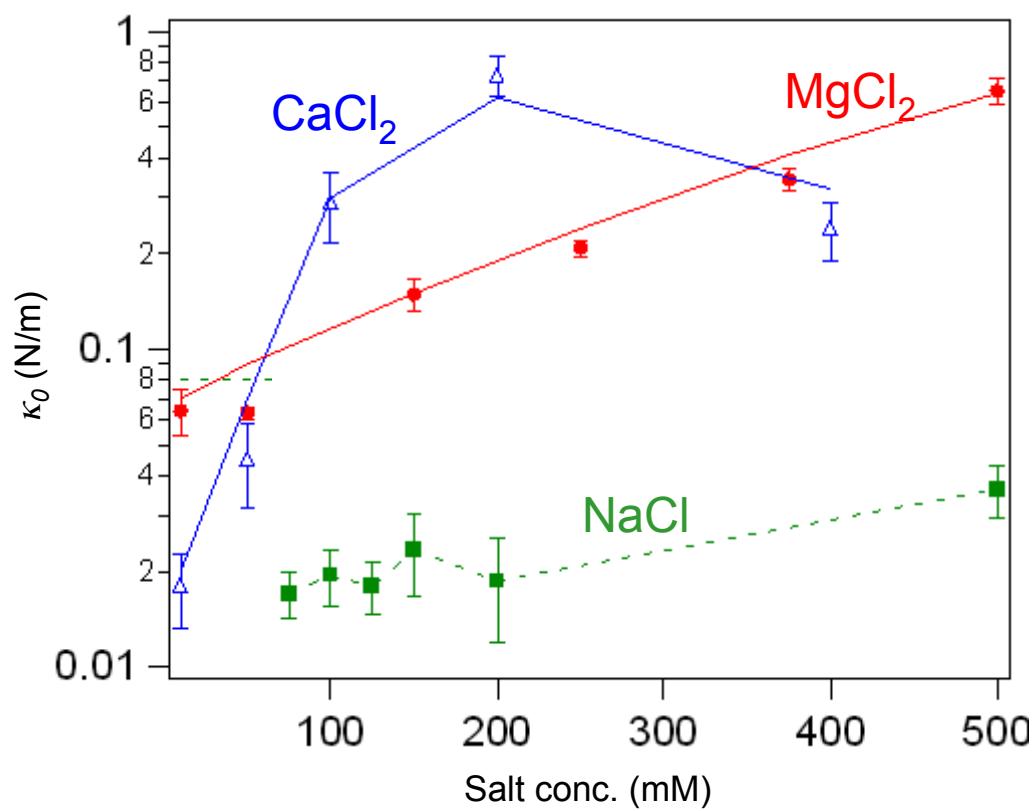
Ion bridge model

Pantina and Furst, Langmuir, 22:5282–5288, 2006.

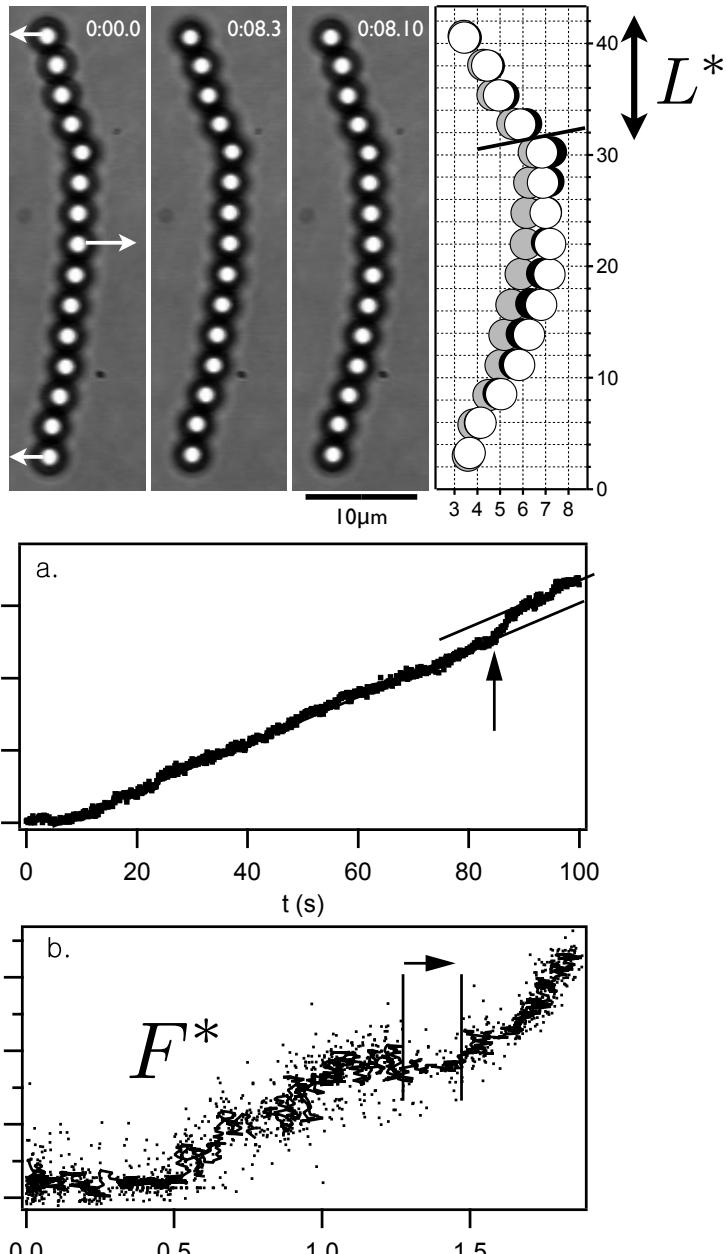
$$W_{SL} = W_{SL}^0 + W_{SL}^{ES} + W_{SL}^{div}$$

vdw Coulomb ion bridge

$$\kappa_0 = \frac{3\pi E}{4a^3} \left(\frac{3\pi a^2 W_{SL}}{2K} \right)^{4/3}$$

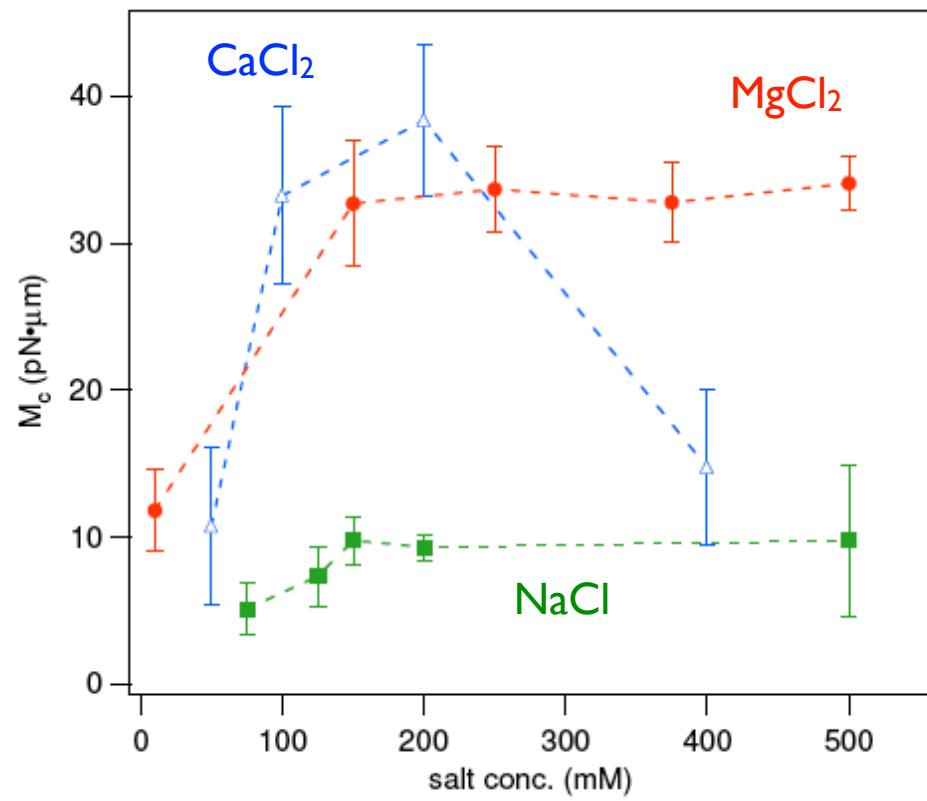


Non-linear mechanics



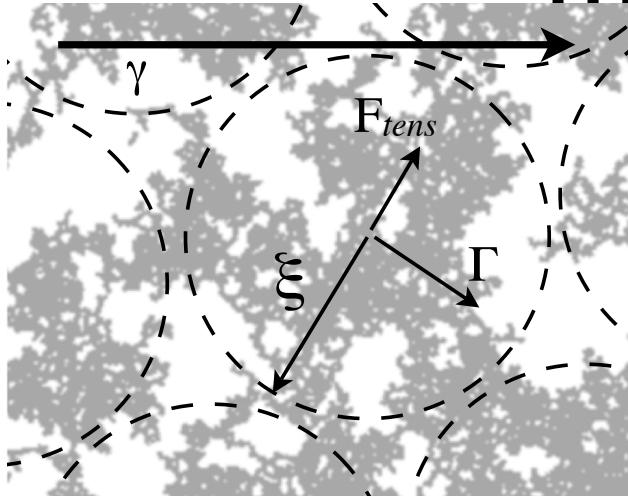
Critical bending moment

$$M_c = L^* F^*$$

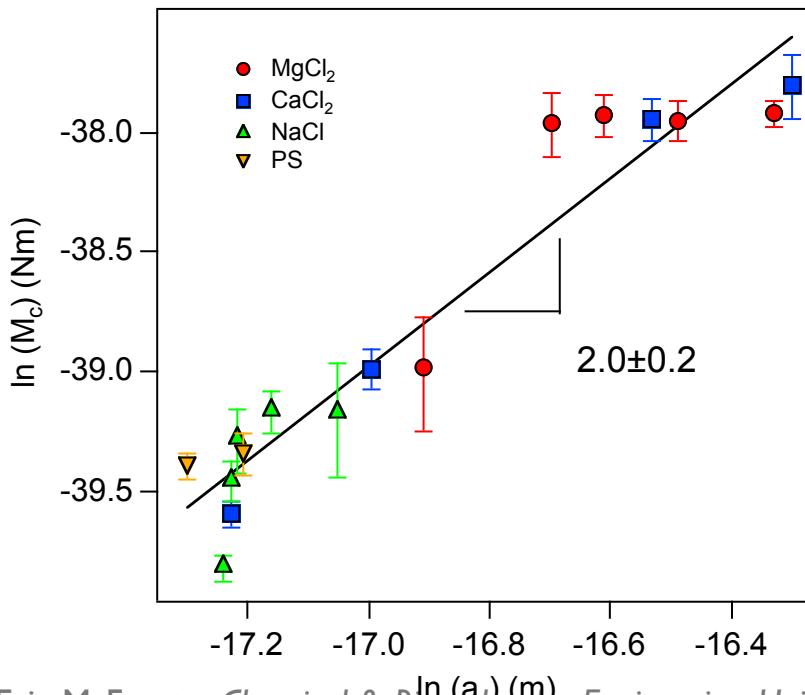


New microscopic yielding mechanism

Yield stress due to critical moment



Critical moment versus contact radius



$$\xi \sim a(\phi/\phi_e)^{3/(d_f-3)}$$

Maximum tensile force

$$F_{tens} \sim \xi^2 \sigma_y$$

Maximum bending moment

$$\Gamma \sim \xi^3 \sigma_y$$

Predicted yield stress

$$\sigma_y^{bend} \sim M_c a^{-3} (\phi/\phi_e)^{3/(3-d_f)}$$

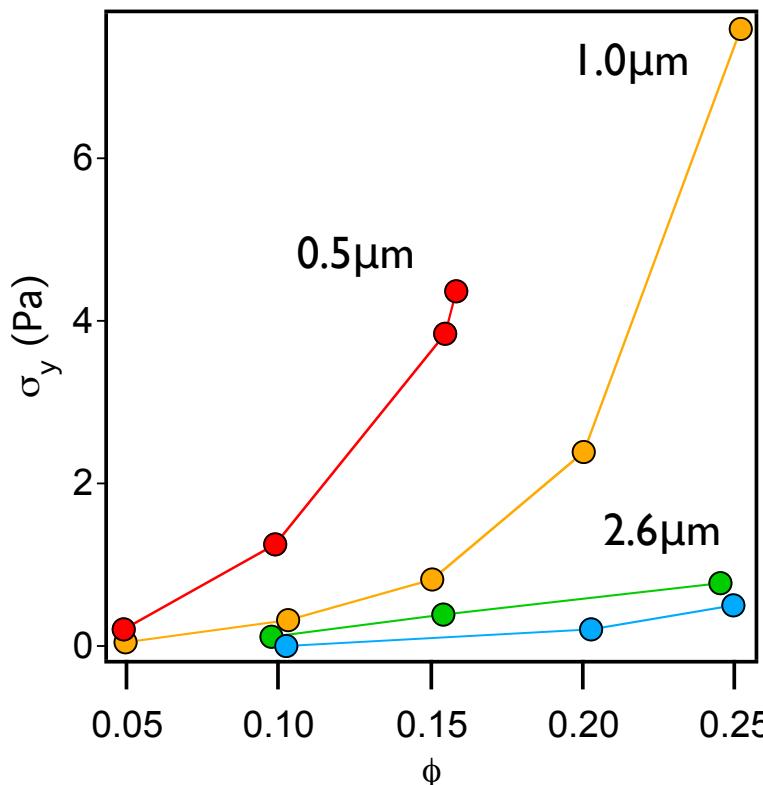
Predicted particle size dependence

$$M_c \sim a^{4/3}$$

$$\sigma_y^{bend} \sim a^{-5/3}$$

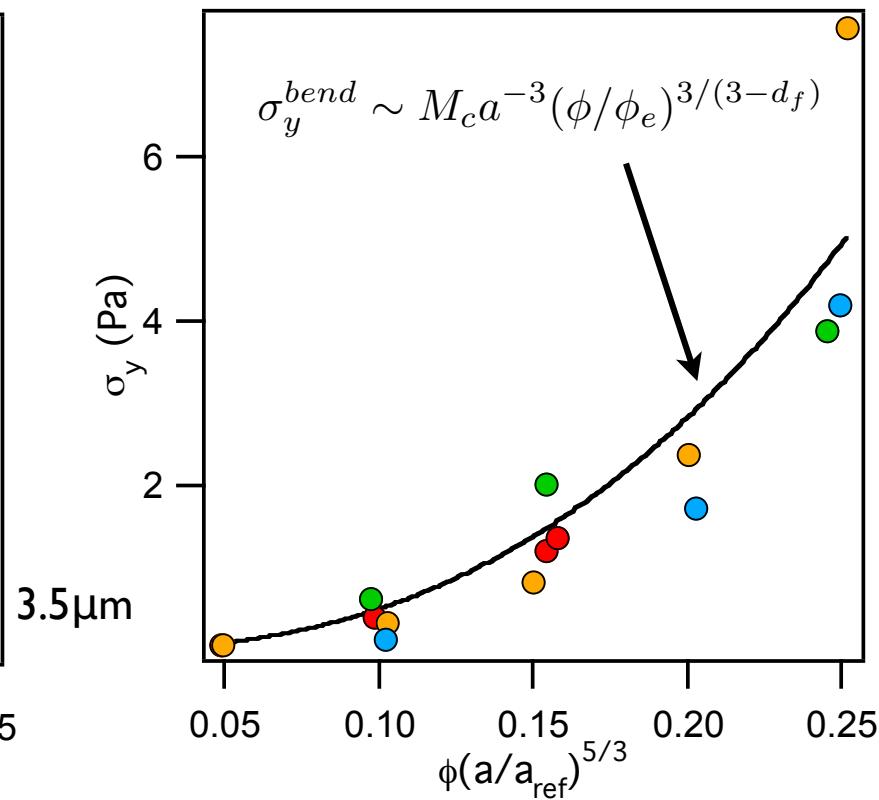
Comparison to bulk rheology

Bulk yield stress, PS latices



R. Buscall et al., J. Non-Newt. Fluid Mech., 24:183–202, 1987.

Expected from critical moment



If yield due to maximum tensile force, $\sigma_y \sim F_{tens} \xi^{-2} \approx 1\text{kPa}$